

# **Numerical simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V**

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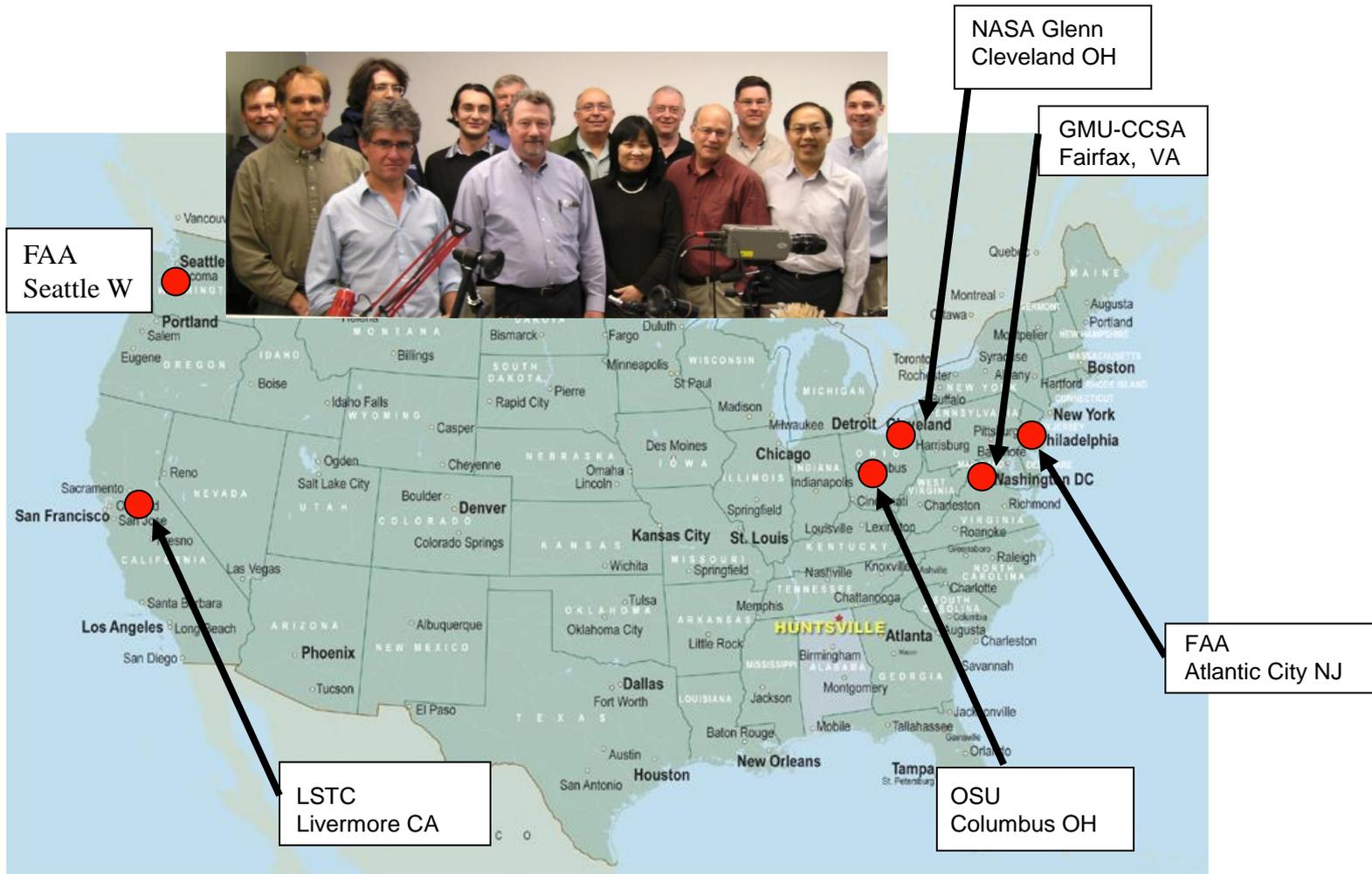
# Introduction



- **Part of a research program conducted by FAA William J Hughes Technical Center (NJ)**
- **material testing performed by OSU**
- **ballistic testing performed by NASA/GRC**
- **numerical simulations performed by GMU-CCSA**
  
- **involved the implementation in LS-DYNA of a tabulated generalisation of the Johnson-Cook material law with regularisation to accommodate simulation of failure in ductile materials : MAT\_224 or MAT\_TABULATED\_JOHNSON\_COOK**
  
- **previously published results in :**
  - **A Generalized, Three Dimensional Definition, Description and Derived Limits of the Triaxial Failure of Metals, Carney, DuBois, Buyuk, Kan, Earth&Sky, march 2008**



# FAA engine safety working group

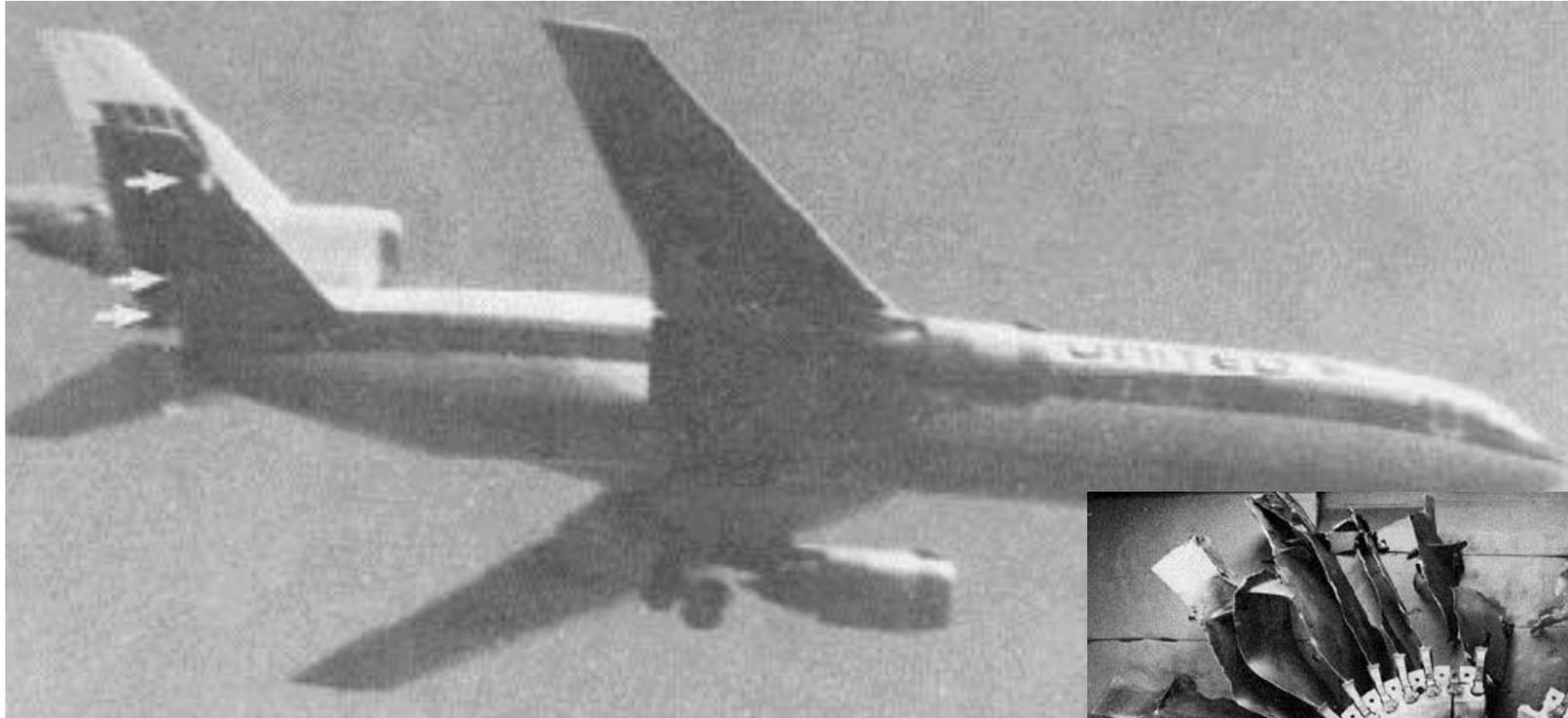


## Background : blade-out events

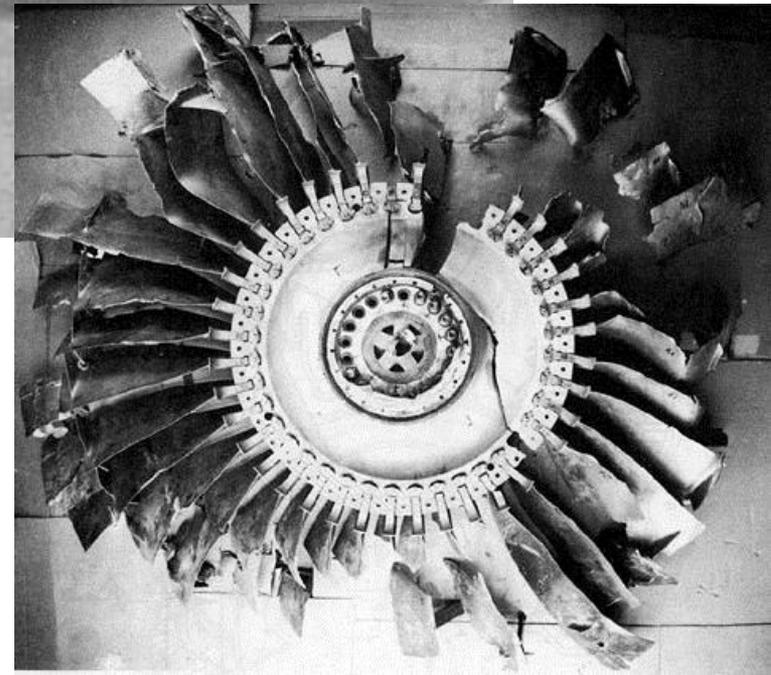
- **Aircraft Safety depends upon sound engine containment designs, and upon realistic evaluation of the damage from uncontained engine debris.**
- **The program addresses the modeling of impact between the blades and case, or between the fragments and non-engine aircraft structure**
- **The program has developed an extensive material test database and has modeled many different tests to evaluate the overall applicability of a single material model to the larger overall problem**

## Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

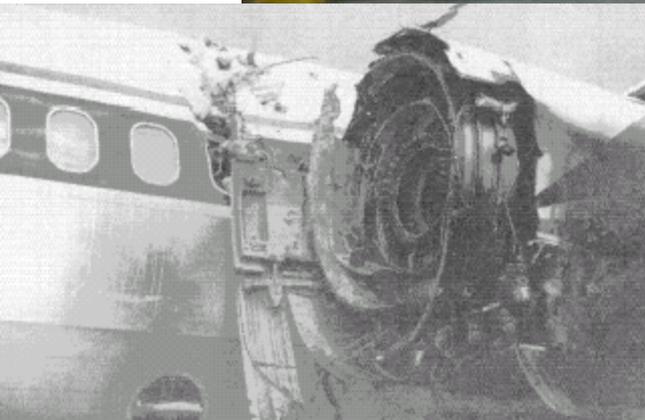
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**Sioux City, 1989**



## Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



**Pensacola RTO, 1996**

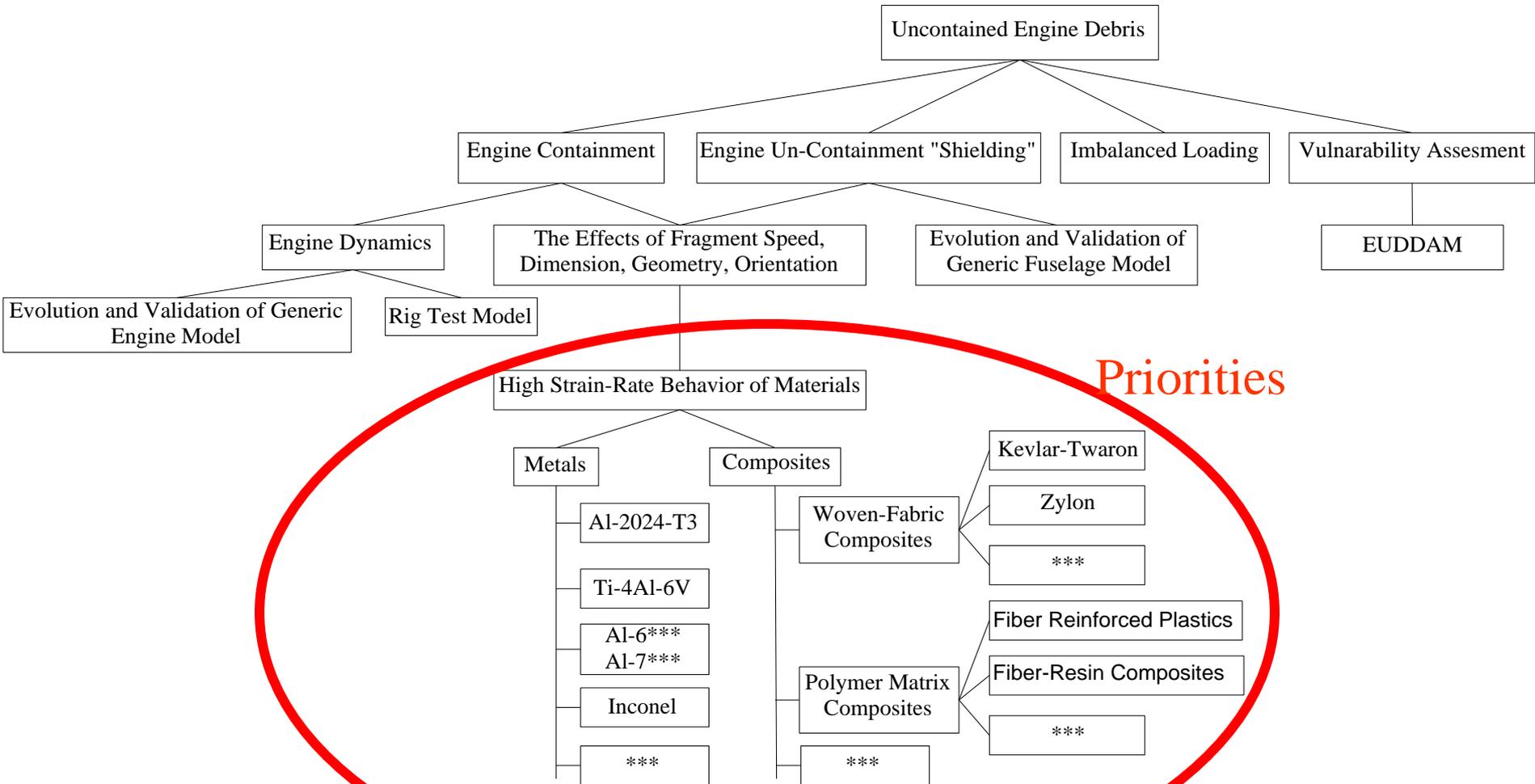
## Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



Fan blades of Trent

**Mandatory full scale engine containment test**

# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



## References

- Ravi Shriram Yatnalkar, *Experimental Investigation of Plastic Deformation of Ti-6Al-4V under Various Loading Conditions*, B.E, Thesis, The Ohio State University, 2010
- Jeremy Daniel Seidt, *Plastic Deformation and Ductile Fracture of 2024-T351 Aluminum under Various Loading Conditions*, Dissertation, The Ohio State University, 2010
- Jeremiah Thomas Hammer, *Plastic Deformation and Ductile fracture of Ti-6Al-4V under Various loading Conditions*, Dissertation, The Ohio State University, 2012
- Murat Buyuk, *Development of a tabulated thermo-viscoplastic material model with regularized failure for dynamic ductile failure prediction of structures under impact loading*, Dissertation, The George Washington University, 2013

## Development of MAT\_224 in LS-DYNA

- **started development in november 2006**
- **production version available in ls971-R4.2**
- **current presentation is based on implementation in ls971-R6.1**
- **developed on the basis of MAT\_024 with VP=1**
- **available for fully and underintegrated shell and solid elements**
- **full keyword code : \*MAT\_TABULATED\_JOHNSON\_COOK**

## Development history of MAT\_224

- R4-52798 ( june 2009 ) had the first official release of MAT\_224, load curve driven failure only
- R4-54523 : implementation of the NUMINP option by Tobias Erhardt from Dynamore
- R4-54921 : implementation of tabulated failure by Paul Du Bois and Murat Buyuk
- R5-59419 : ( july 2010 ) QA at LSTC by Gunther Blankenhorn
  
- **Further development driven by demand from industrial users :**
  
- R5-63444 : ( april 2011 ) optional coupling with EOS
- R6-68028 : E-modulus can be a function of temperature
- R6-68055 : ( july 2011 ) MAT\_224 is coupled with type 13 tetrahedrons
- R6-70523 : ( december 2011 ) tabulated regularisation
- R6-71932 : coupled with \*CONSTRAINED\_TIED\_NODES\_FAILURE
- R6-72543 : ( february 2012 ) NUMINT=-200 means no erosion will occur if d=1
- Dev-87525 : ( february 2014 ) BETA<0 refers to a load curve in function of strain rate

## **Development of a material and failure model for Ti-6-4**



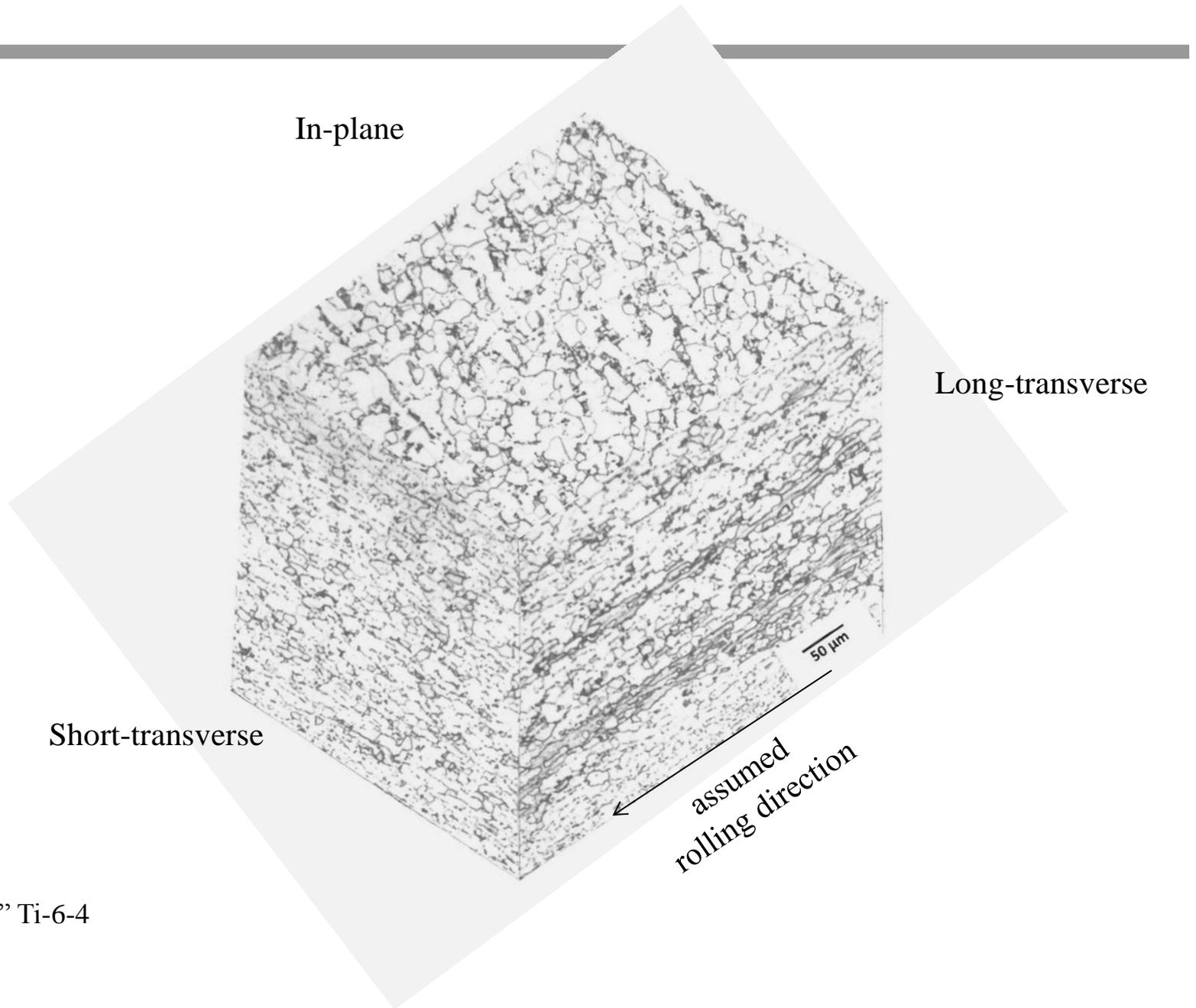
FAA-OSU-GWU-NASA Columbus 2007

**Part 1 : material model**

## **Careful material selection**

- **Material properties depend upon the microstructure : grain size, texture, orientation, surface quality...**
- **Manufacturing operations (rolling, annealing ) influence the microstructure and therefore the material properties**
- **Specific material models may be needed for individual components**
- **Some microstructural aspects of the material can be studied under the electron-microscope**

# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



## Ti-6-4 Texture

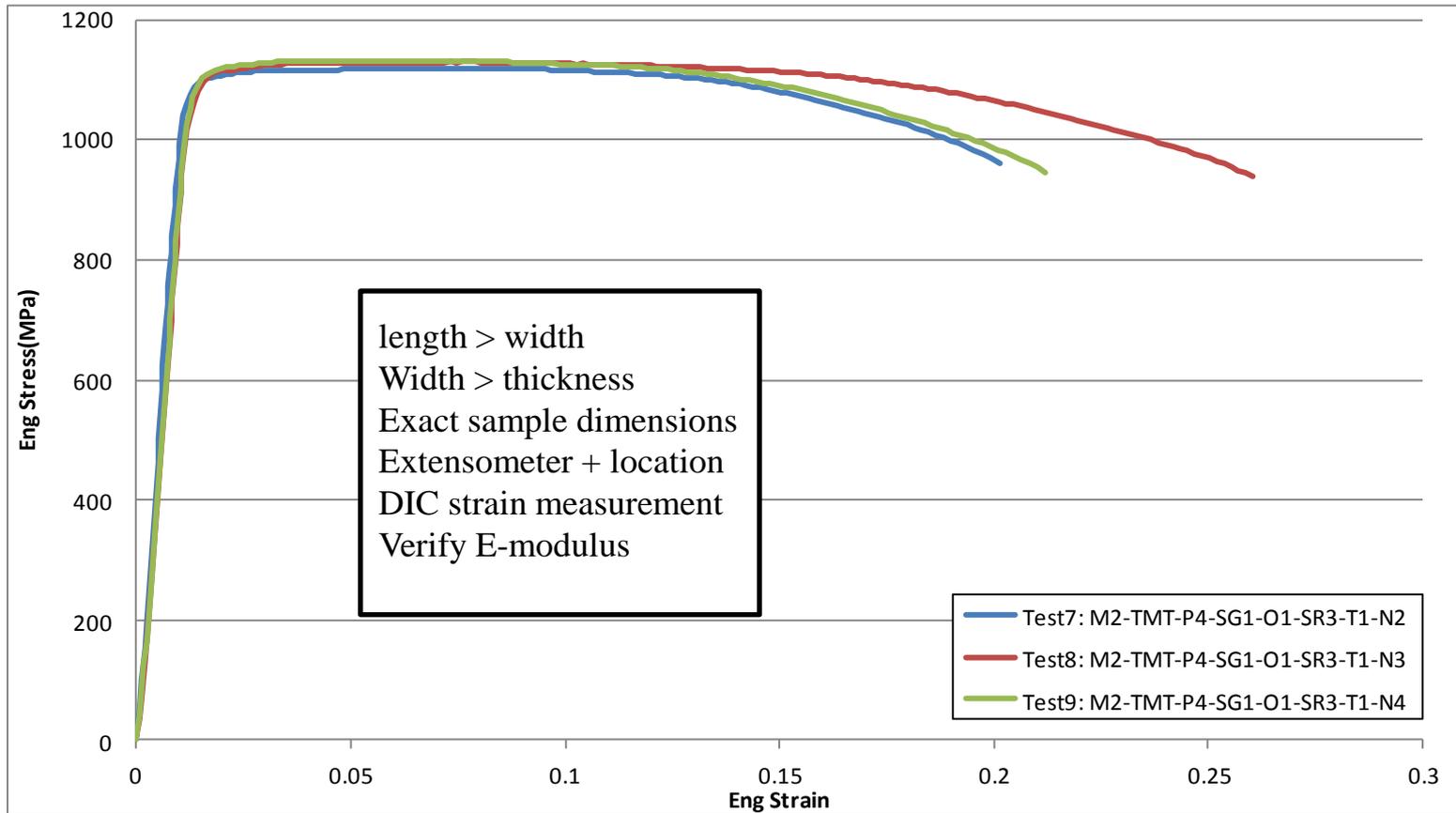
	Sample	Texture Index	Predominant Texture Type	Degrees off-center
Batch 2	BLM 45 0.270"	1.24	Transverse	31
	BLM 45 0.530"	1.13	Basal	27
	BLM 46 0.270"	1.33	Transverse	24
	BLM 47 0.425"	1.54	Transverse	-
Batch 1	0.09"	1.40	Transverse	-
	0.135"	1.17	Basal	30
	0.25"	1.48	Basal	18
	0.50"	1.03	Basal, Transverse	-

- **Texture index** denotes degree of anisotropy

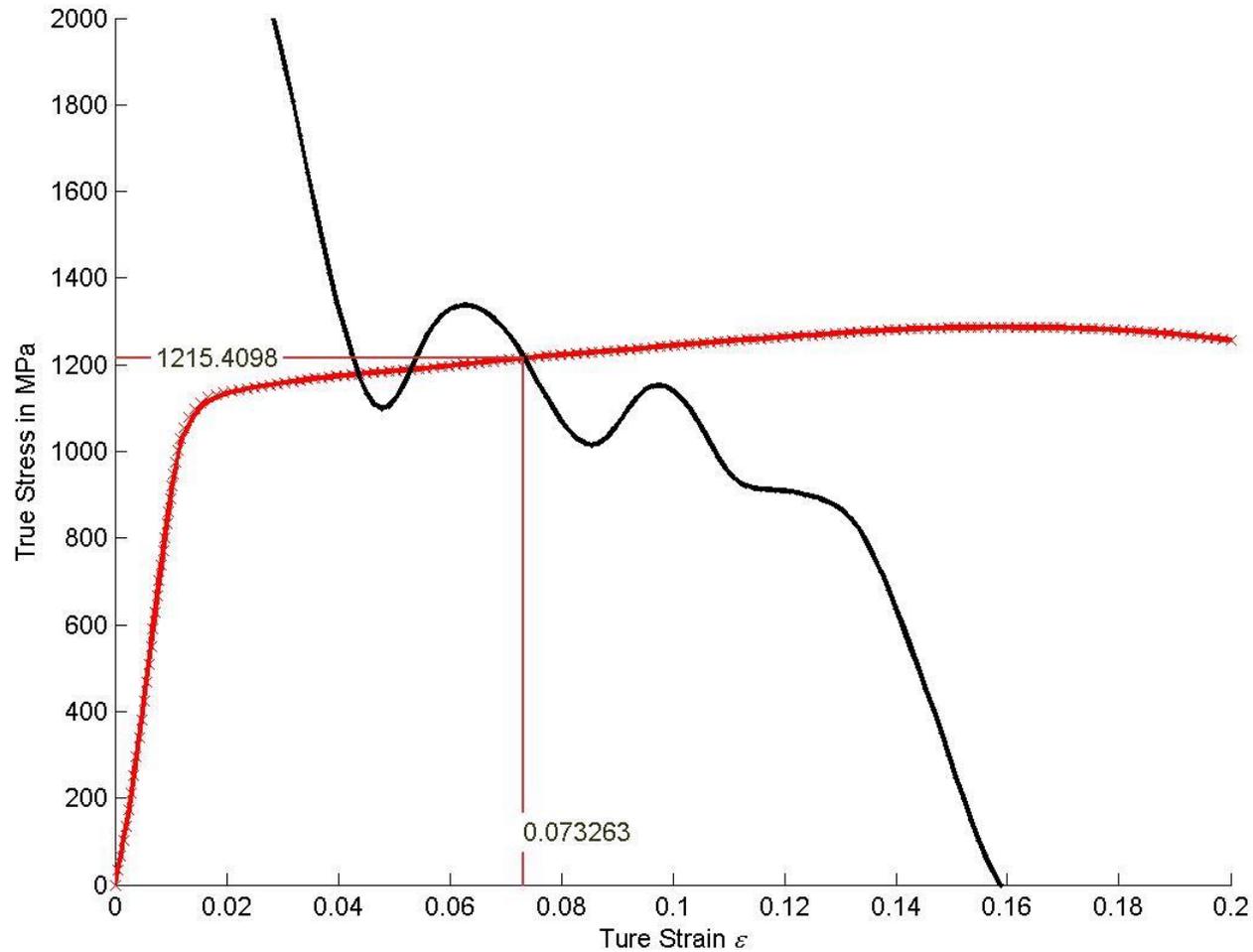
Index ranges from 1 (isotropic) to  $\infty$  (single crystal)

Aluminum foil (highly textured) has an index of 4.14

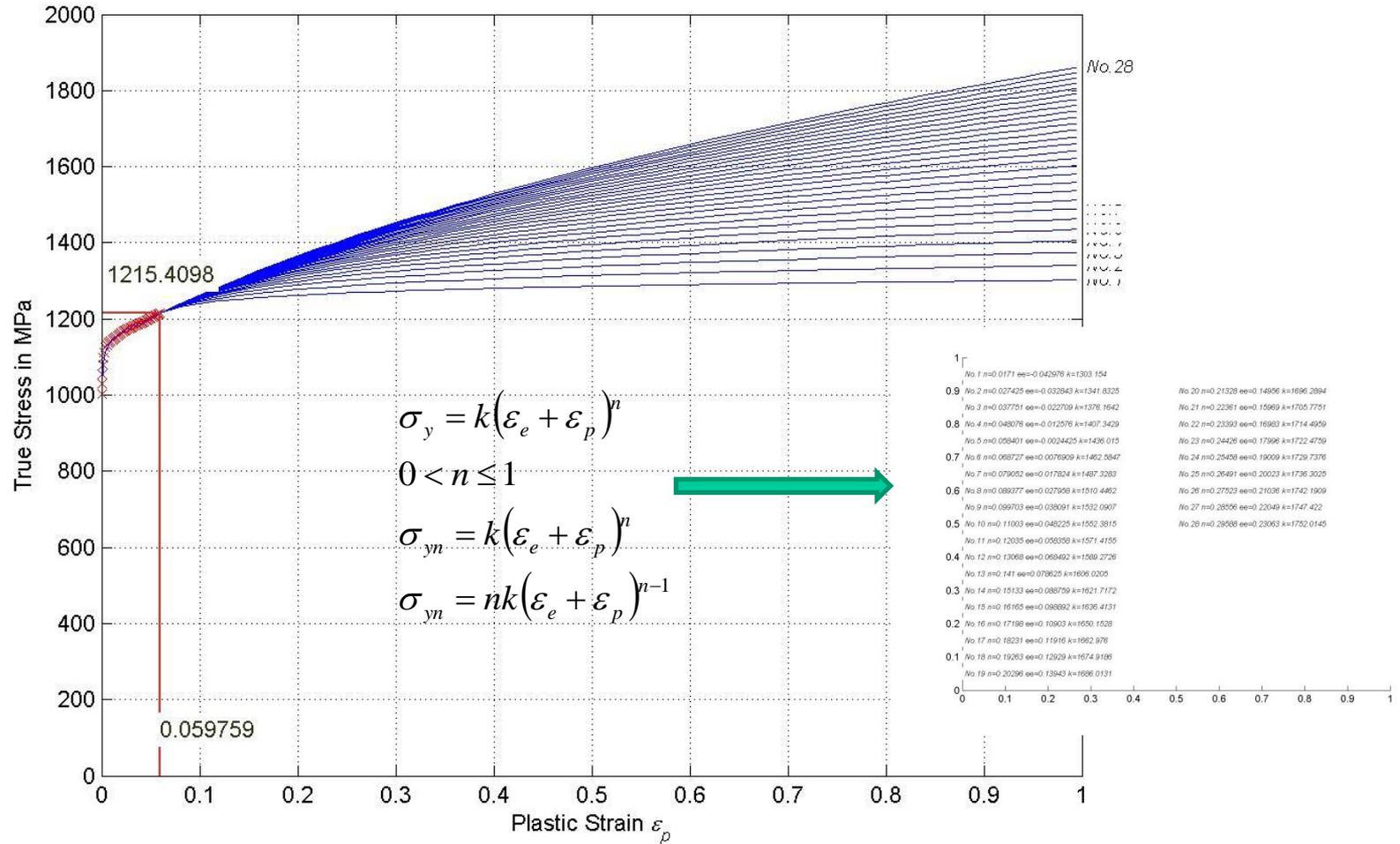
## Example of Ti-6-4 0.5" plate stock at strain rate = 1/s



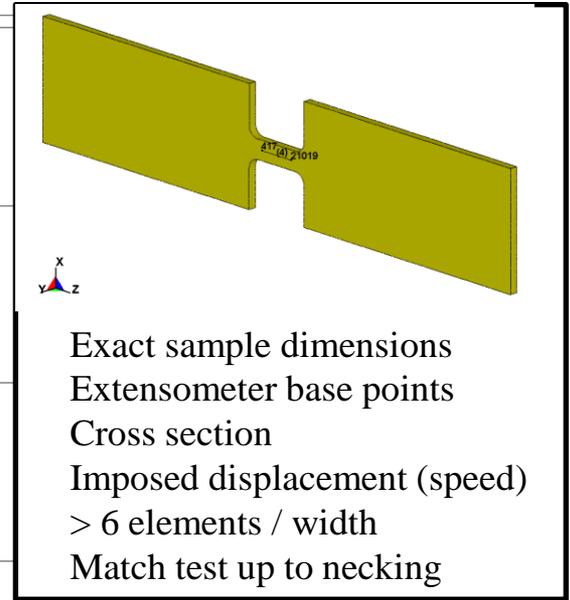
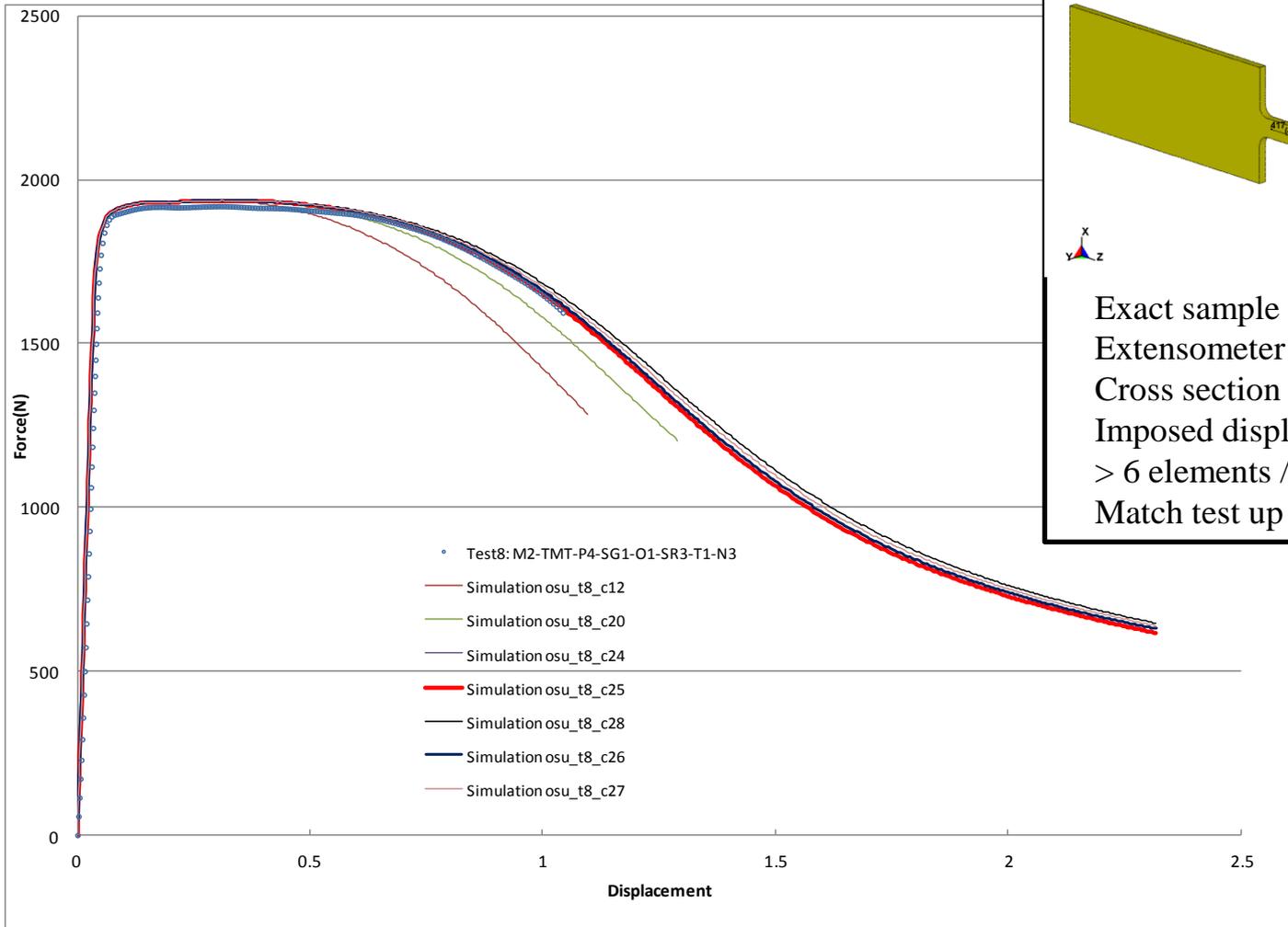
## Example of Ti-6-4 at strain rate = 1/s



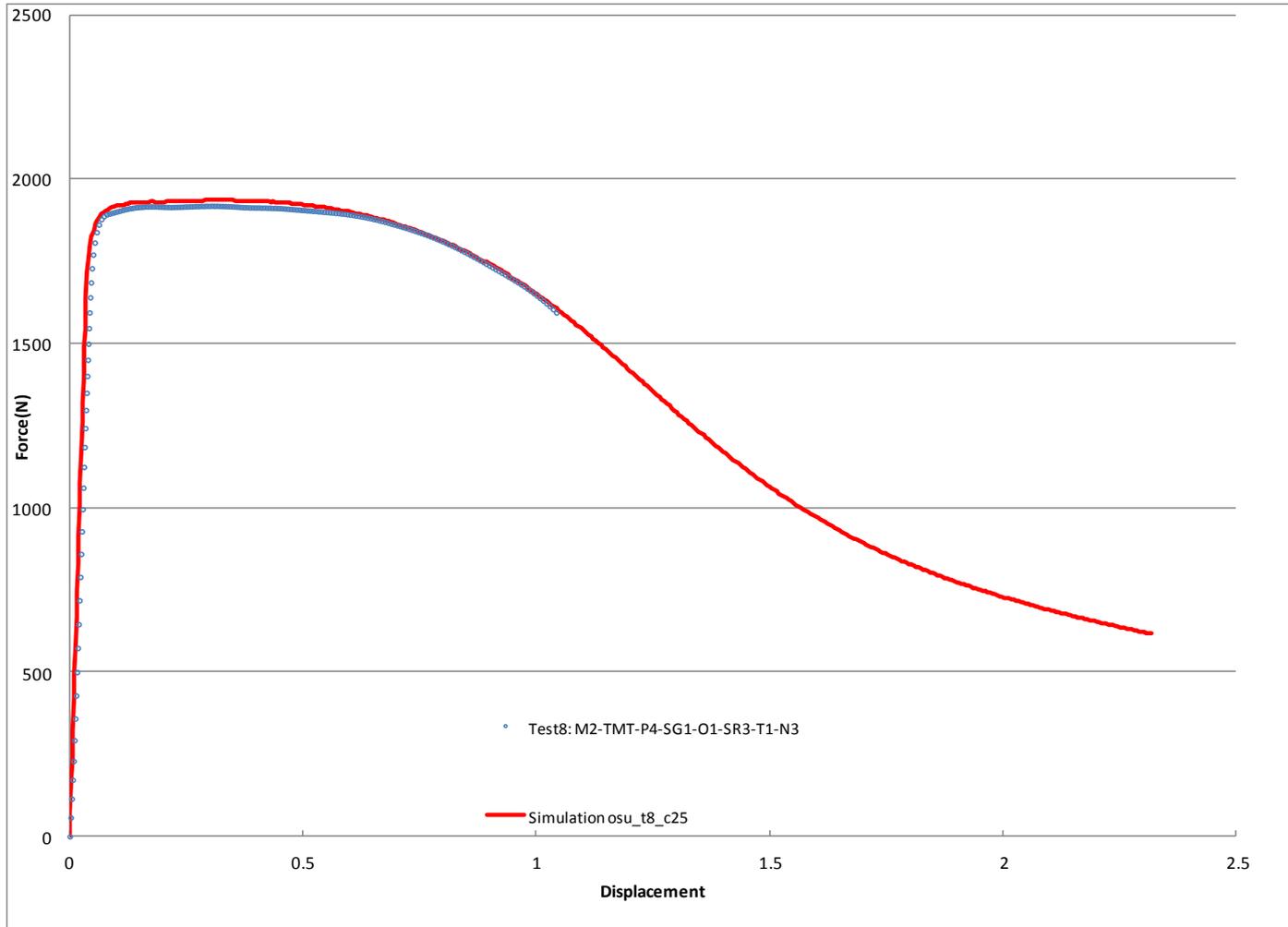
## Example of Ti-6-4 at strain rate = 1/s



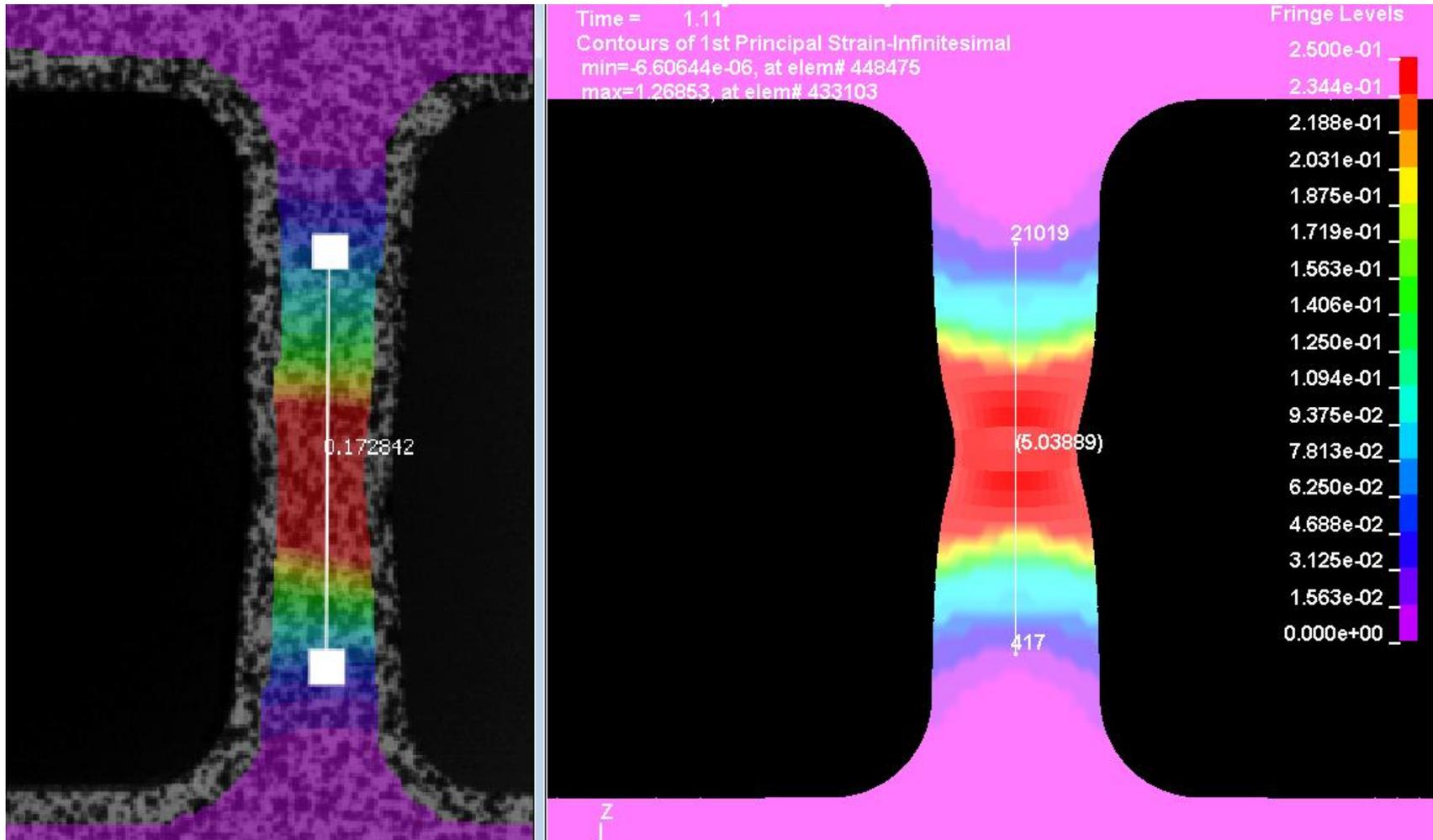
## Example of Ti-6-4 at strain rate = 1/s



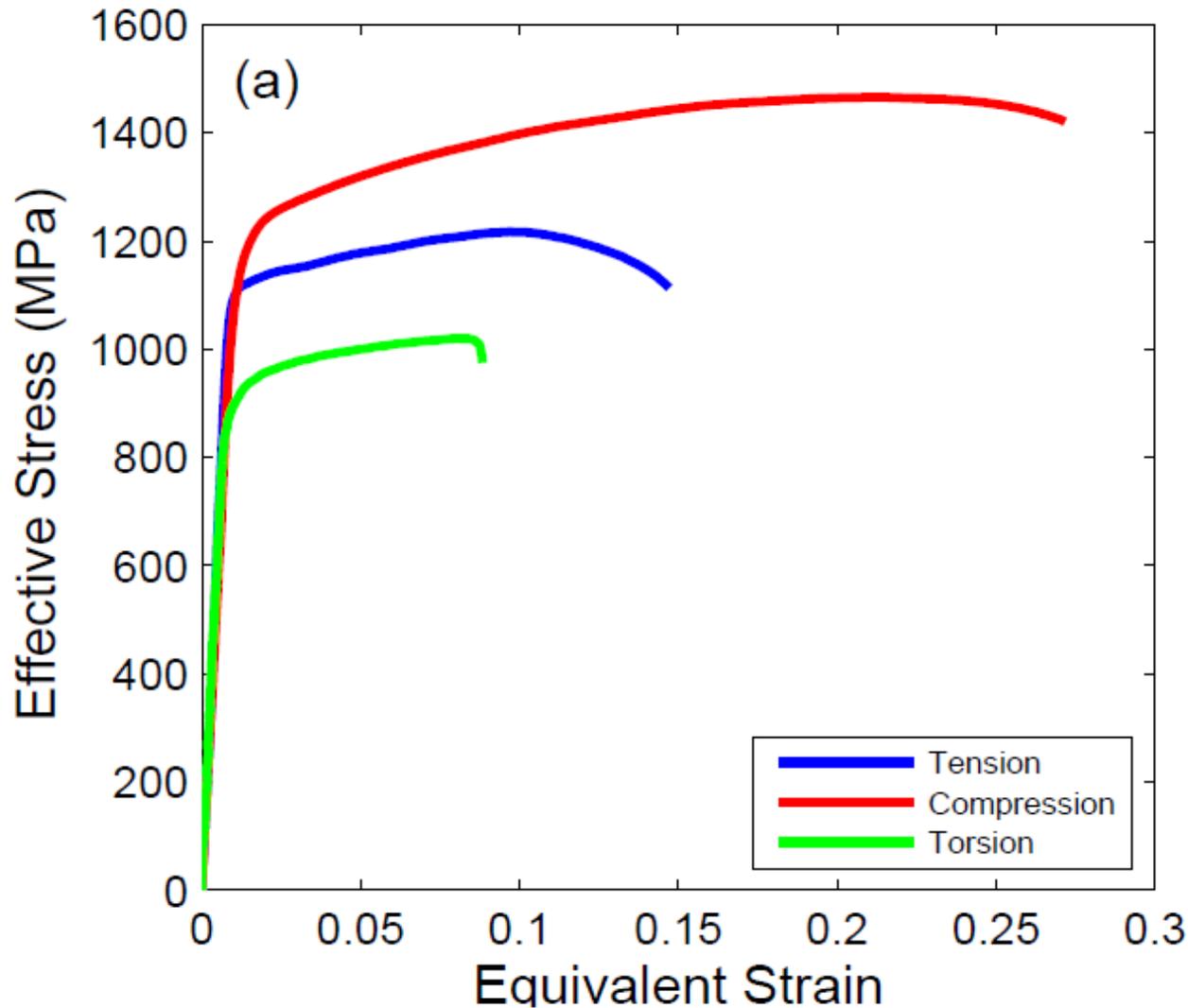
## Example of Ti-6-4 at strain rate = 1/s



## Example of Ti-6-4 at strain rate = 1/s



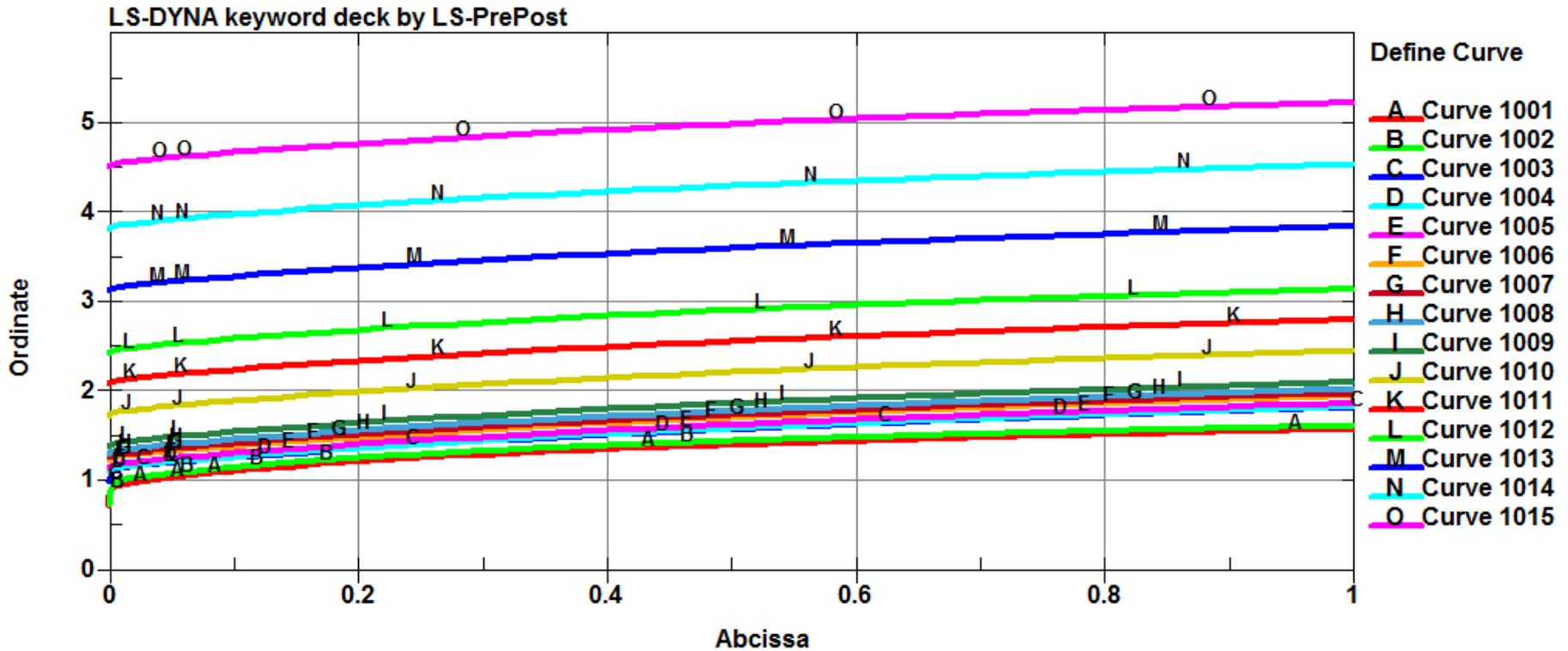
## Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



- Plate asymmetry
- MAT\_224 uses **tension data** to generate yield curve
  - Tension/Torsion will be over estimated
  - Compression/Torsion will be under estimated
- MAT\_224\_GYS allows for asymmetrical input



## Strain Rate Curves (table 1000)



Covered the range from 0.0001/s to 50000/s

## **conclusion**

- **Realistic material data are needed beyond necking (localisation) and up to failure if a predictive simulation with respect to failure is to be achieved**
- **A procedure was established allowing for reasonable accuracy as well as cost efficiency**
- **Value of DIC cannot be overestimated**

# **adiabatic instability**



**Shear bands**  
**Adiabatic vs isothermal instabilities**

## **adiabatic instabilities and strain localisation**

- **shear bands seem to be the main instability that is observed in high velocity impact phenomena**
- **plugging is caused by a thermoplastic shear instability, (so is the fragmentation of an exploding cylinder)**

## **Adiabatic instabilities : Shear bands**

- **Typical shear band widths are 10 to 100 microns**
- **High local values of shear strain : 5 to 100**
- **Ultra high local shear strain rates : 10000/s to 1000000/s**
- **Local temperature rises several hundred degrees**
- **High propagation speeds : typically 1000 m/s**
- **Not a crack**
- **Shear bands are precursors to rupture**
  
- **Cfr. Tzavaras**

## **Adiabatic instabilities : Localisation mechanism**

- **Under isothermal conditions metals strain harden**
- **Under high strain rates conditions change from isothermal to adiabatic**
- **If heat diffusion is too slow to equalize temperatures in the timescale of the loading then shear bands may occur**
- **Non-uniform strains induce non-uniform heating**
- **Hot spots are softer, cold spots are harder, this amplifies the non-uniformity of the strain field**
- **Two effects oppose the instability : heat diffusion and momentum diffusion induced by strain rate dependency**

## adiabatic instabilities and strain localisation

- Assume an elasto-plastic material with strain hardening and linear thermal softening :

$$\sigma_y = f(\varepsilon_p)h(T) = f(\varepsilon_p) \left( 1 - \frac{T - T_R}{T_M - T_R} \right)$$

- The maximum force criterion under plane stress conditions gives :

$$\frac{d\sigma_y}{d\varepsilon_p} = \sigma_y \frac{(4 - 3a - 3a^2 + 4a^3)}{4(1 + a^2 - a)^{3/2}} \quad a = \frac{\sigma_2}{\sigma_1}$$

- With special cases for uniaxial tension and pure shear :

$$\frac{d\sigma_y}{d\varepsilon_p} = \sigma_y \quad a = 0$$

$$\frac{d\sigma_y}{d\varepsilon_p} = 0 \quad a = -1$$

- The previous formulas contain total derivatives : yield stress is considered a function of equivalent plastic strain AND temperature

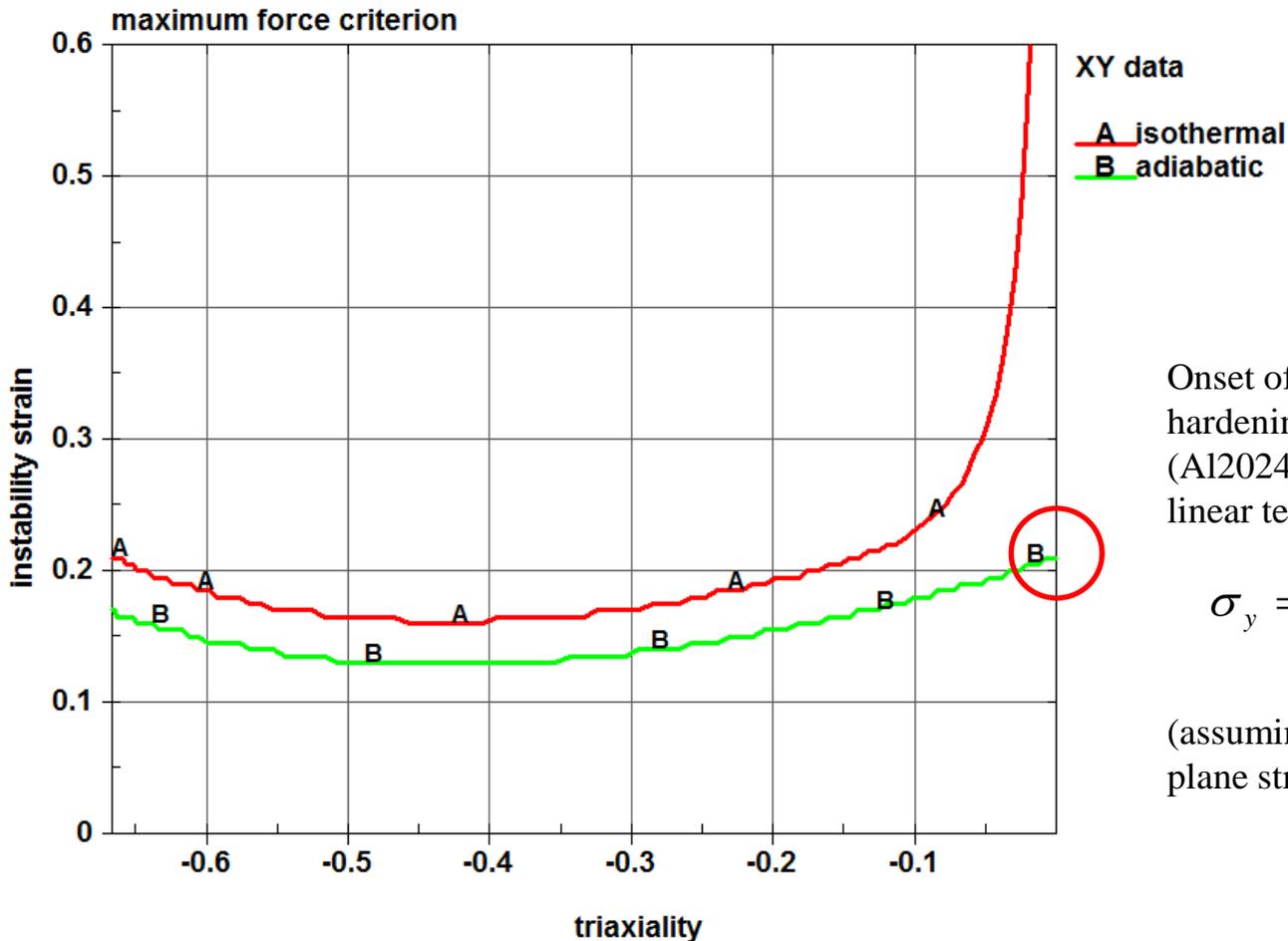
## adiabatic instabilities and strain localisation

- We can derive an adiabatic instability condition :

$$\left. \begin{aligned} \frac{d\sigma_y}{d\varepsilon_p} &= \sigma_y \frac{(4 - 3a - 3a^2 + 4a^3)}{4(1 + a^2 - a)^{3/2}} \\ \frac{d\sigma_y}{d\varepsilon_p} &= \frac{\partial\sigma_y}{\partial\varepsilon_p} + \frac{\partial\sigma_y}{\partial T} \frac{dT}{d\varepsilon_p} \end{aligned} \right\} \Rightarrow \frac{\partial\sigma_y}{\partial\varepsilon_p} = \sigma_y \frac{(4 - 3a - 3a^2 + 4a^3)}{4(1 + a^2 - a)^{3/2}} - \frac{\partial\sigma_y}{\partial T} \frac{dT}{d\varepsilon_p}$$

$$\left. \begin{aligned} dT &= \frac{dW}{\rho C_p} = \frac{\sigma_y d\varepsilon_p}{\rho C_p} \Rightarrow \frac{dT}{d\varepsilon_p} = \frac{\sigma_y}{\rho C_p} \\ \frac{\partial\sigma_y}{\partial T} &= -\frac{f(\varepsilon_p)}{T_M - T_R} \\ \frac{\partial\sigma_y}{\partial\varepsilon_p} &= \frac{\partial f(\varepsilon_p)}{\partial\varepsilon_p} \frac{T - T_R}{T_M - T_R} \end{aligned} \right\} \Rightarrow \frac{\partial f(\varepsilon_p)}{\partial\varepsilon_p} = f(\varepsilon_p) \frac{(4 - 3a - 3a^2 + 4a^3)}{4(1 + a^2 - a)^{3/2}} + \frac{f^2(\varepsilon_p)}{(T_M - T_R)\rho C_p}$$

## adiabatic instabilities and strain localisation



Onset of instability for a strain hardening elasto-plastic material (Al2024) with linear temperature dependency :

$$\sigma_y = f(\varepsilon_p) \frac{T - T_R}{T_M - T_R}$$

(assuming small elastic strains and plane stress )

## adiabatic instabilities and strain localisation

- Analytical solution for the onset of shear banding in the case of a Johnson-Cook law with linear temperature dependency :

$$\left. \begin{aligned} \frac{d\sigma_y}{d\varepsilon_p} &= 0 \\ \frac{d\sigma_y}{d\varepsilon_p} &= \frac{\partial\sigma_y}{\partial\varepsilon_p} + \frac{\partial\sigma_y}{\partial T} \frac{dT}{d\varepsilon_p} \end{aligned} \right\} \Rightarrow \frac{\partial\sigma_y}{\partial\varepsilon_p} = -\frac{\partial\sigma_y}{\partial T} \frac{dT}{d\varepsilon_p}$$

$$dT = \frac{dW}{\rho C_p} = \frac{\sigma_y d\varepsilon_p}{\rho C_p} \Rightarrow \frac{dT}{d\varepsilon_p} = \frac{a + b\varepsilon_p^n}{\rho C_p}$$

$$\frac{\partial\sigma_y}{\partial T} = -\frac{a + b\varepsilon_p^n}{T_M - T_R} \quad (\varepsilon_p = \text{cte})$$

$$\Rightarrow \frac{\partial\sigma_y}{\partial\varepsilon_p} = -\frac{\partial\sigma_y}{\partial T} \frac{dT}{d\varepsilon_p} \Rightarrow bn\varepsilon_p^{n-1} = \frac{(a + b\varepsilon_p^n)^2}{(T_M - T_R)\rho C}$$

## adiabatic instabilities and strain localisation

- Analytical solution for the onset of adiabatic diffuse necking in the case of a Johnson-Cook law with linear temperature dependency :

$$\left. \begin{aligned} \frac{d\sigma_y}{d\varepsilon_p} &= \sigma_y \\ \frac{d\sigma_y}{d\varepsilon_p} &= \frac{\partial\sigma_y}{\partial\varepsilon_p} + \frac{\partial\sigma_y}{\partial T} \frac{dT}{d\varepsilon_p} \end{aligned} \right\} \Rightarrow \frac{\partial\sigma_y}{\partial\varepsilon_p} = \sigma_y - \frac{\partial\sigma_y}{\partial T} \frac{dT}{d\varepsilon_p}$$

$$dT = \frac{dW}{\rho C_p} = \frac{\sigma_y d\varepsilon_p}{\rho C_p} \Rightarrow \frac{dT}{d\varepsilon_p} = \frac{a + b\varepsilon_p^n}{\rho C_p}$$

$$\frac{\partial\sigma_y}{\partial T} = -\frac{a + b\varepsilon_p^n}{T_M - T_R} \quad (\varepsilon_p = cte)$$

$$\Rightarrow \frac{\partial\sigma_y}{\partial\varepsilon_p} = \sigma_y - \frac{\partial\sigma_y}{\partial T} \frac{dT}{d\varepsilon_p} \Rightarrow bn\varepsilon_p^{n-1} = (a + b\varepsilon_p^n) + \frac{(a + b\varepsilon_p^n)^2}{(T_M - T_R)\rho C}$$

## Material instabilities and strain localisation

- example for quasistatic localisation plastic strain values :

$$\left. \begin{array}{l} a = 0.2GPa \\ b = 0.6GPa \\ n = 0.4 \\ T_M - T_R = 775 - 293 \\ \rho = 7.8E - 6kg / mm^3 \\ C_p = 445kJ / kg^\circ K \end{array} \right\} \Rightarrow \left\{ \begin{array}{ll} \varepsilon_p = 0.254 & \text{isothermal tension} \\ \varepsilon_p = 0.186 & \text{adiabatic tension} \\ \varepsilon_p = 0.299 & \text{adiabatic shear} \end{array} \right.$$

## conclusion

- Triaxiality dependent regularisation needs to be adapted to the conditions of the impact ( temperature, rate...)
- Every material needs an individual assessment with respect to occurrence of plastic instability
- **For a predictive simulation with respect to failure, the mesh size must take the width of the shear band into account**
- The width of the shear band depends a.o. on the hardness and strength of the material ( see reference )
- Depending on the width of the shear band ( 10 to 50 micron ) the number of elements over the thickness of the target plate needed for a failure simulation can be 100 or more ( see reference )

## **references**

- **Are numerical simulations of ballistic impact predictive ? Borvik, Hopperstad, Langseth, Berstad, 2011**

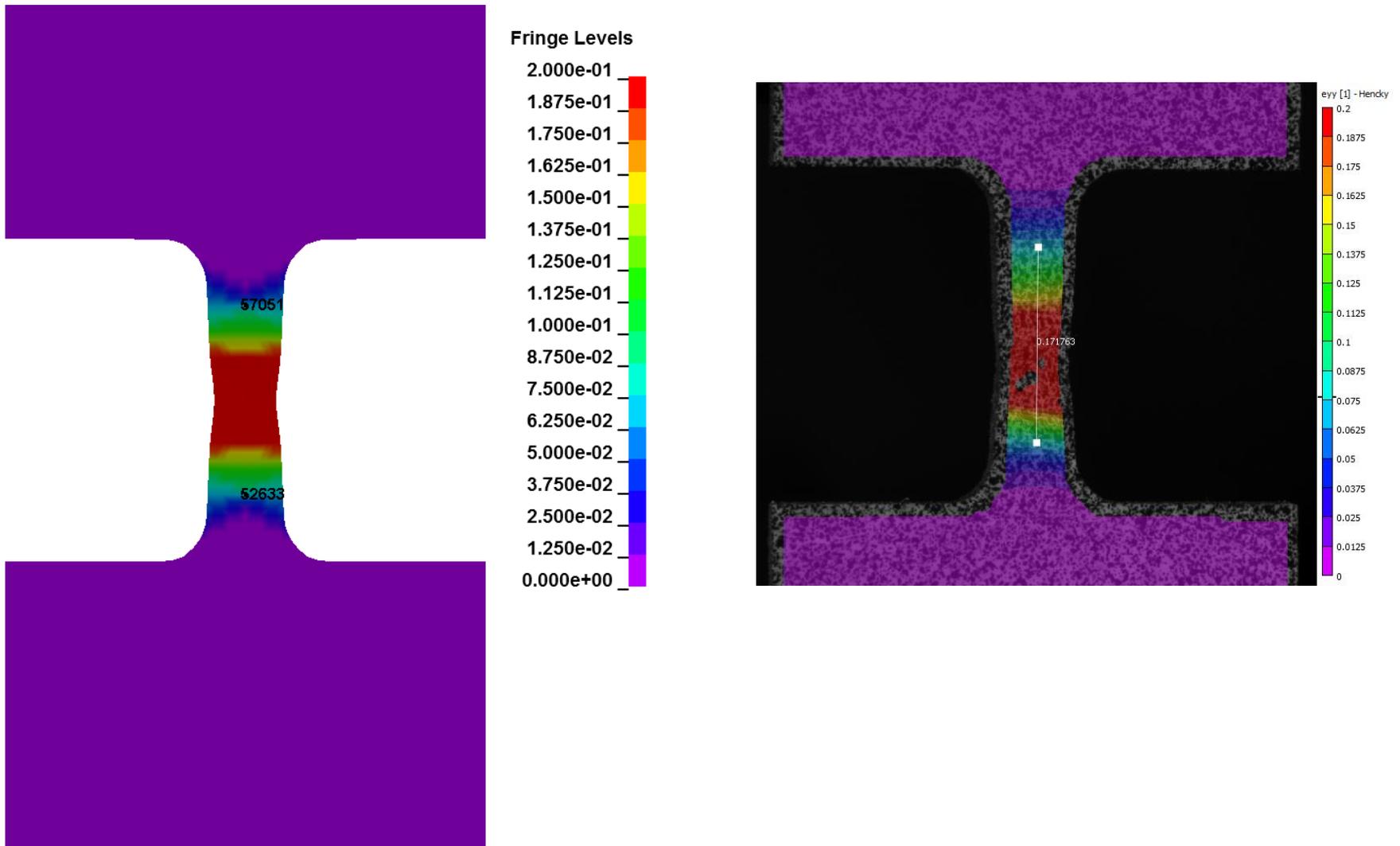
## **Development of a material and failure model for Ti-6-4**



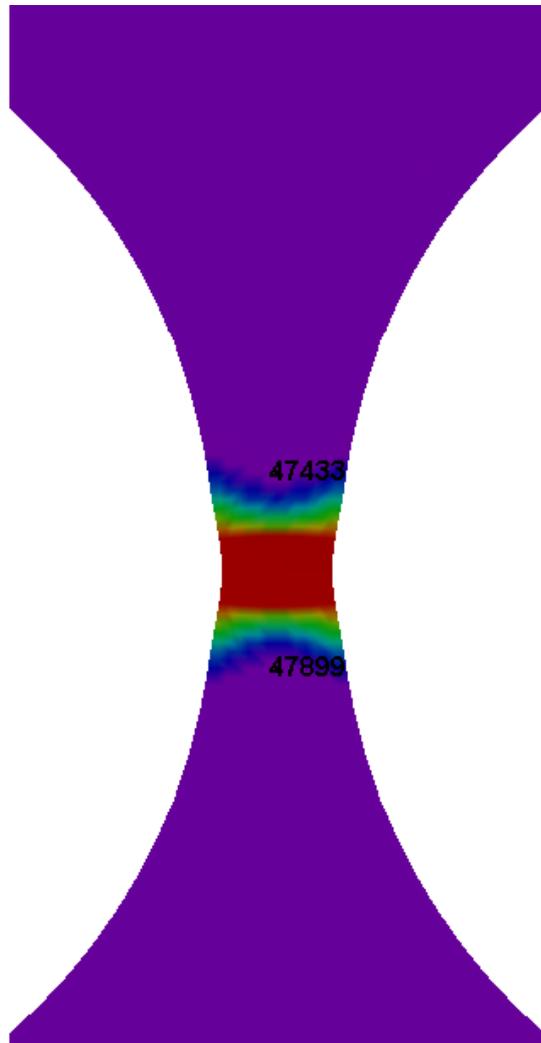
FAA-OSU-GWU-NASA Columbus 2007

**Part 2 : failure model**

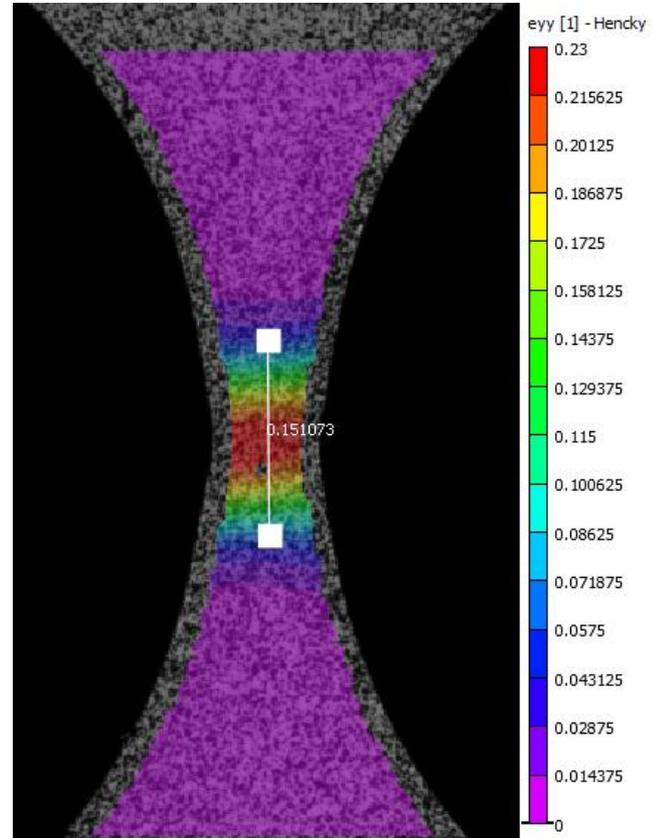
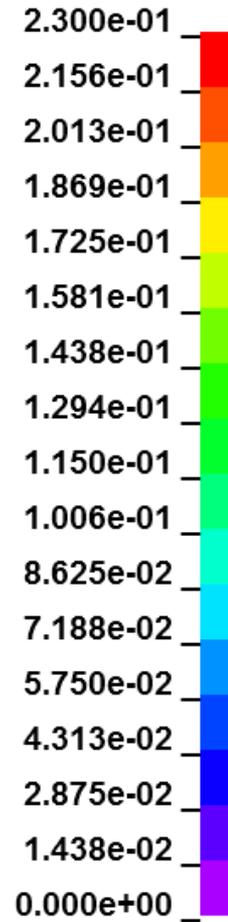
## DIC Comparison – Plane Stress 1



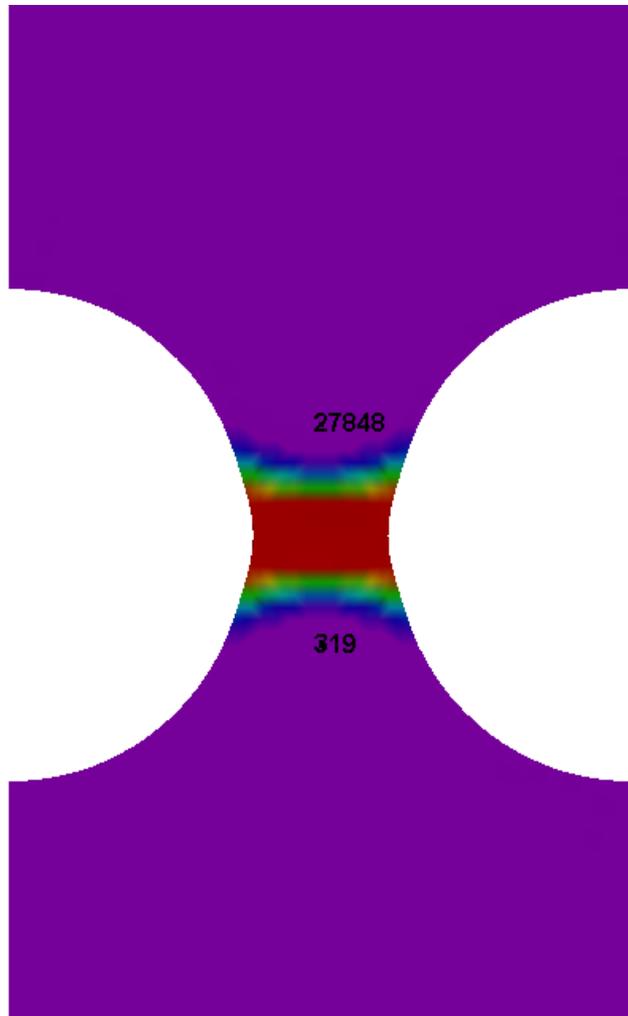
## DIC Comparison – Plane Stress 2



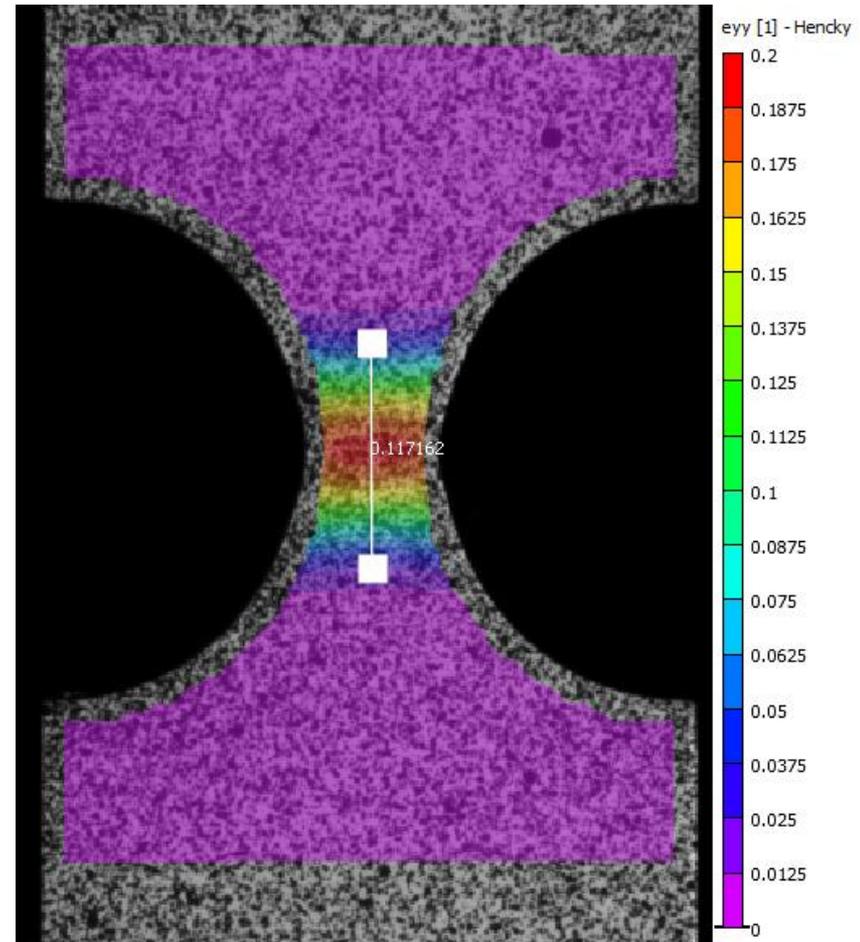
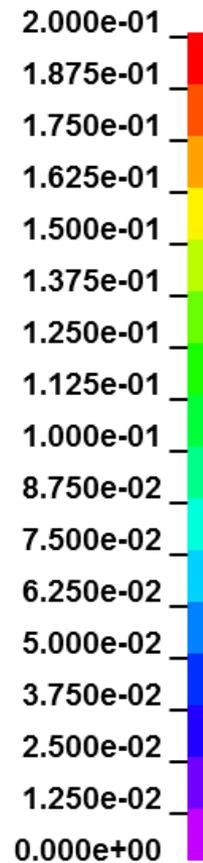
Fringe Levels



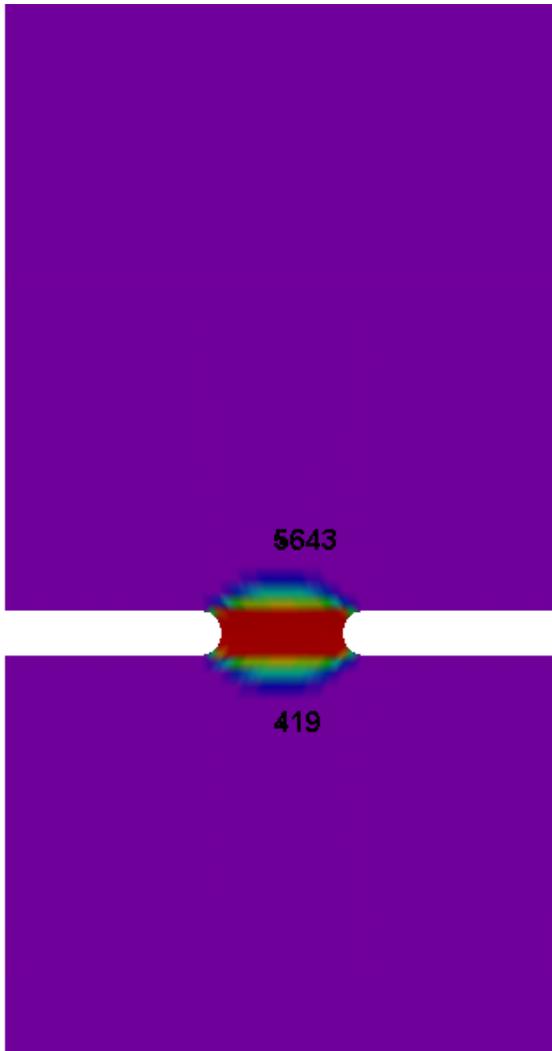
## DIC Comparison – Plane Stress 3



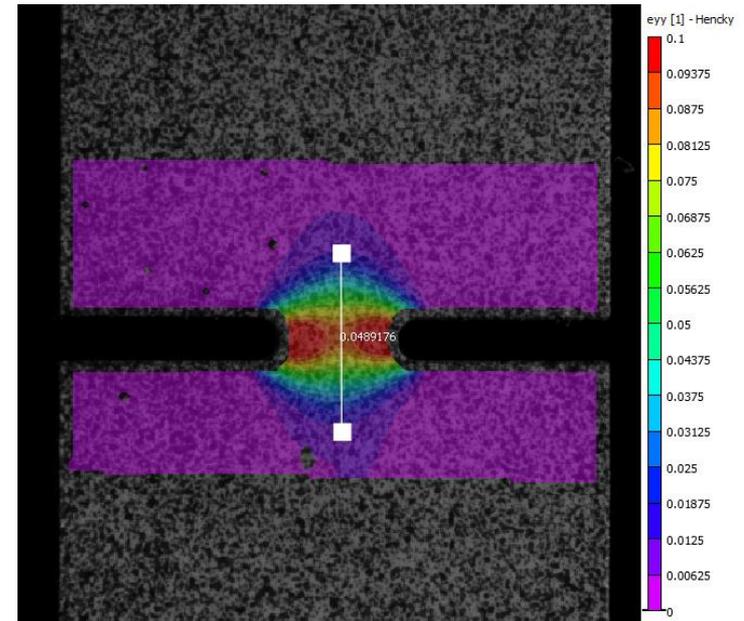
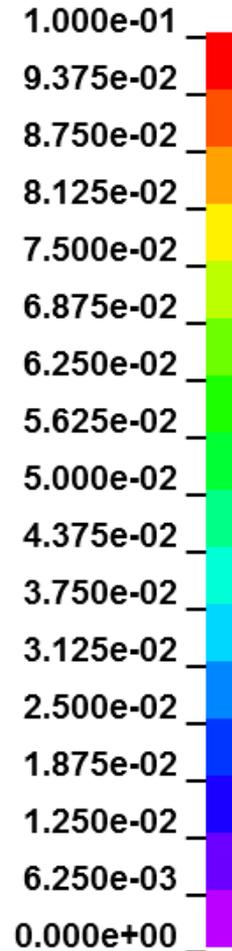
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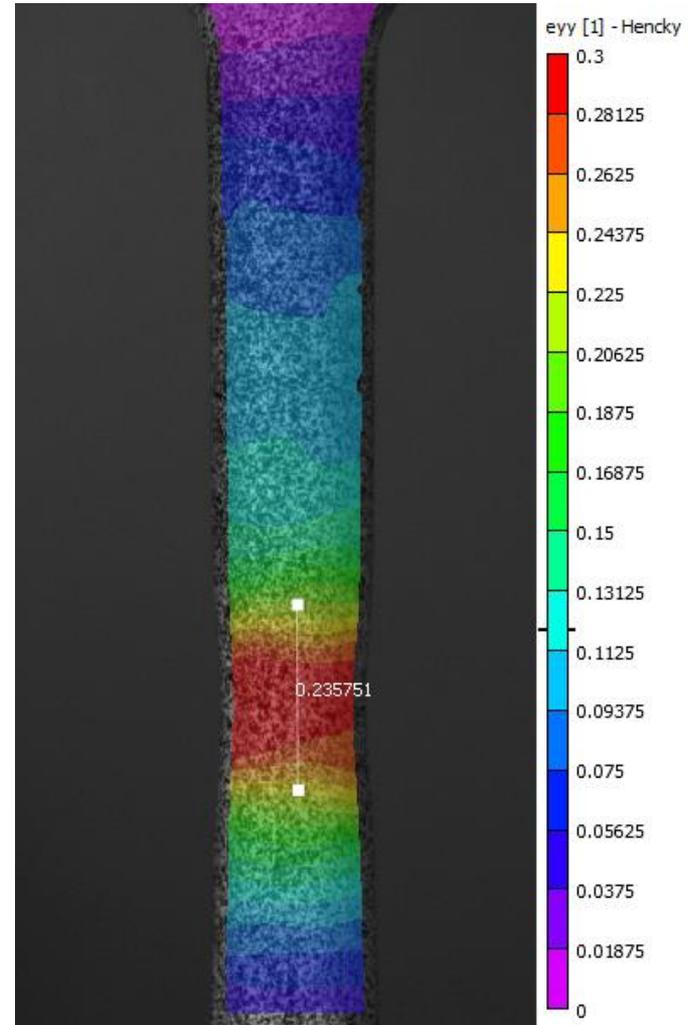
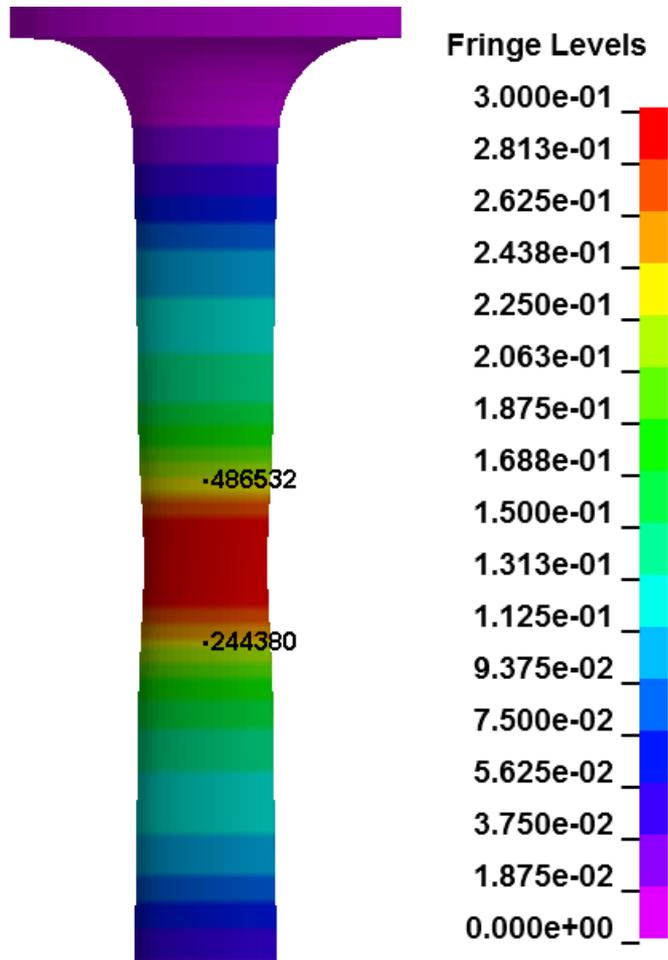
## DIC Comparison – Plane Stress 4



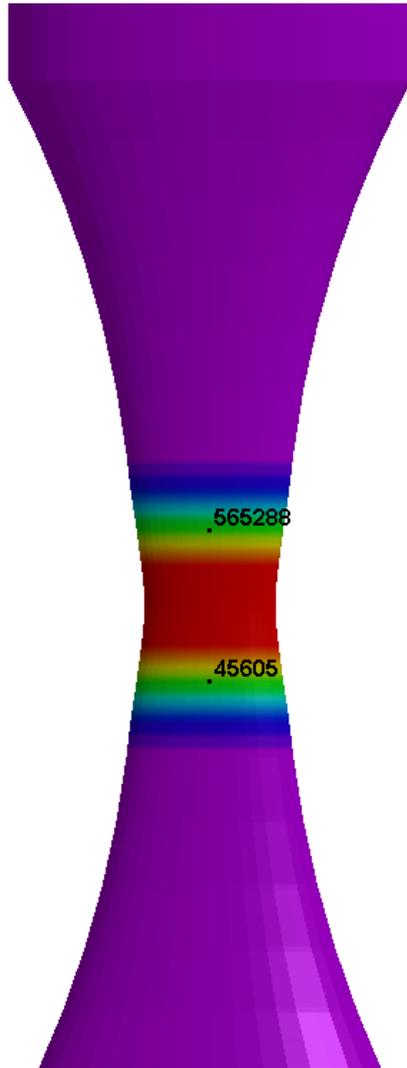
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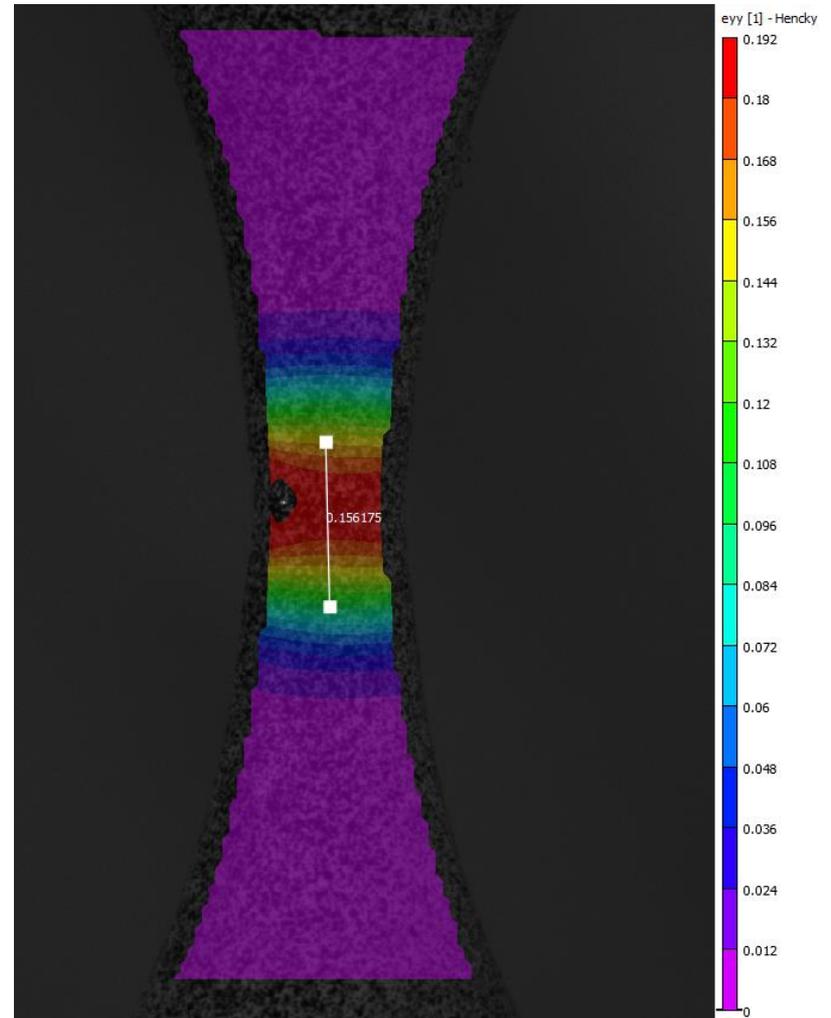
## DIC Comparison – Axisymmetric 1



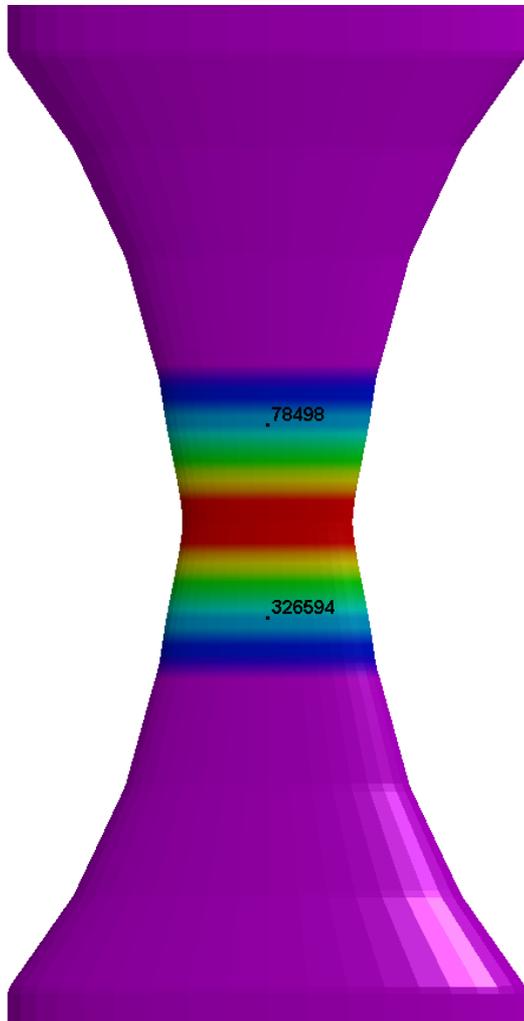
## DIC Comparison – Axisymmetric 2



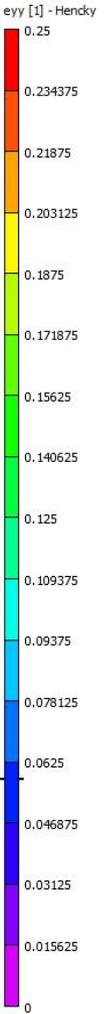
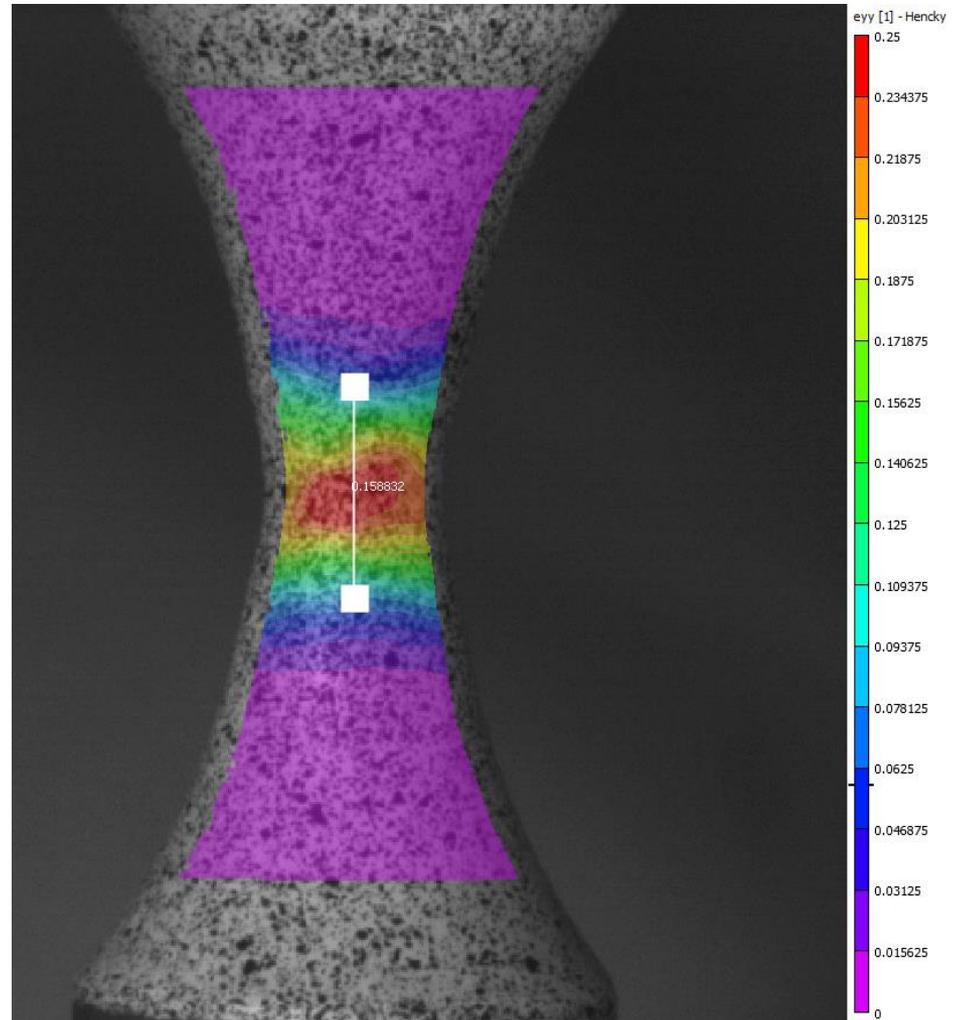
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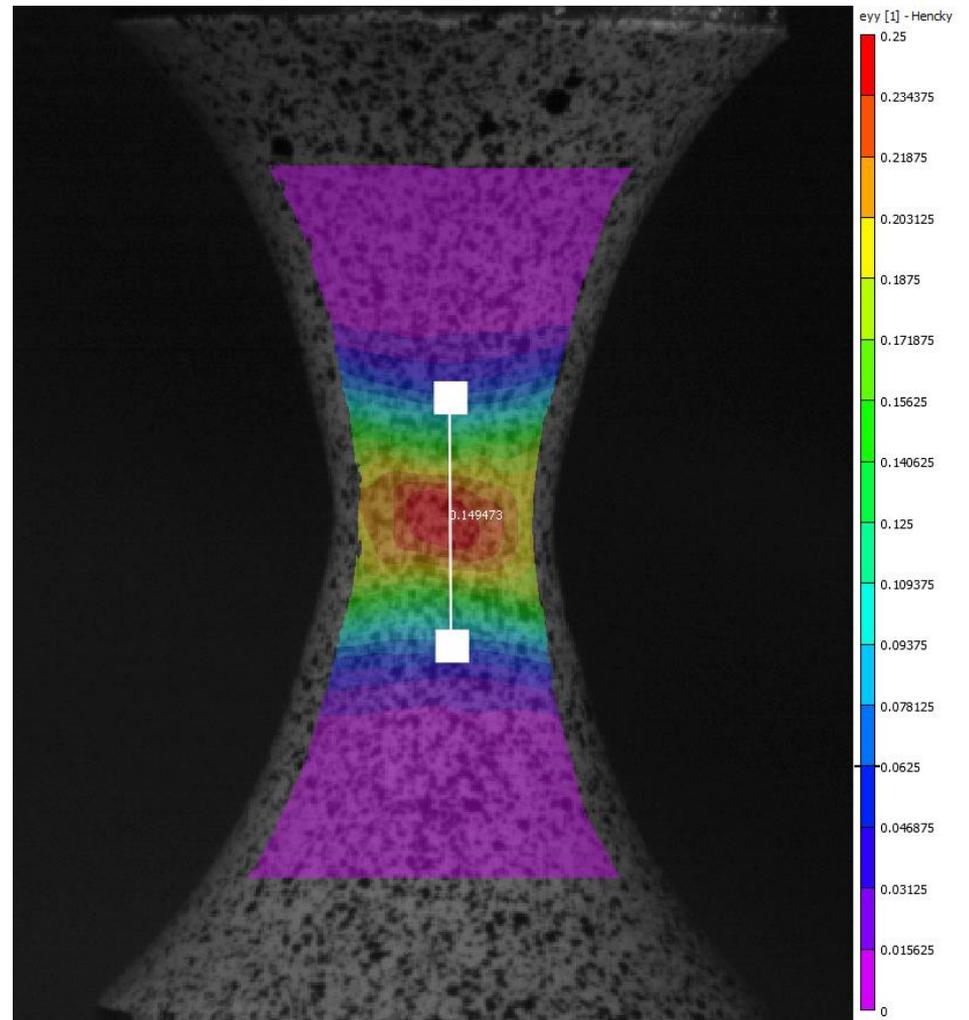
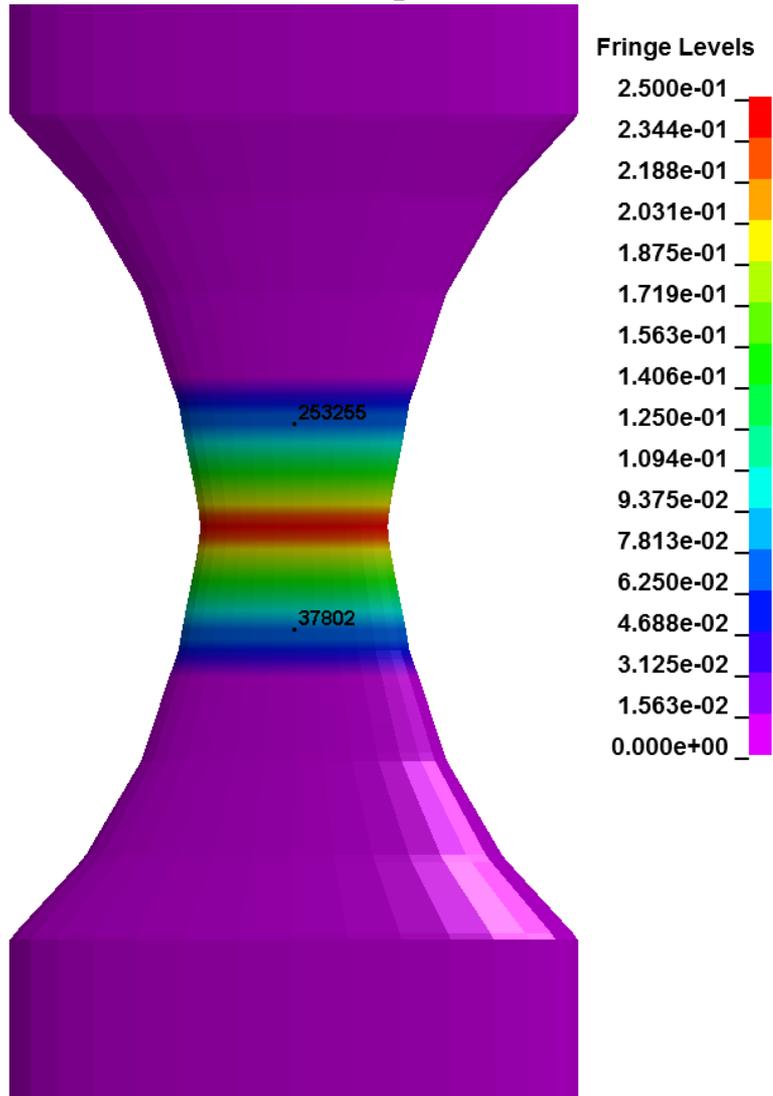
## DIC Comparison – Axisymmetric 3



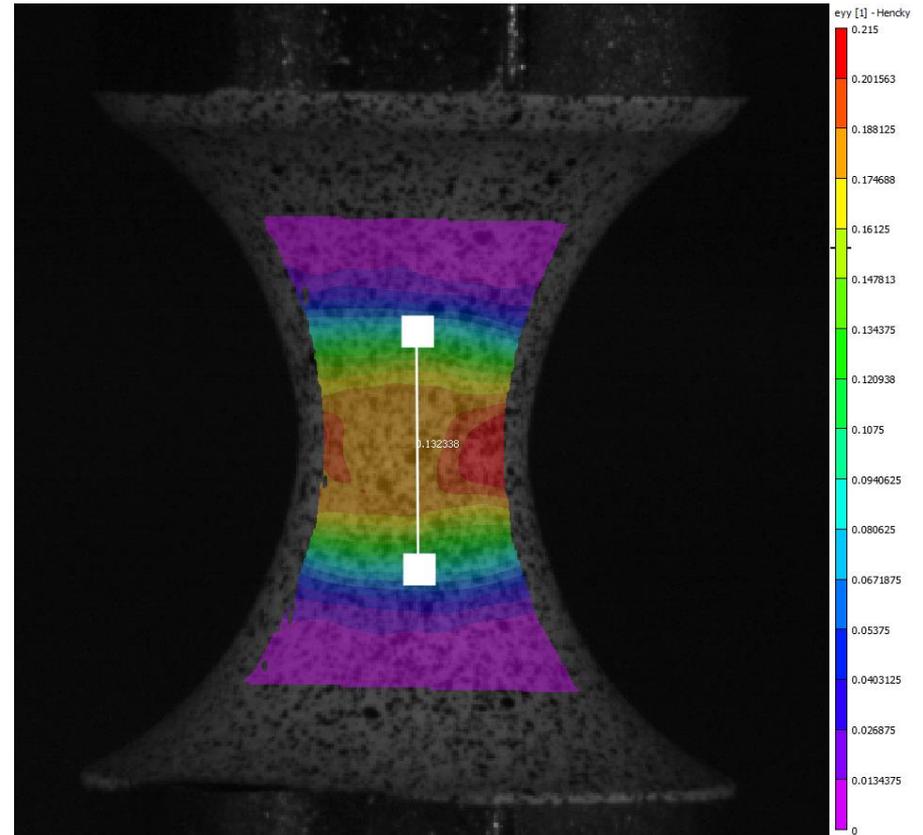
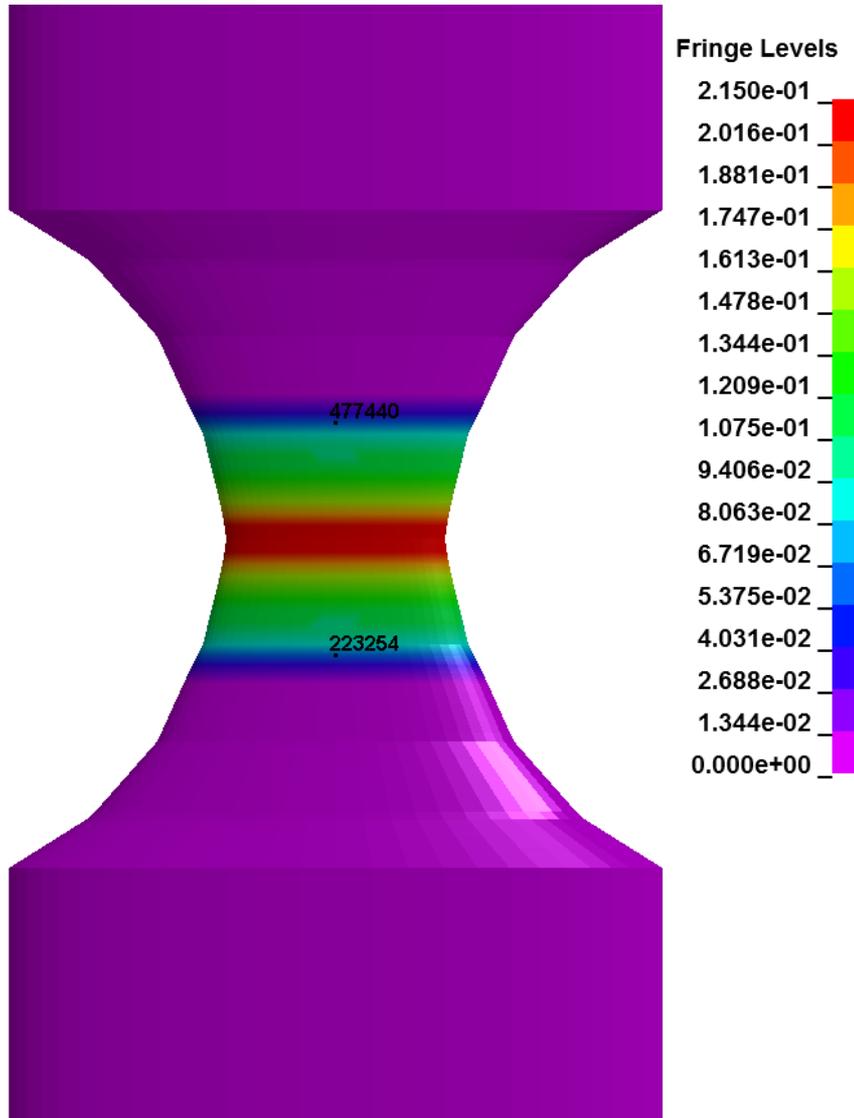
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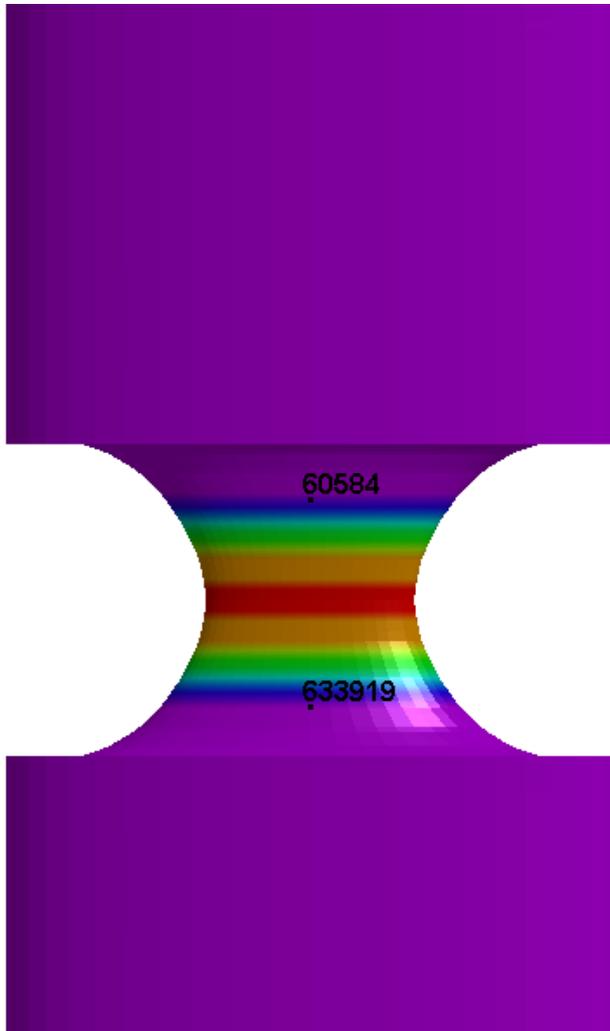
## DIC Comparison – Axisymmetric 4



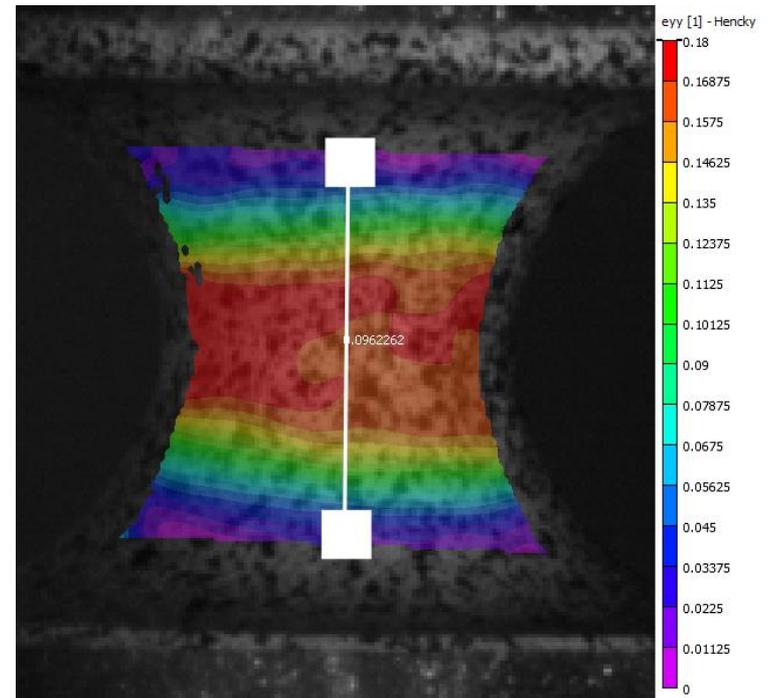
## DIC Comparison – Axisymmetric 5



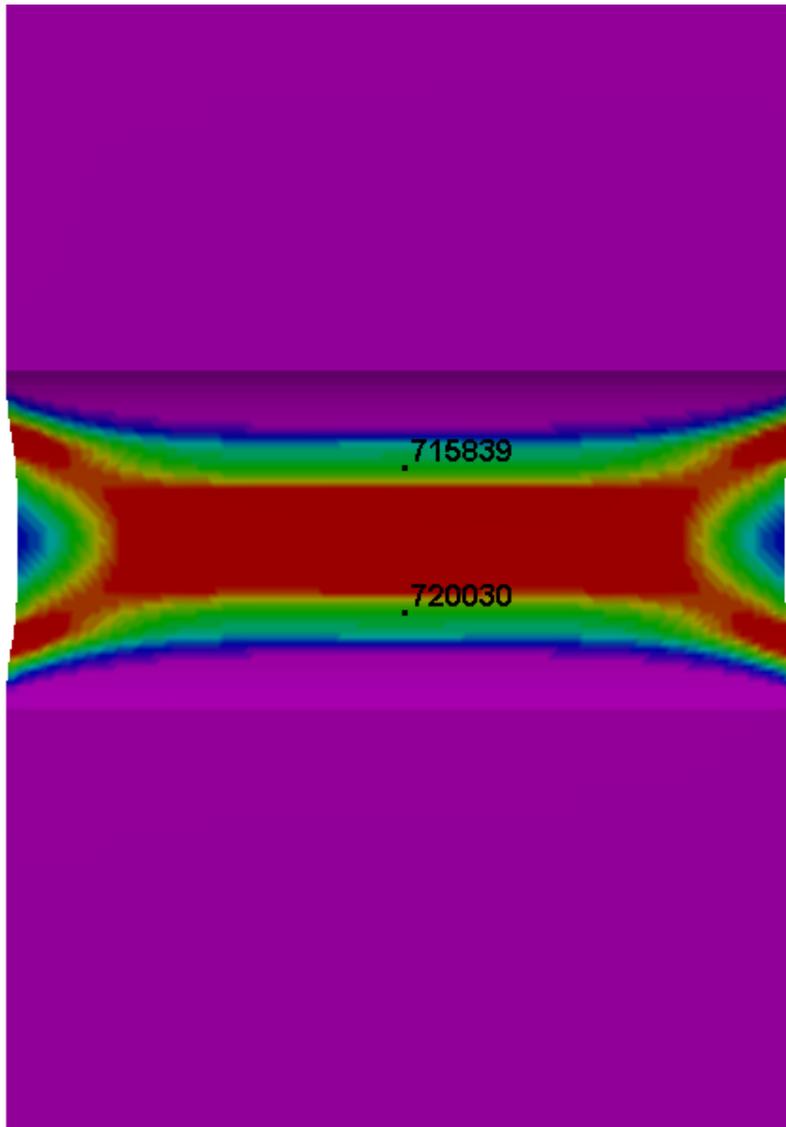
## DIC Comparison – Axisymmetric 6



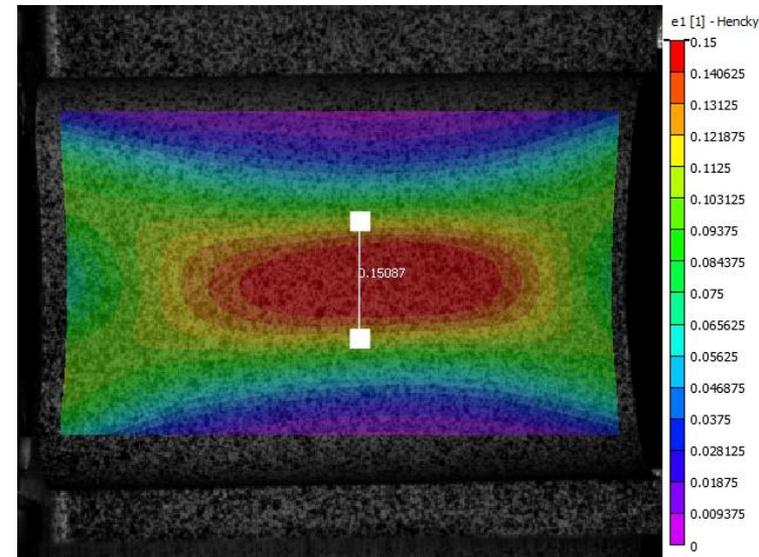
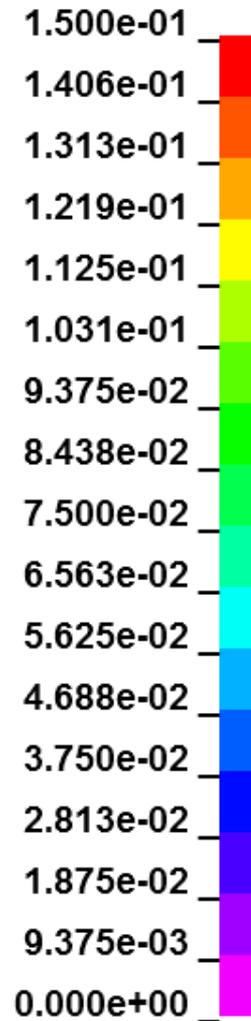
Fringe Levels



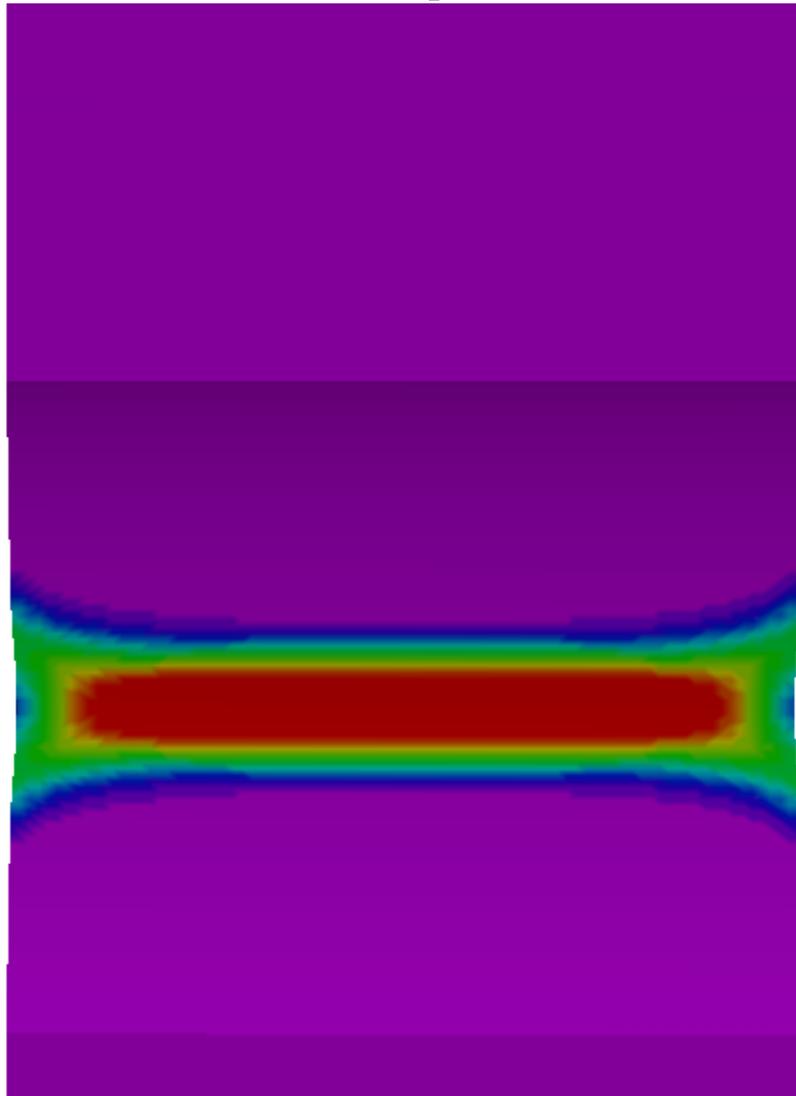
## DIC Comparison – Plane Strain 1



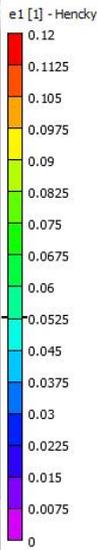
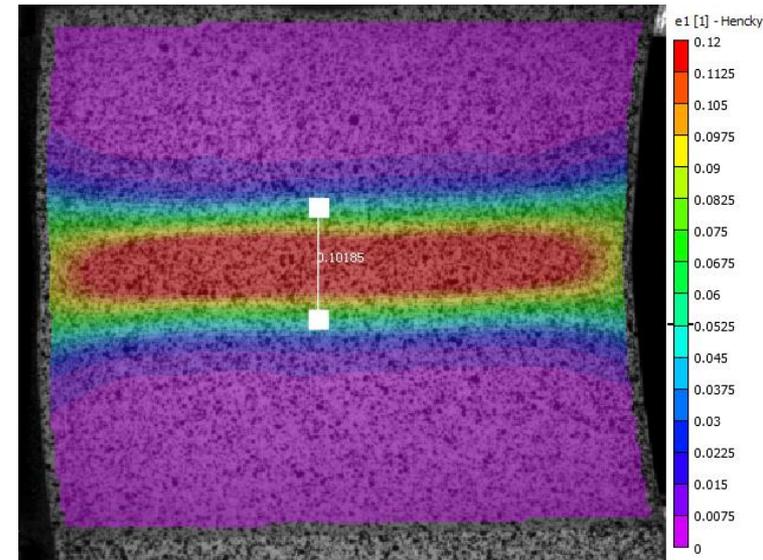
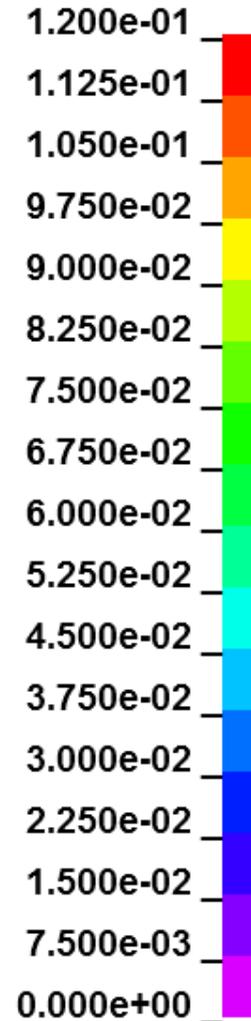
Fringe Levels



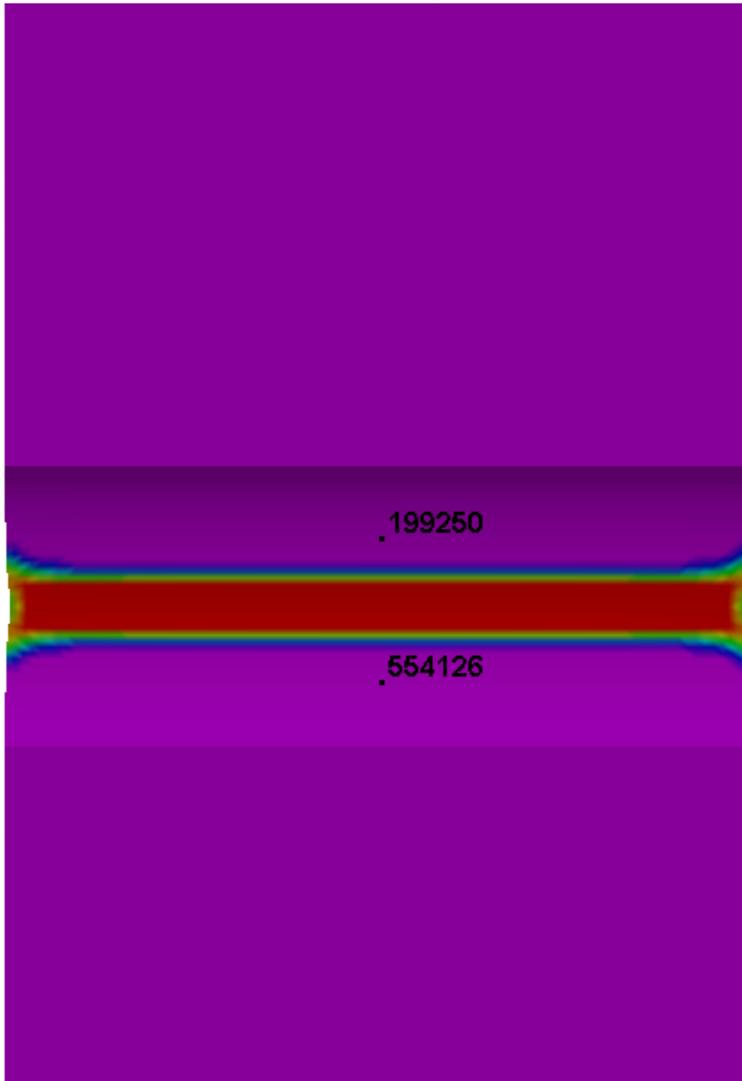
## DIC Comparison – Plane Strain 2



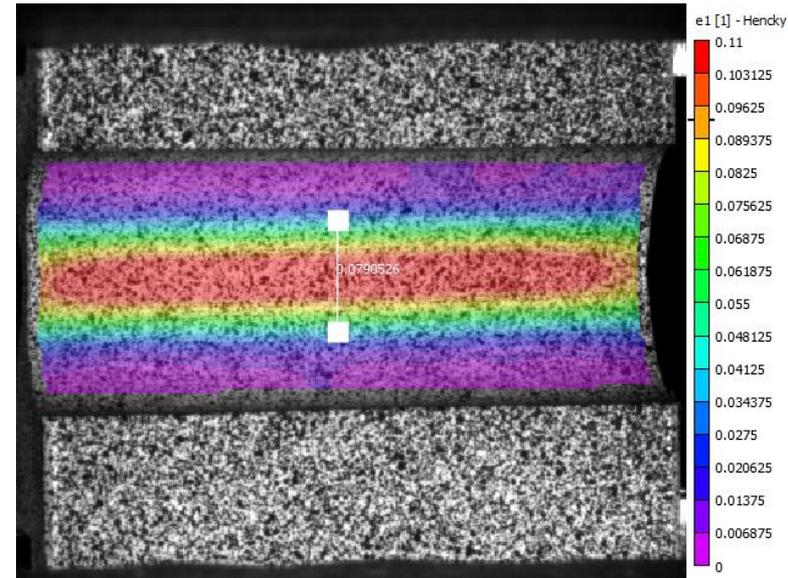
Fringe Levels



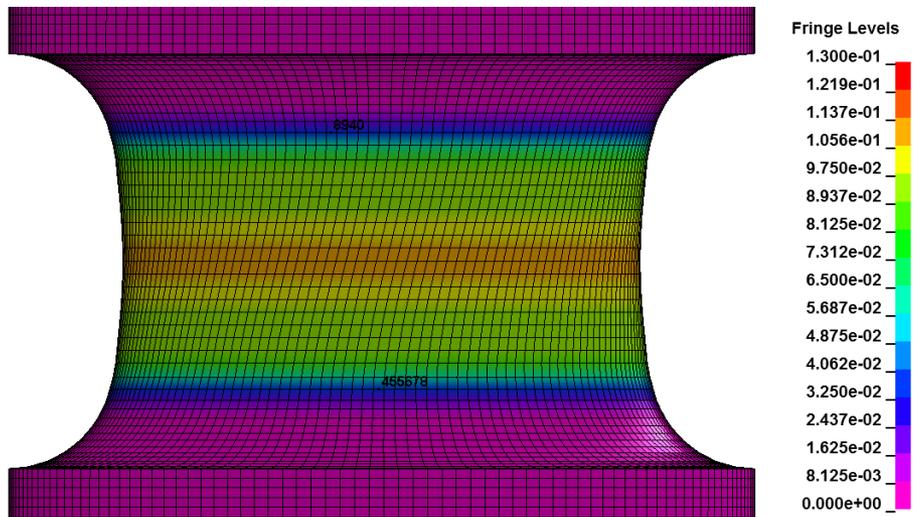
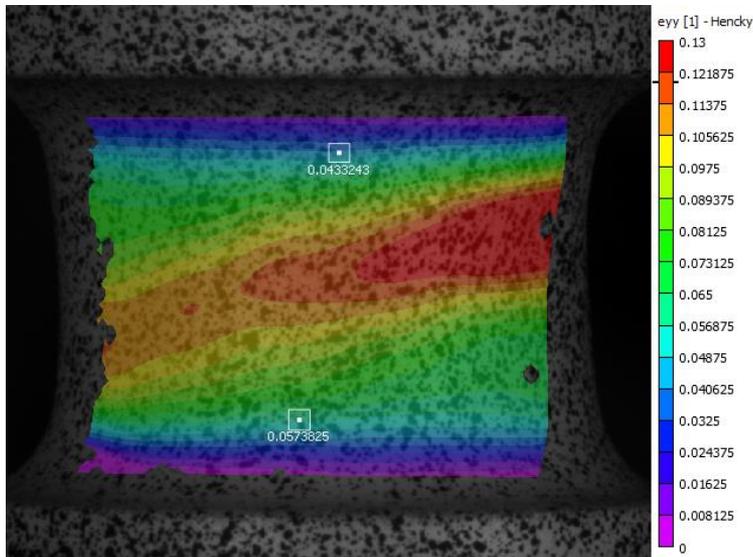
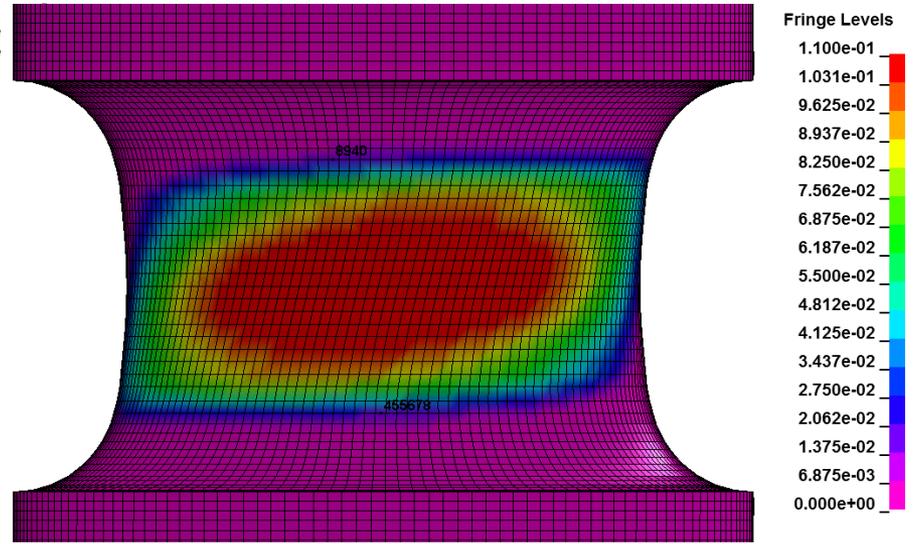
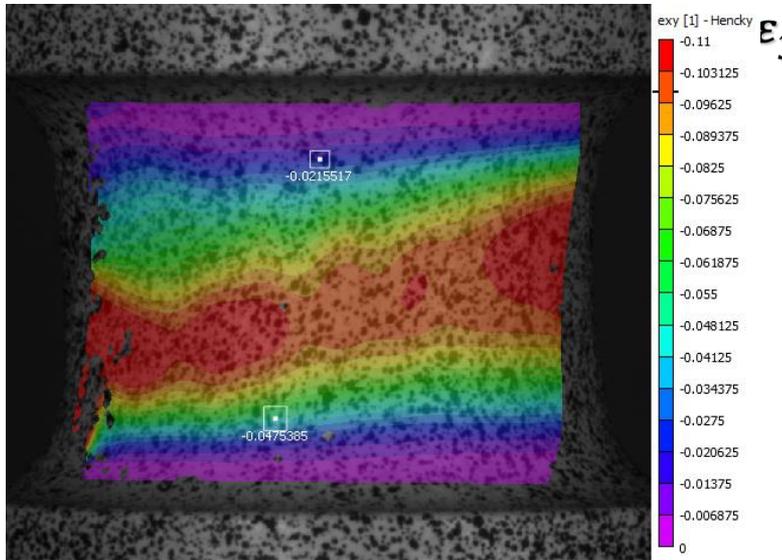
## DIC Comparison – Plane Strain 3



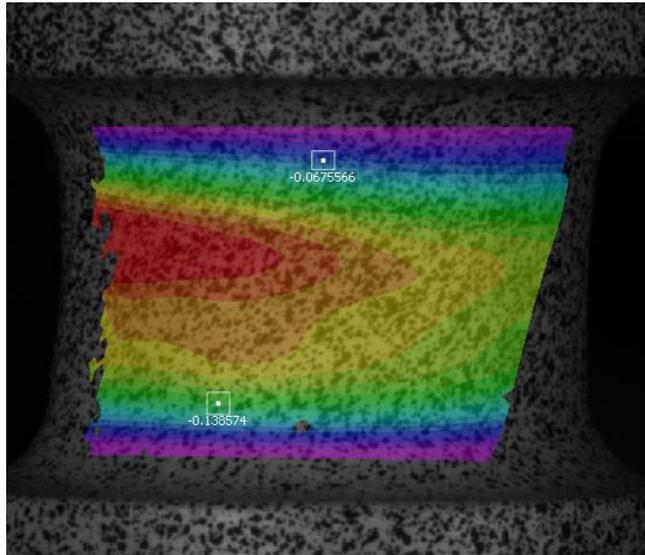
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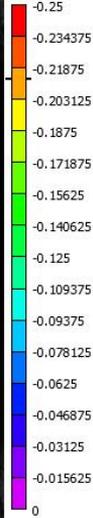
# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



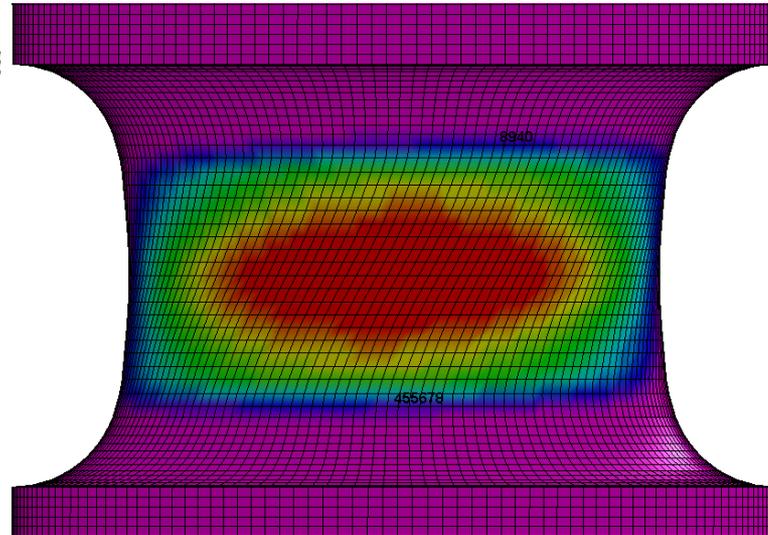
# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



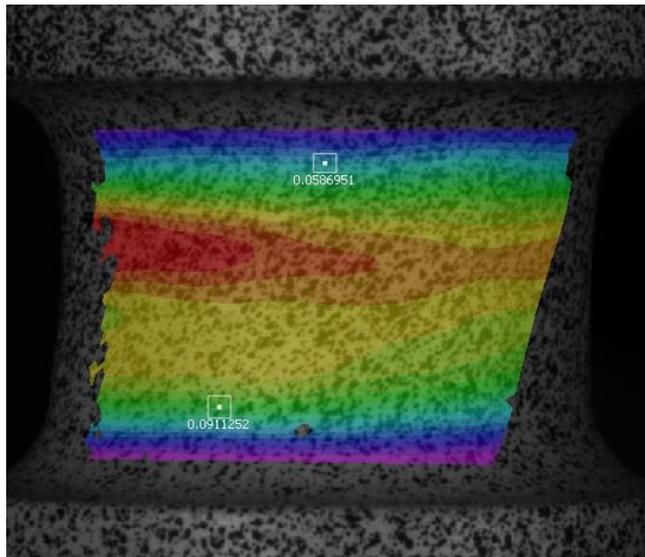
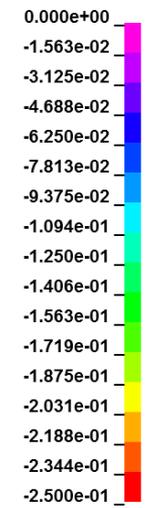
exy [1] - Hencky



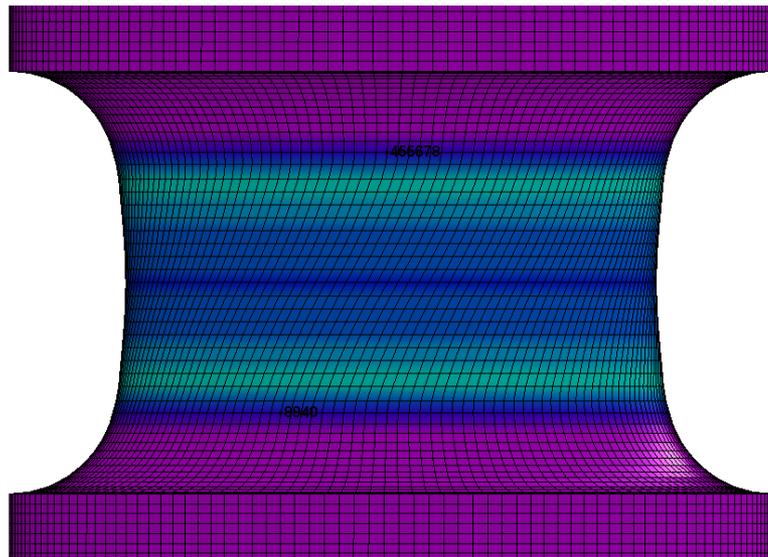
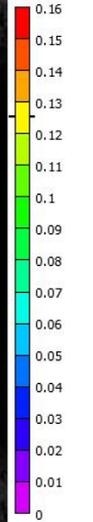
$\epsilon_{xy}, \epsilon$



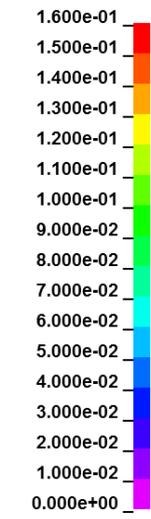
Fringe Levels



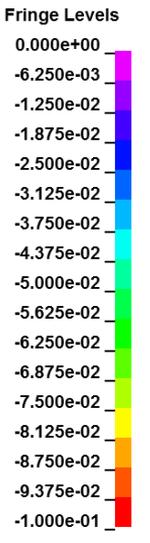
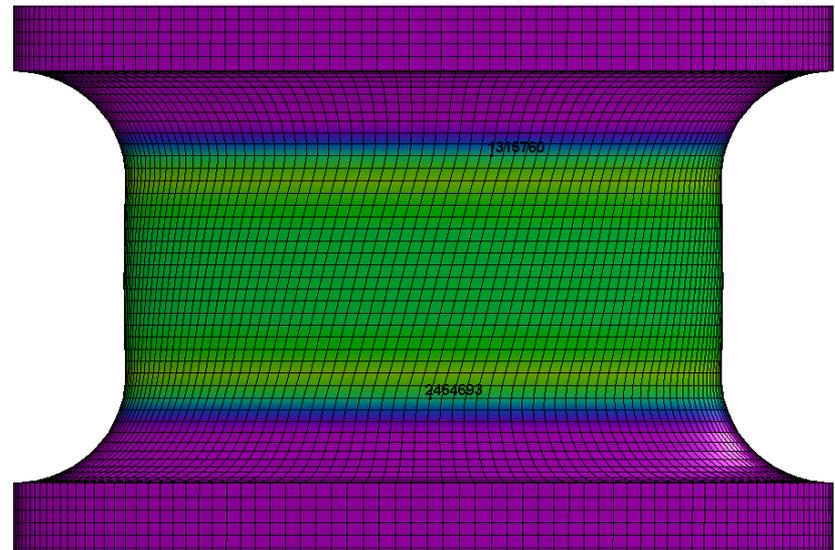
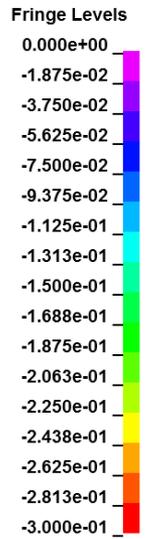
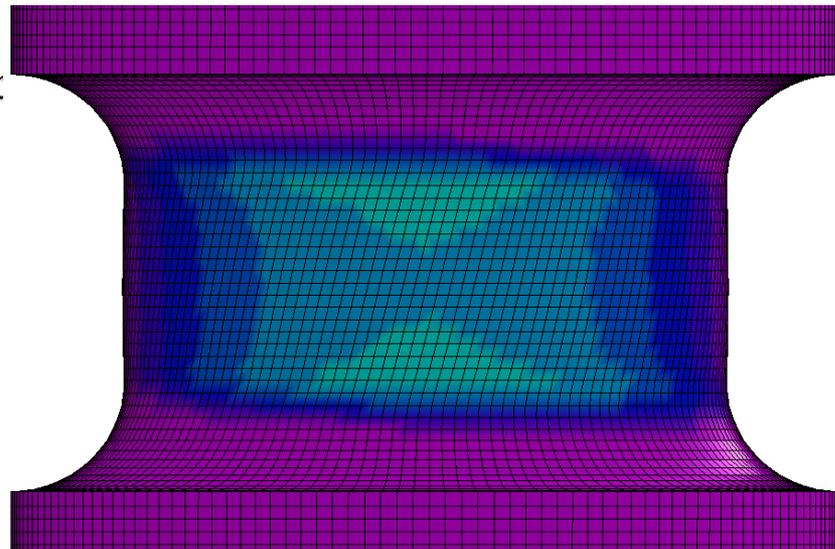
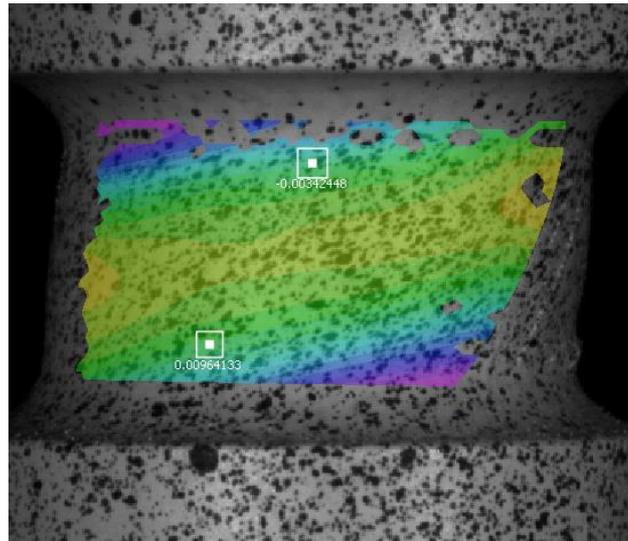
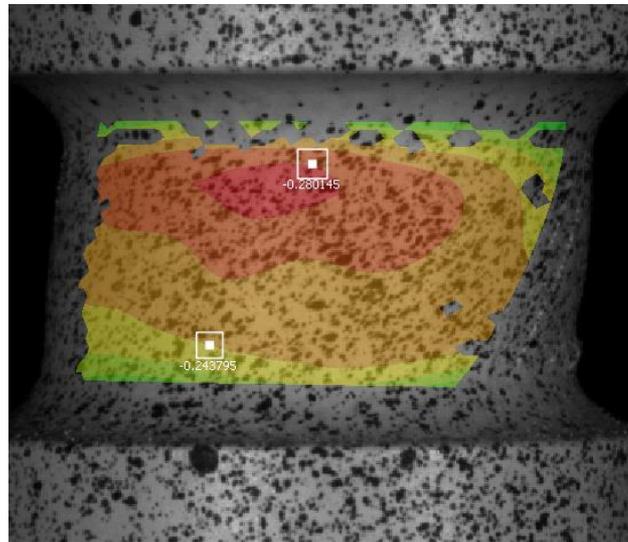
eyy [1] - Hencky



Fringe Levels

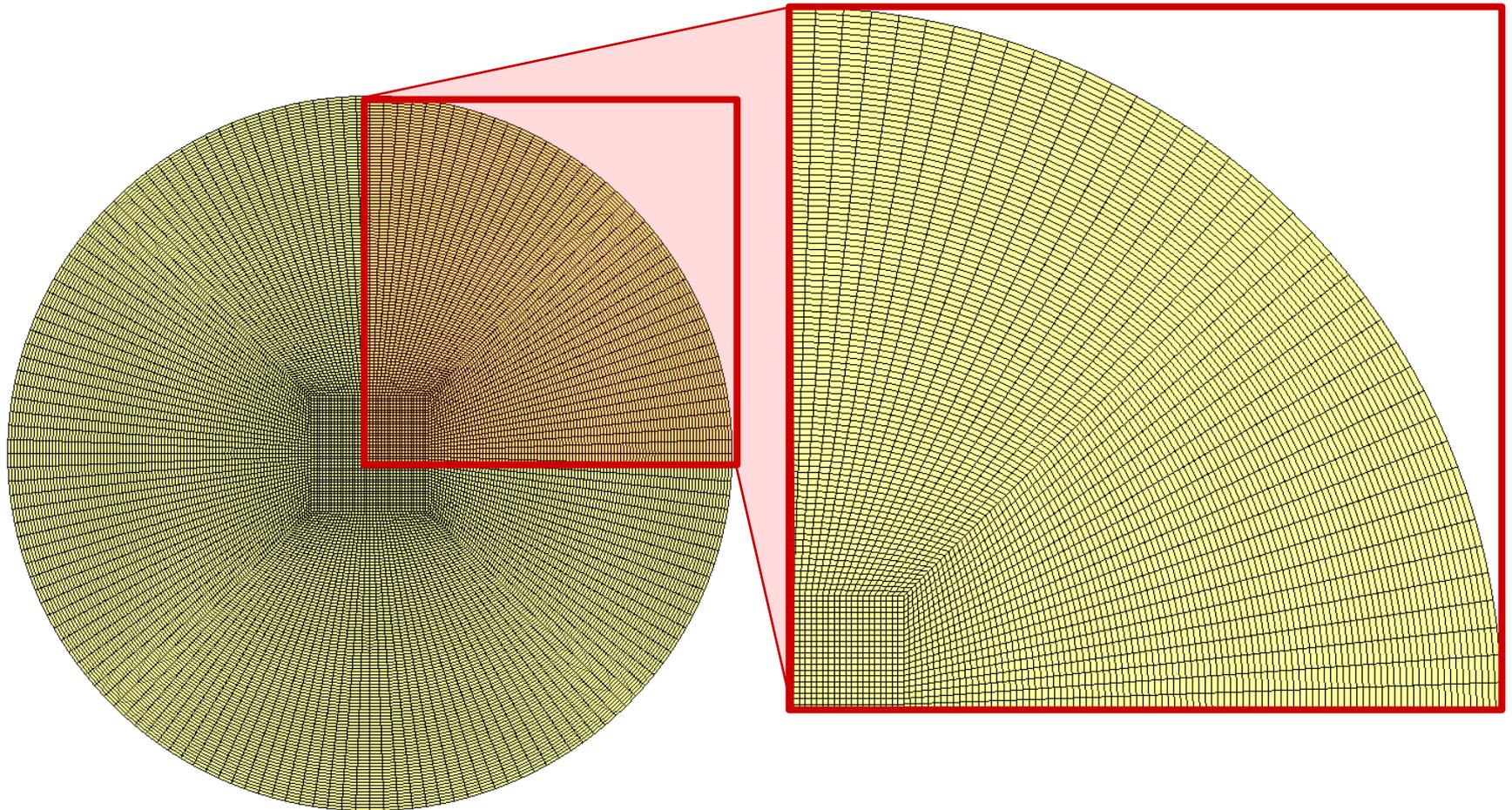


# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

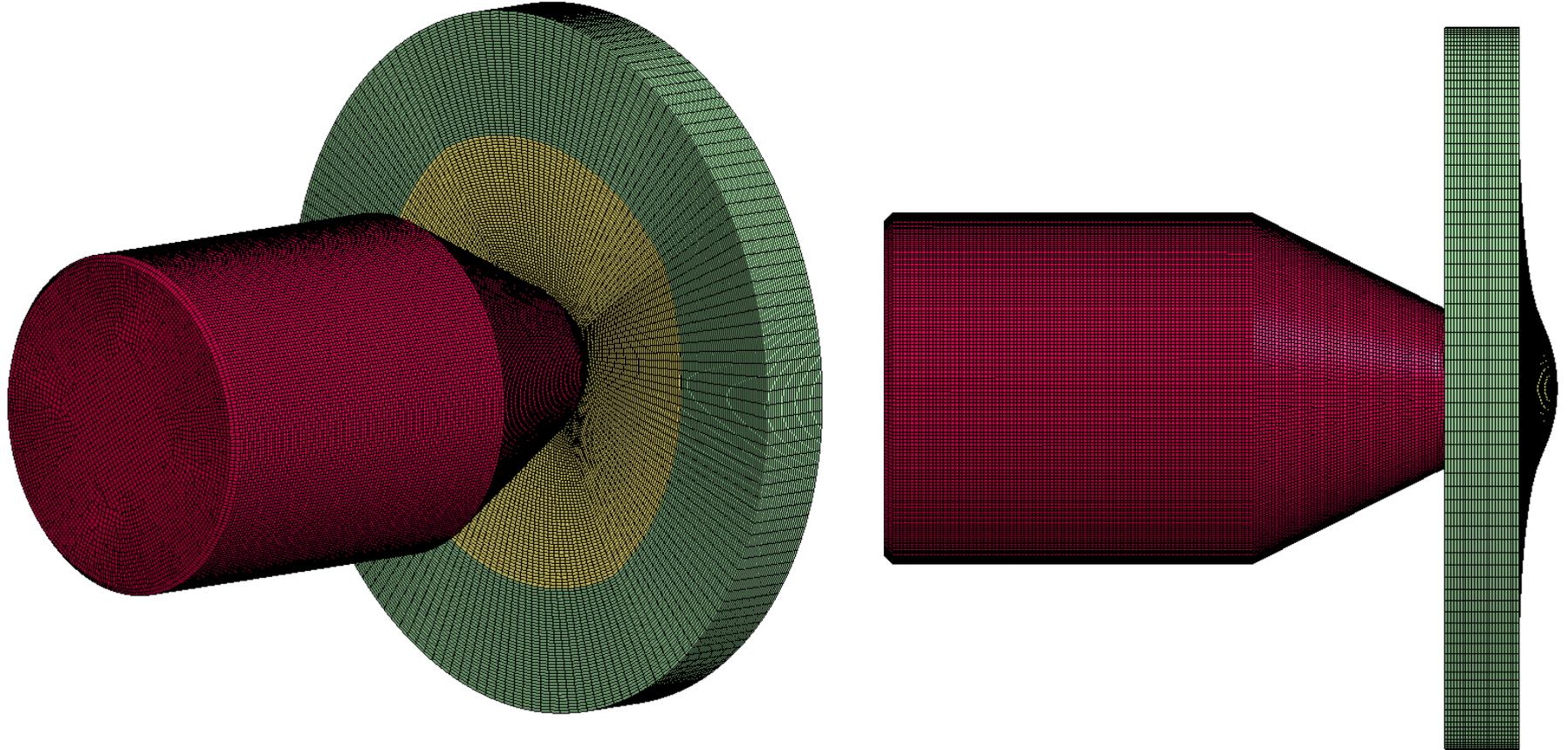


## Large diameter punch tests : Specimen Geometry

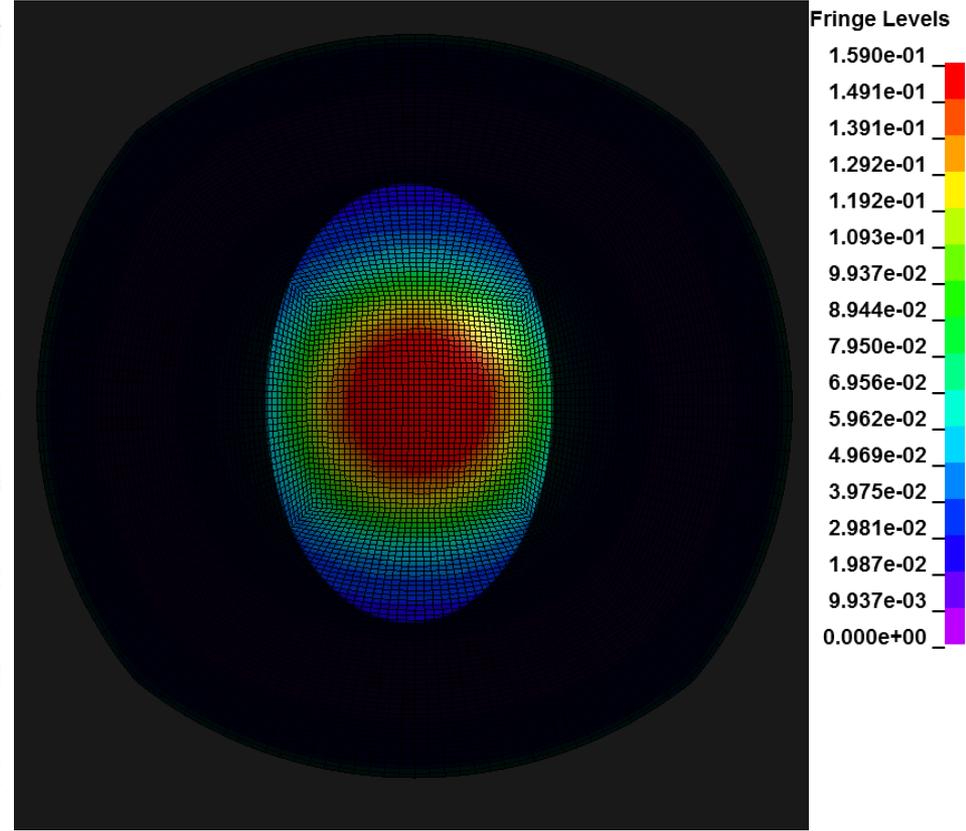
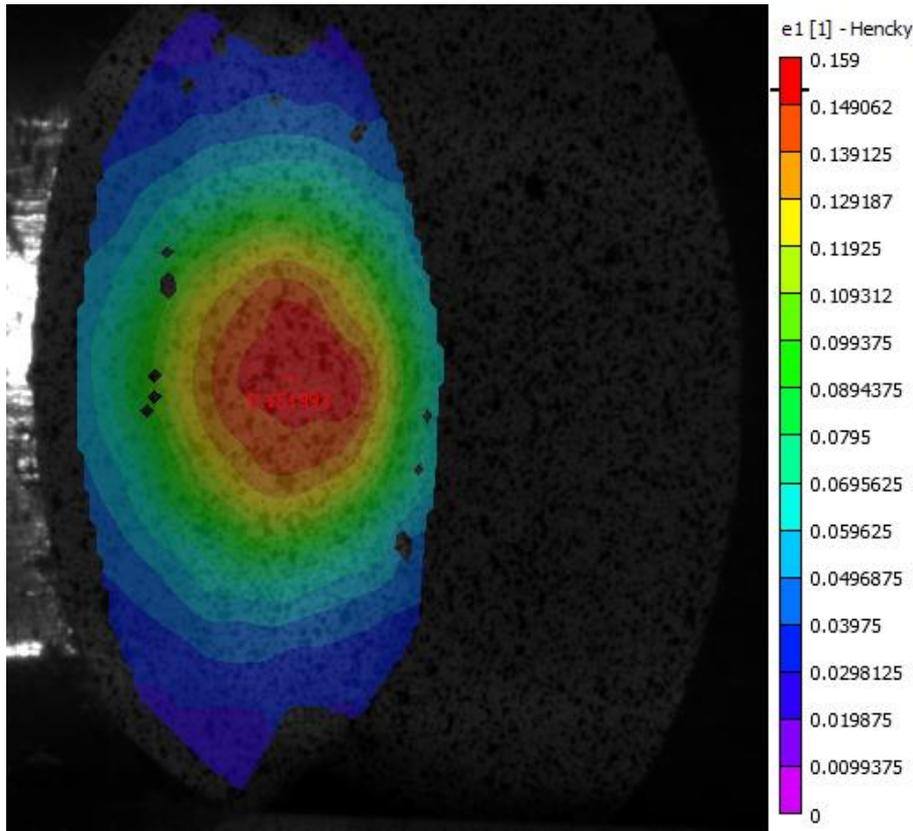
- 420,000 elements (0.2 mm element size)



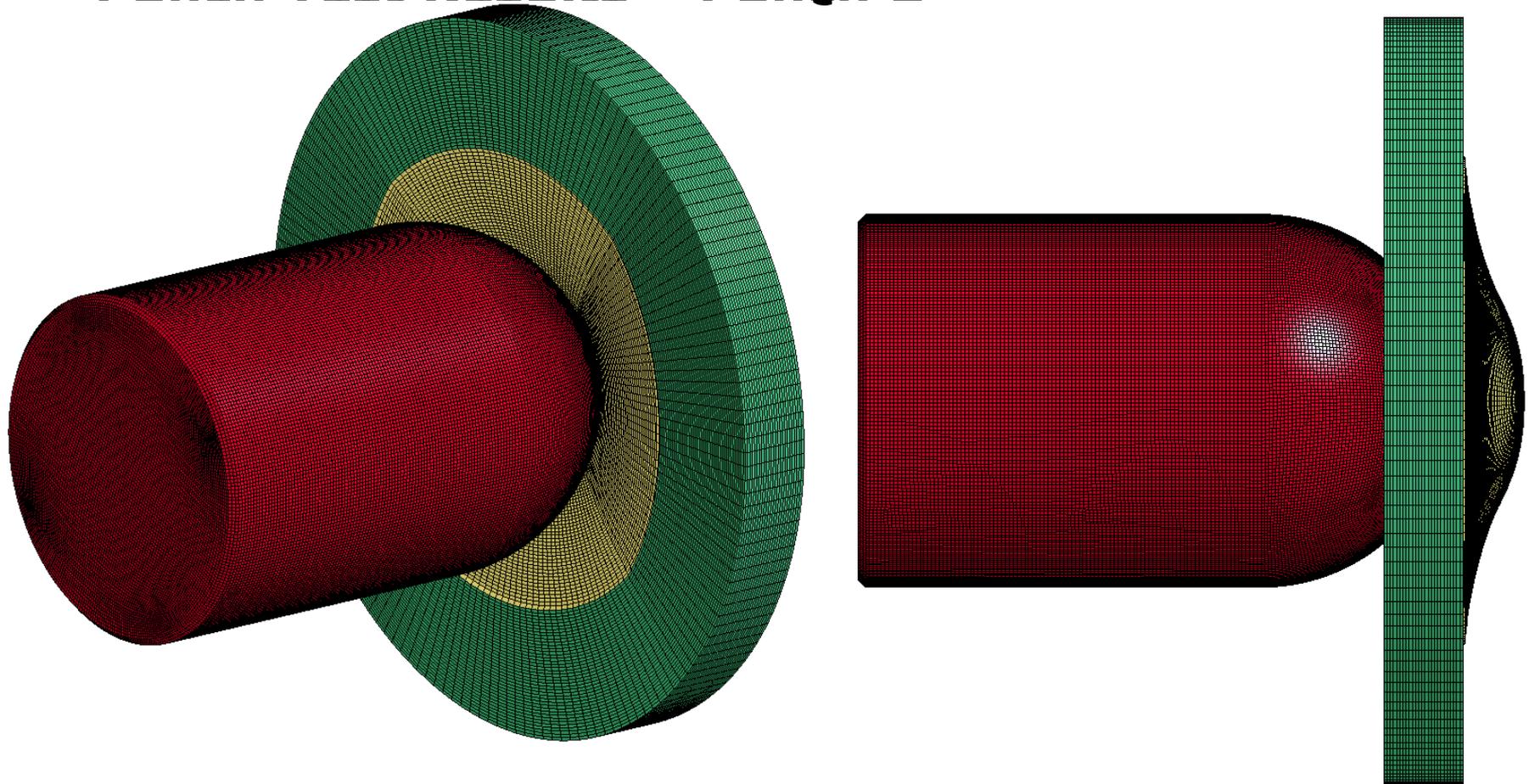
## Punch Test Results – Punch 1



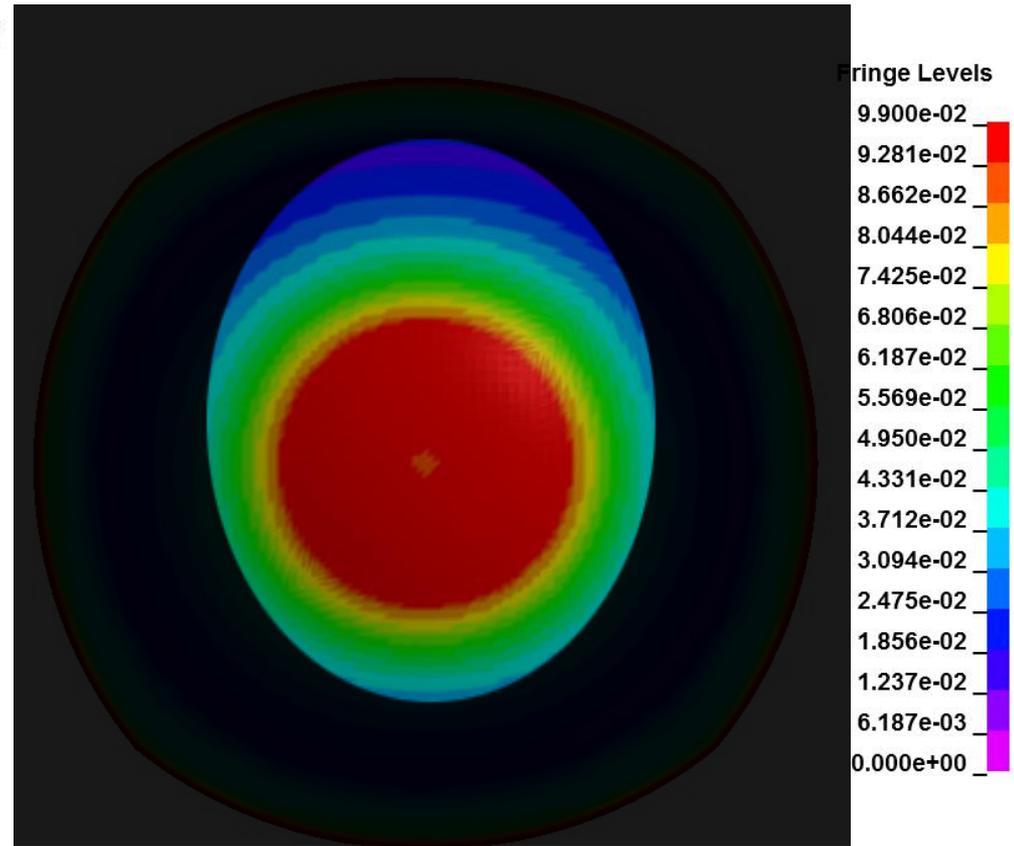
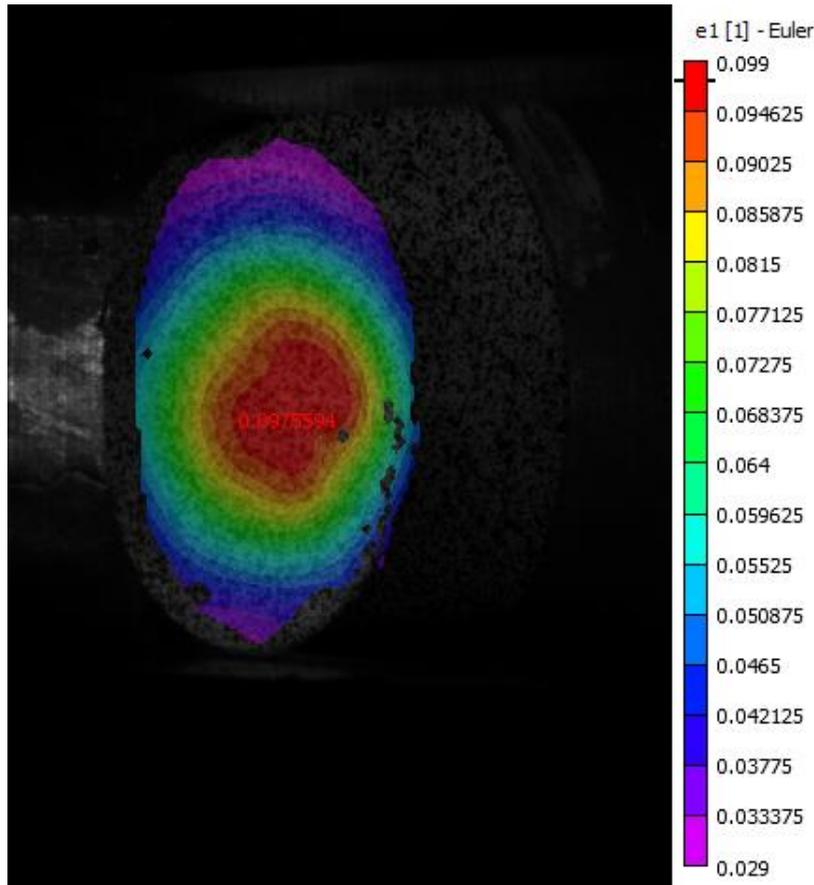
## Punch Test Results – Punch 1



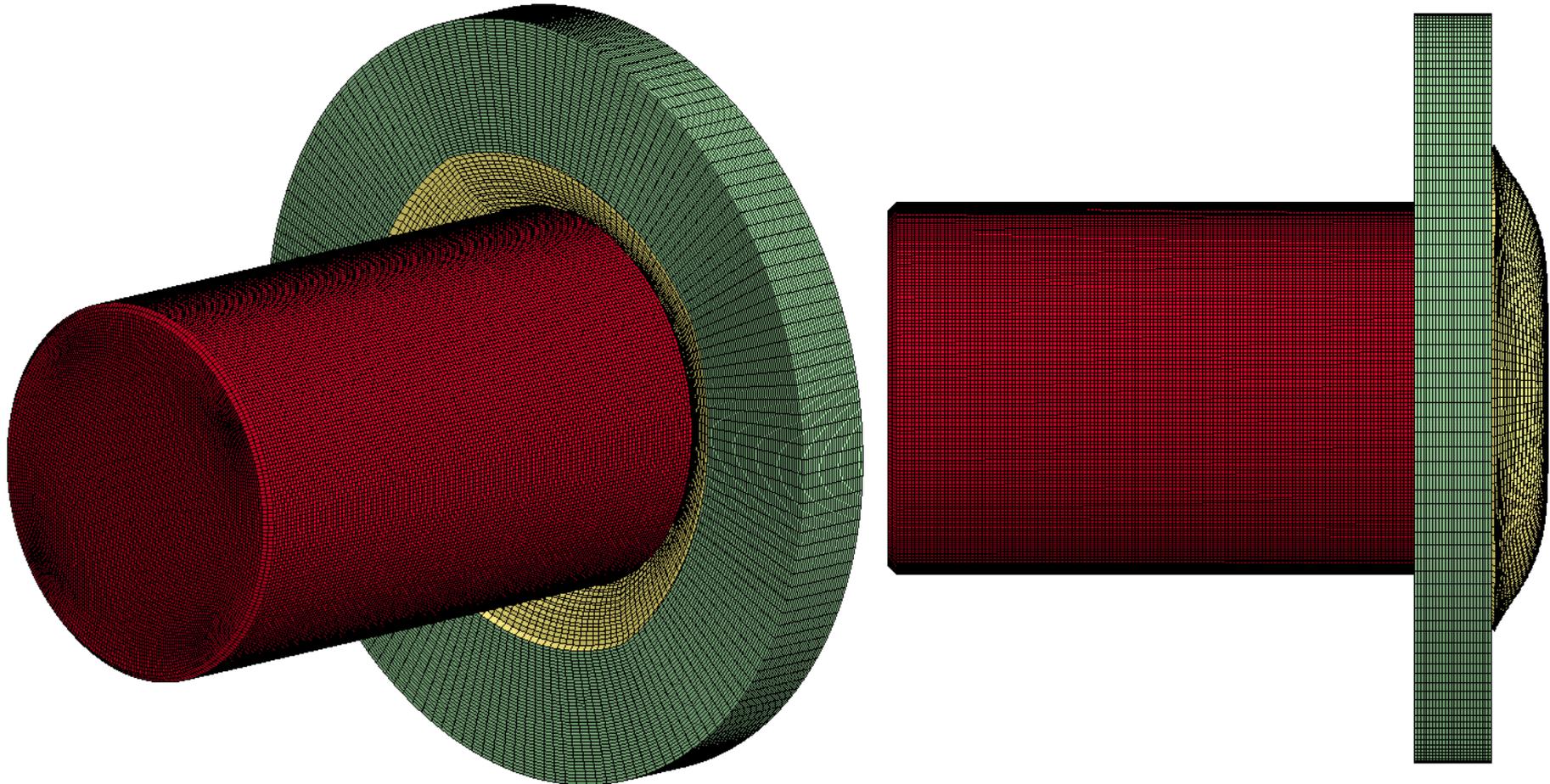
## Punch Test Results – Punch 2



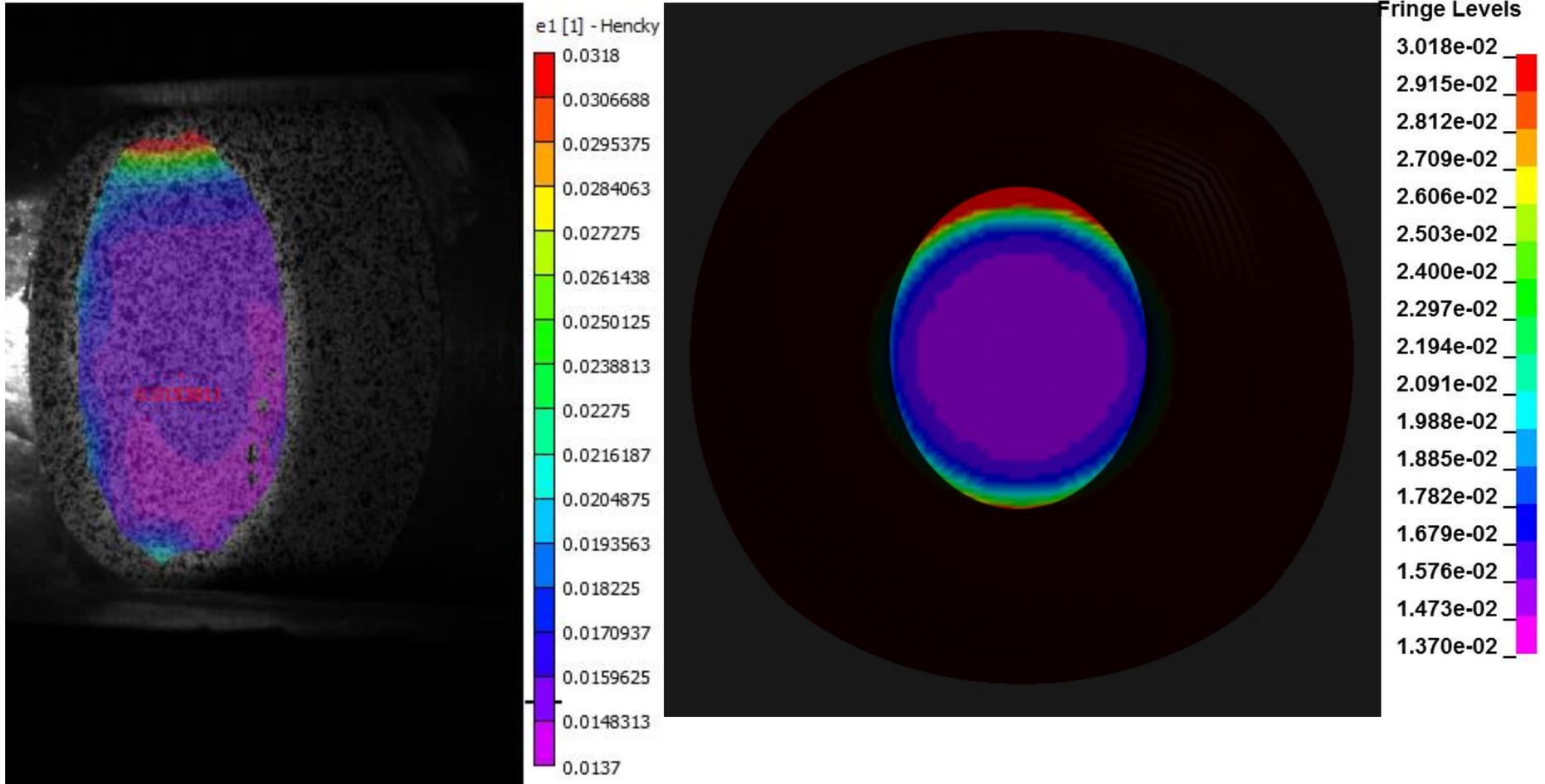
## Punch Test Results – Punch 2



## Punch Test Results – Punch 3



## Punch Test Results – Punch 3



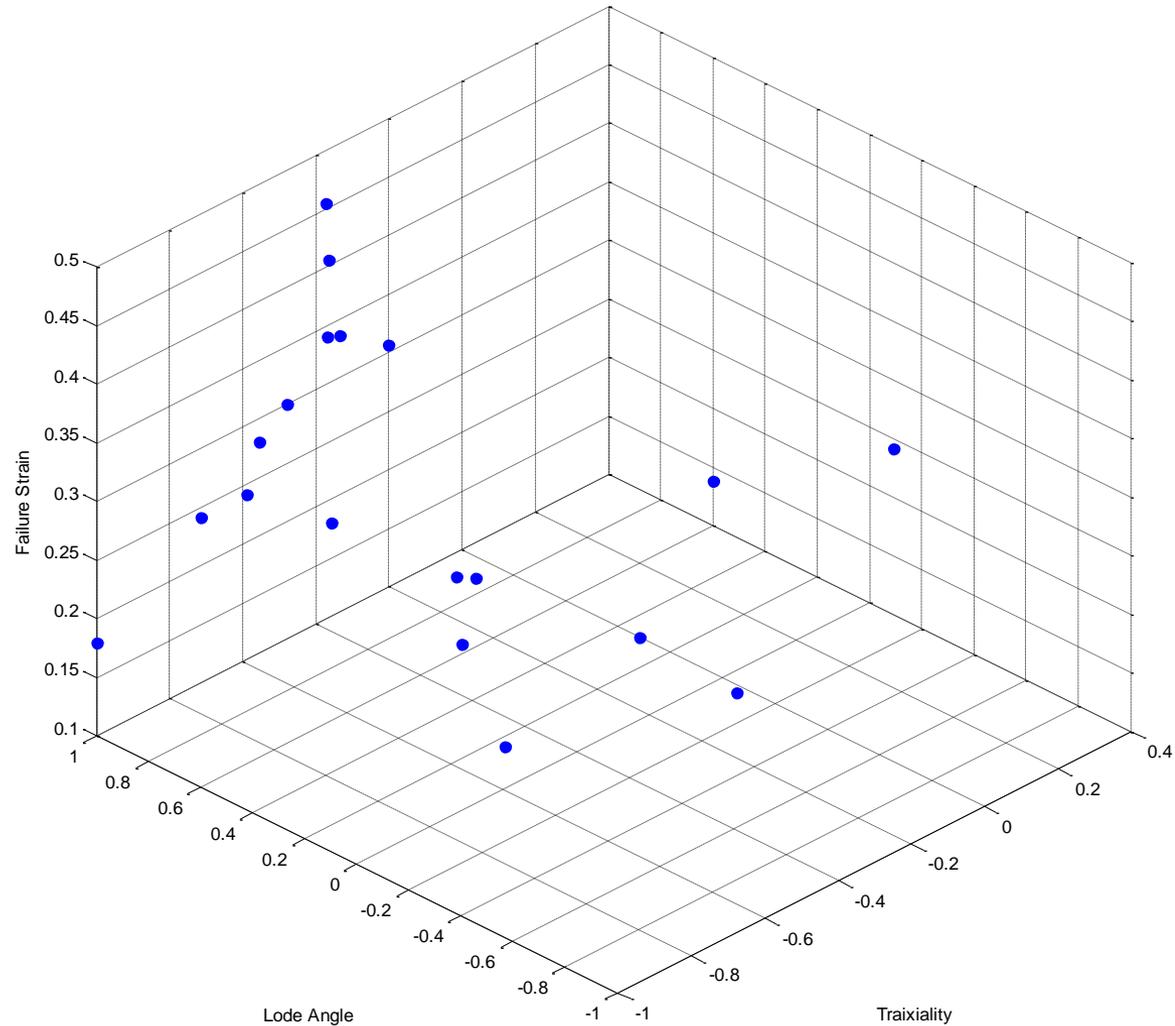
## Completed Failure Table

Test Number	Geometry	OSU			GMU		
		Triaxiality	Lode Parameter	Failure Strain	Triaxiality	Lode Parameter	Failure Strain
SG1	Plane Stress	0.400	0.850	0.590	0.390	0.975	0.460
SG2		0.431	0.719	0.440	0.412	0.935	0.420
SG3		0.489	0.528	0.430	0.475	0.803	0.380
SG4		0.583	0.014	0.140	0.592	0.005	0.135
SG5	Axisymmetric	0.369	1.000	0.310	0.370	1.000	0.340
SG6		0.492	1.000	0.310	0.480	1.000	0.300
SG7		0.564	1.000	0.320	0.553	1.000	0.280
SG8		0.618	1.000	0.270	0.588	1.000	0.240
SG9		0.751	1.000	0.270	0.712	1.000	0.240
SG10		0.956	1.000	0.220	1.000	1.000	0.180
SG11	Plane Strain	0.470	0.506	0.210	0.573	0.146	0.260
SG12		0.660	0.040	0.220	0.643	0.099	0.220
SG13		0.768	0.025	0.210	0.691	0.054	0.290
LR1	Combined Loading	0.252	0.706	0.290	0.289	0.949	0.191
LR2		0.150	0.400	0.510	0.145	0.917	0.325
LR3		0.000	0.000	0.430	0.014	0.059	0.259
LR4		-0.148	-0.394	0.420	-0.147	-0.450	0.321
Punch1	Punch Tests				0.653	-0.969	0.297
Punch2					0.565	-0.475	0.276
Punch3					-0.077	-0.073	0.766

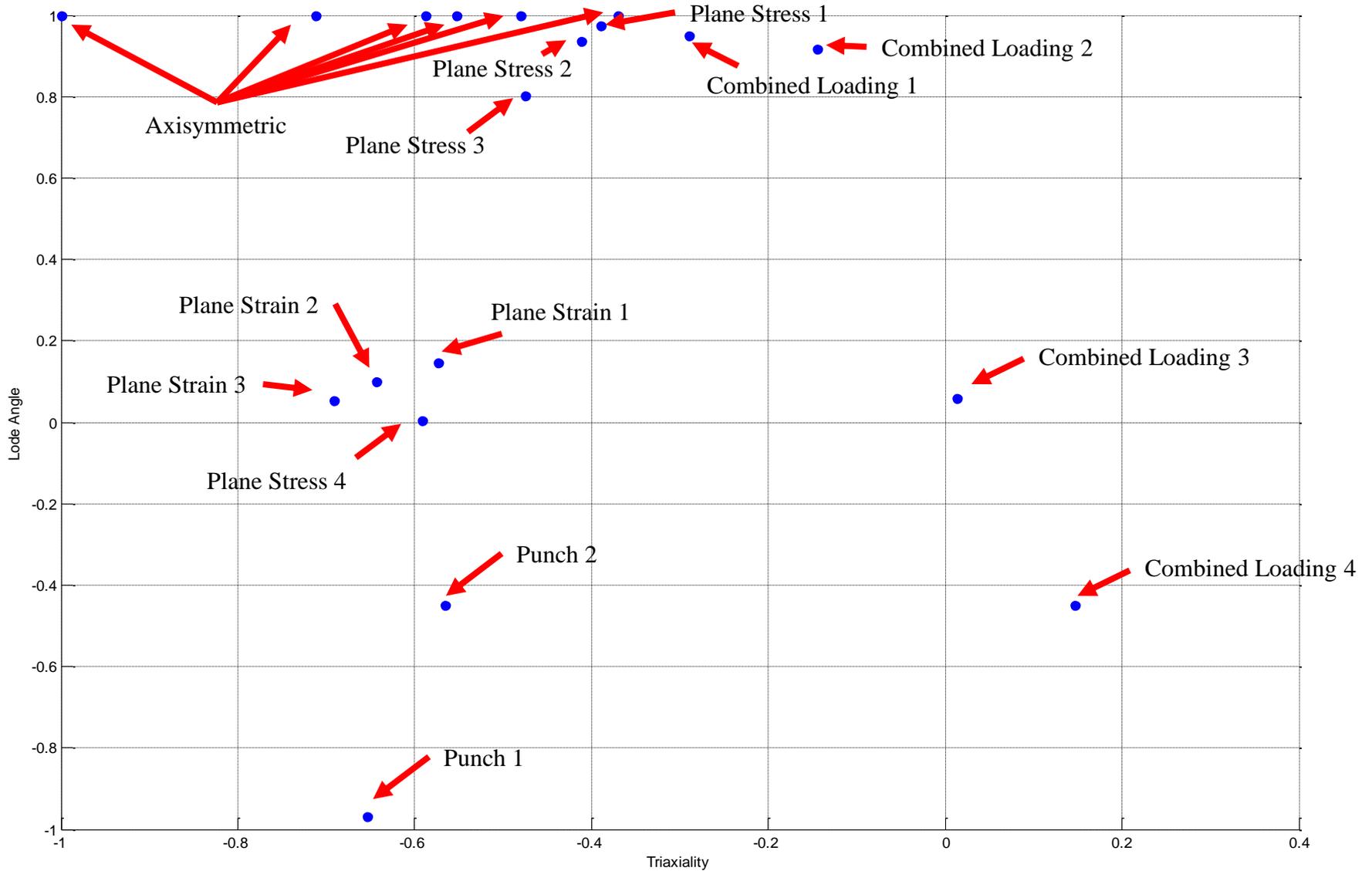
# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

## Data points

Test Number	GMU		
	Triaxiality	Lode Parameter	Failure Strain
SG1	0.390	0.975	0.460
SG2	0.412	0.935	0.420
SG3	0.475	0.803	0.380
SG4	0.592	0.005	0.135
SG5	0.370	1.000	0.340
SG6	0.480	1.000	0.300
SG7	0.553	1.000	0.280
SG8	0.588	1.000	0.240
SG9	0.712	1.000	0.240
SG10	1.000	1.000	0.180
SG11	0.573	0.146	0.260
SG12	0.643	0.099	0.220
SG13	0.691	0.054	0.290
LR1	0.289	0.949	0.191
LR2	0.145	0.917	0.325
LR3	0.014	0.059	0.259
LR4	-0.147	-0.450	0.321
Punch1	0.653	-0.969	0.297
Punch2	0.565	-0.475	0.276
Punch3	-0.077	-0.073	0.766

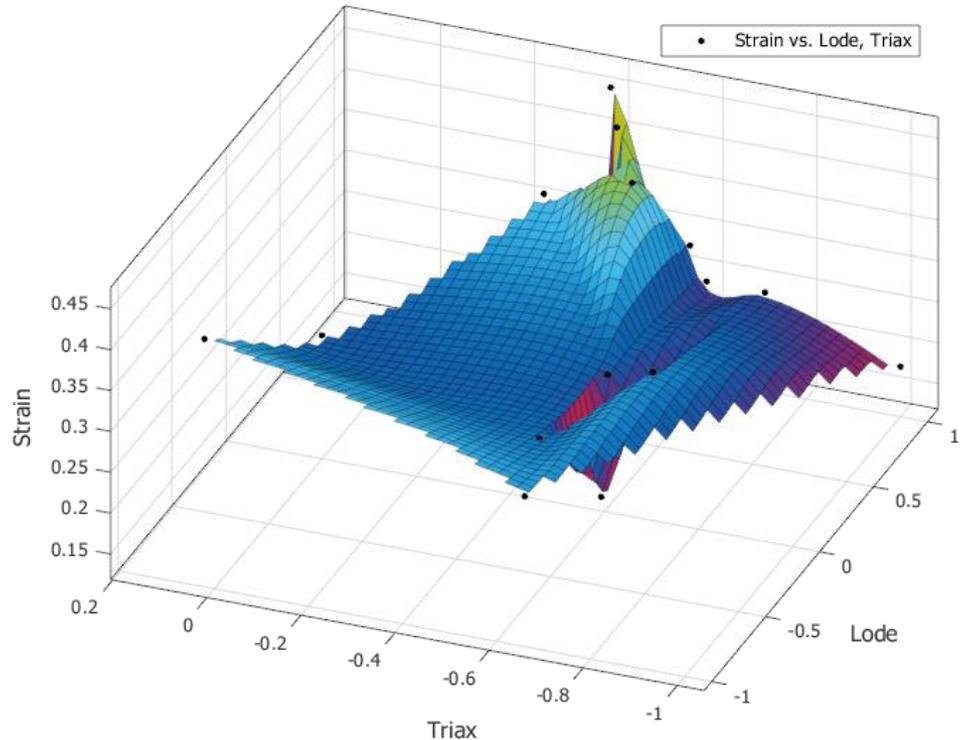


# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

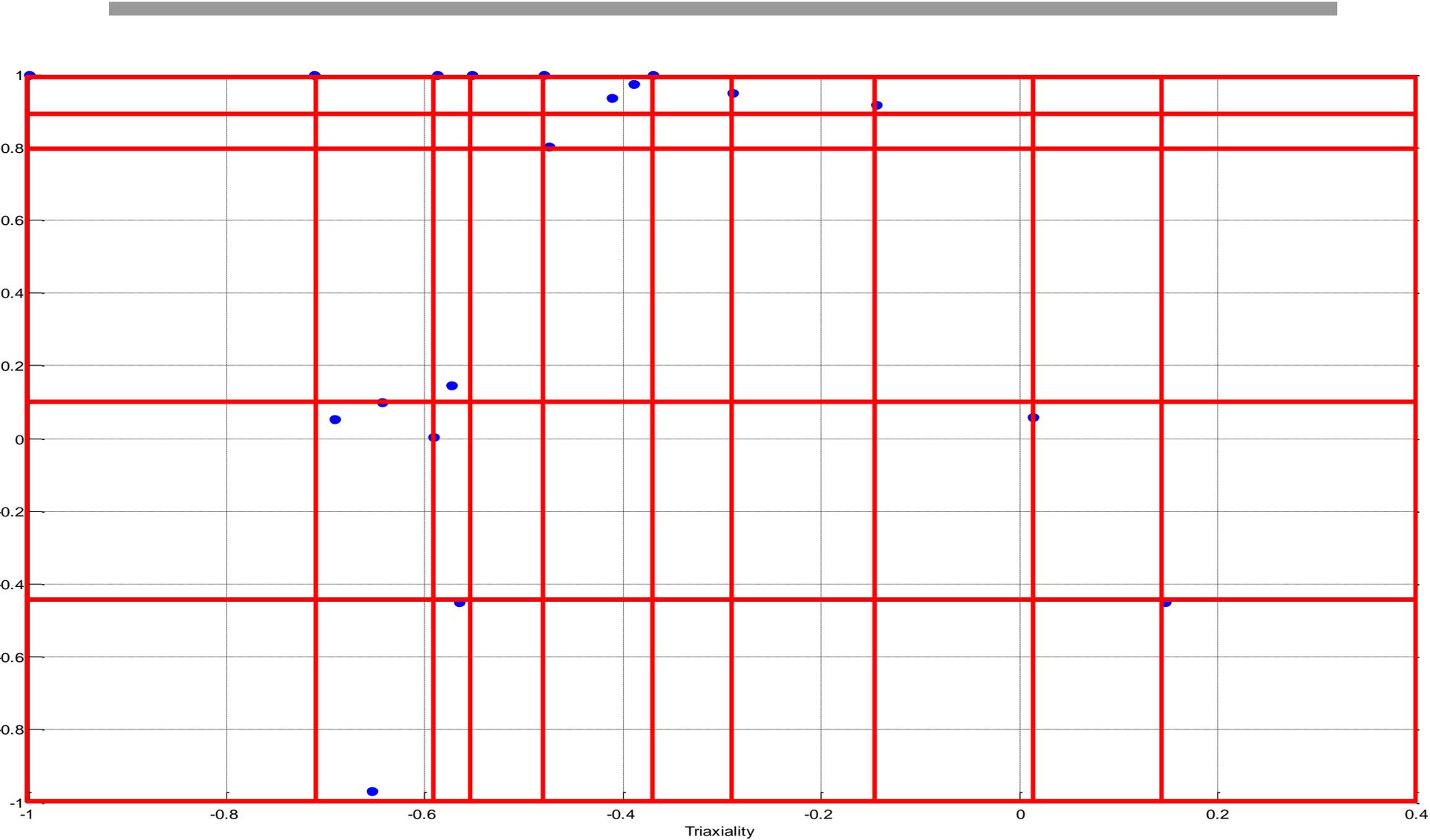


## Failure Surface Plan

- Automatic surface interpolation methods do not work as well for data sets with large voids and clusters
- Also very difficult because the boundaries are not defined with discrete points
- New plan:
  - Cluster
  - Grid
  - Interpolate

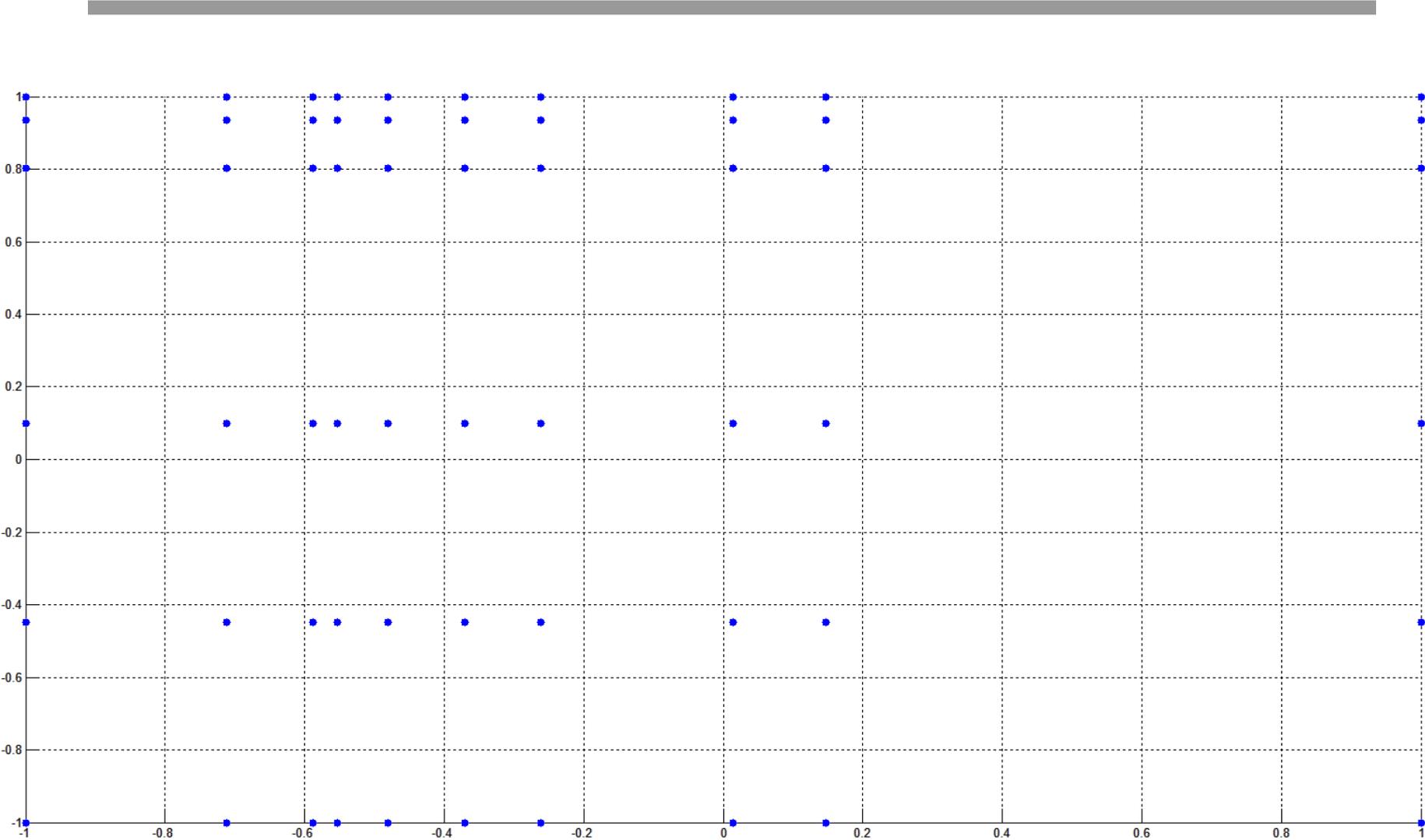


# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



**Cluster and Grid Plan**

# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

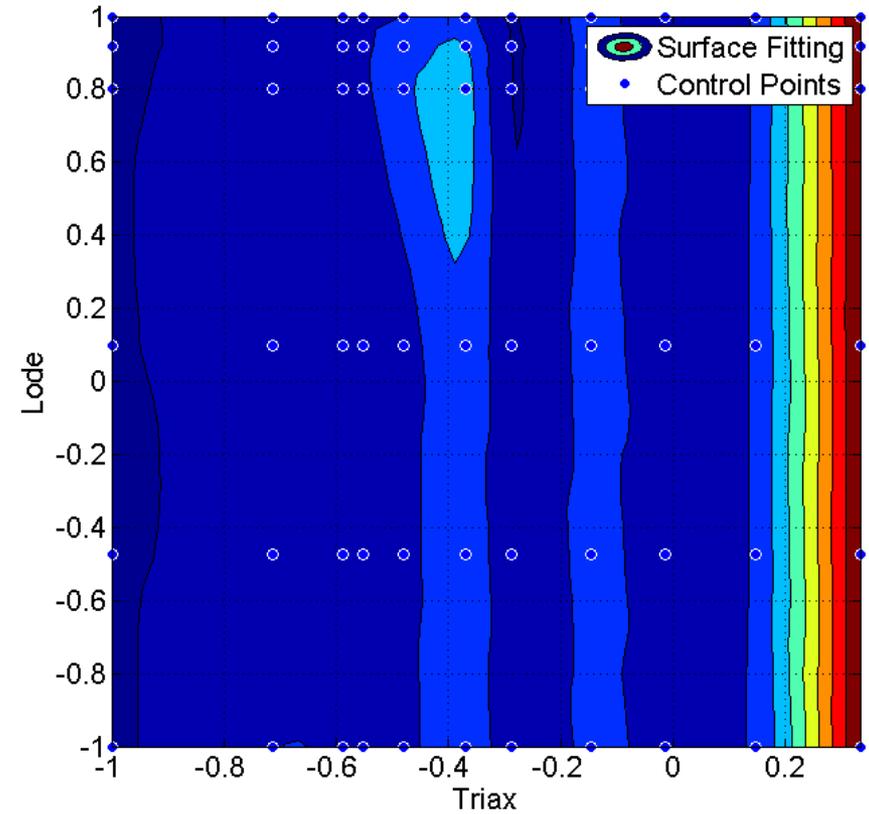
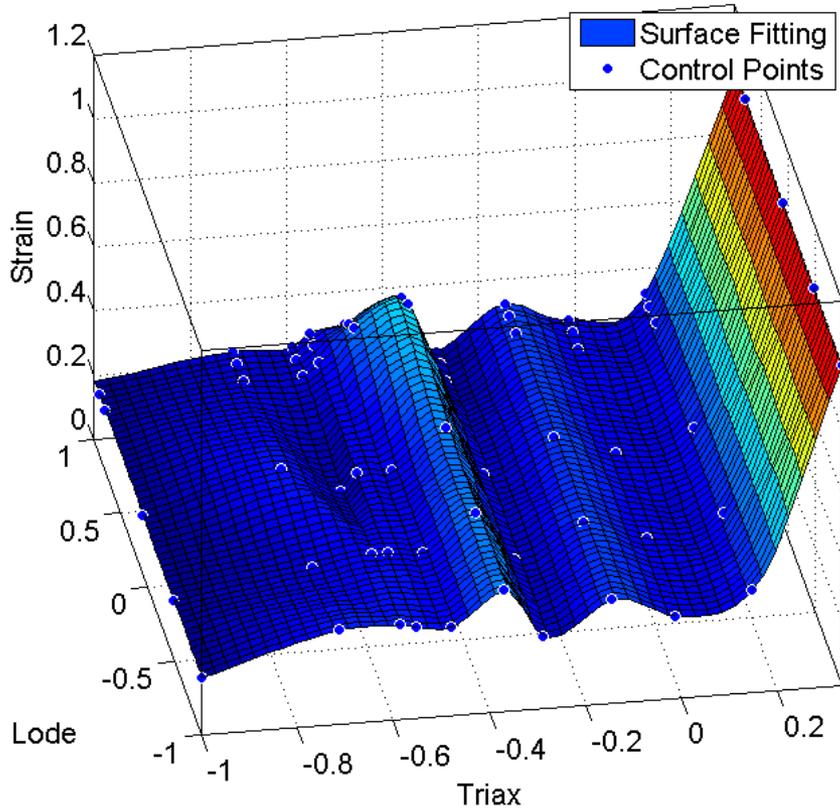


**Grid Points**

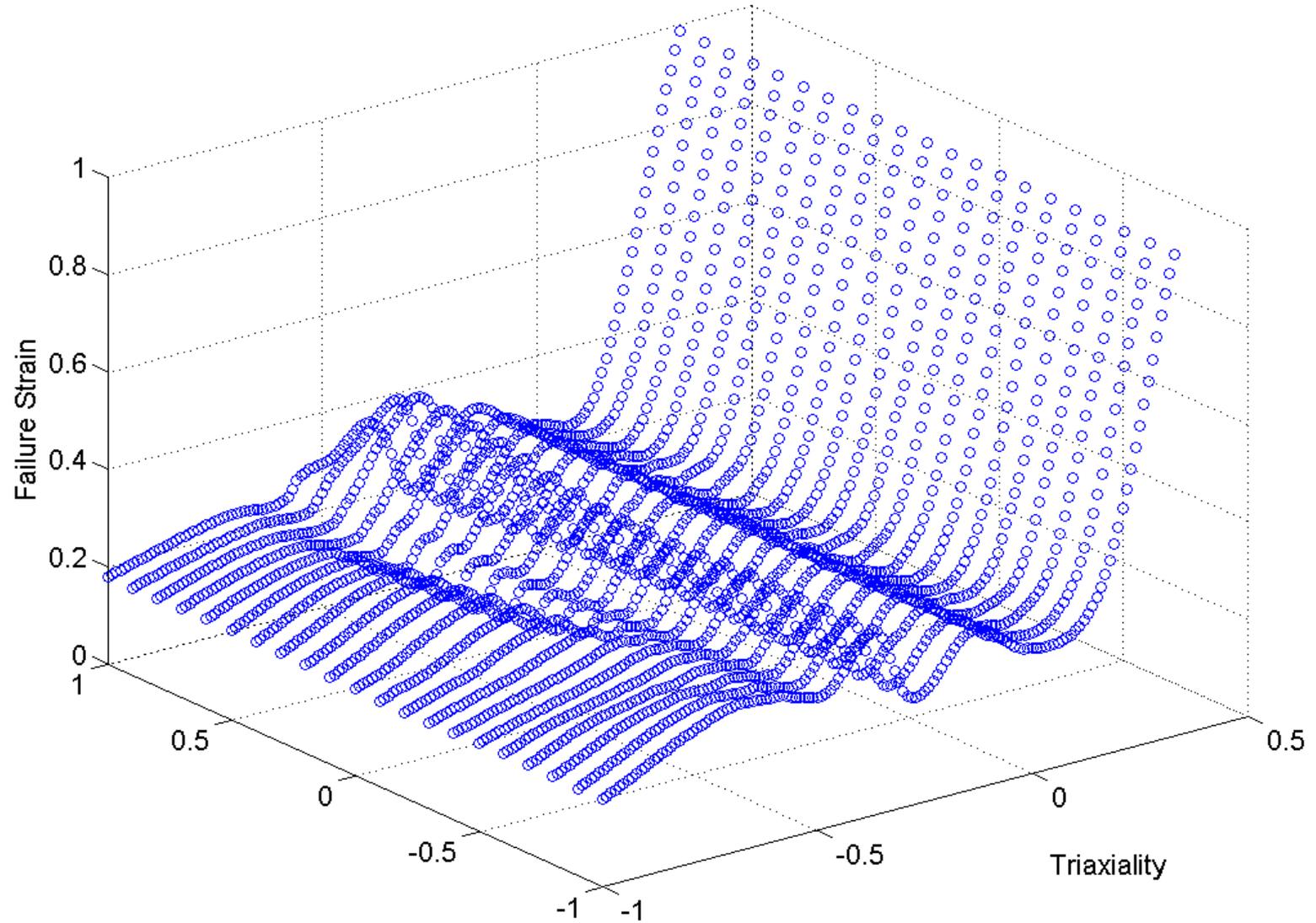
## Failure Surface Fitting

- **MATLAB code was written to implement “control point matrix strategy”**
- **Code implementation:**
  - 1. Initializing matrices for Triaxiality, Lode angle, and failure strain**
  - 2. Populating Triaxiality and Lode angle matrices with grid points**
  - 3. Populating failure strain by selecting nearest test data point**
    - This sometimes included averaging two or three nearest test data points
  - 4. Implement Curve Fitting Toolbox generated code for fitting a surface**
    - From Triaxiality of -1 to 0.33
    - From Lode angle of -1 to 1
  - 5. Create evaluation points using generated fit and store into matrix**
    - Triaxiality: [-1:.01:.33] (every 0.1 units / 134 total points)
    - Lode angle=[-1:.1:1] (every 0.1 units / 21 points)
    - 2814 total points (21 ls-dyna curves)
  - 6. Generate keyword file automatically using evaluation matrices in previous step**

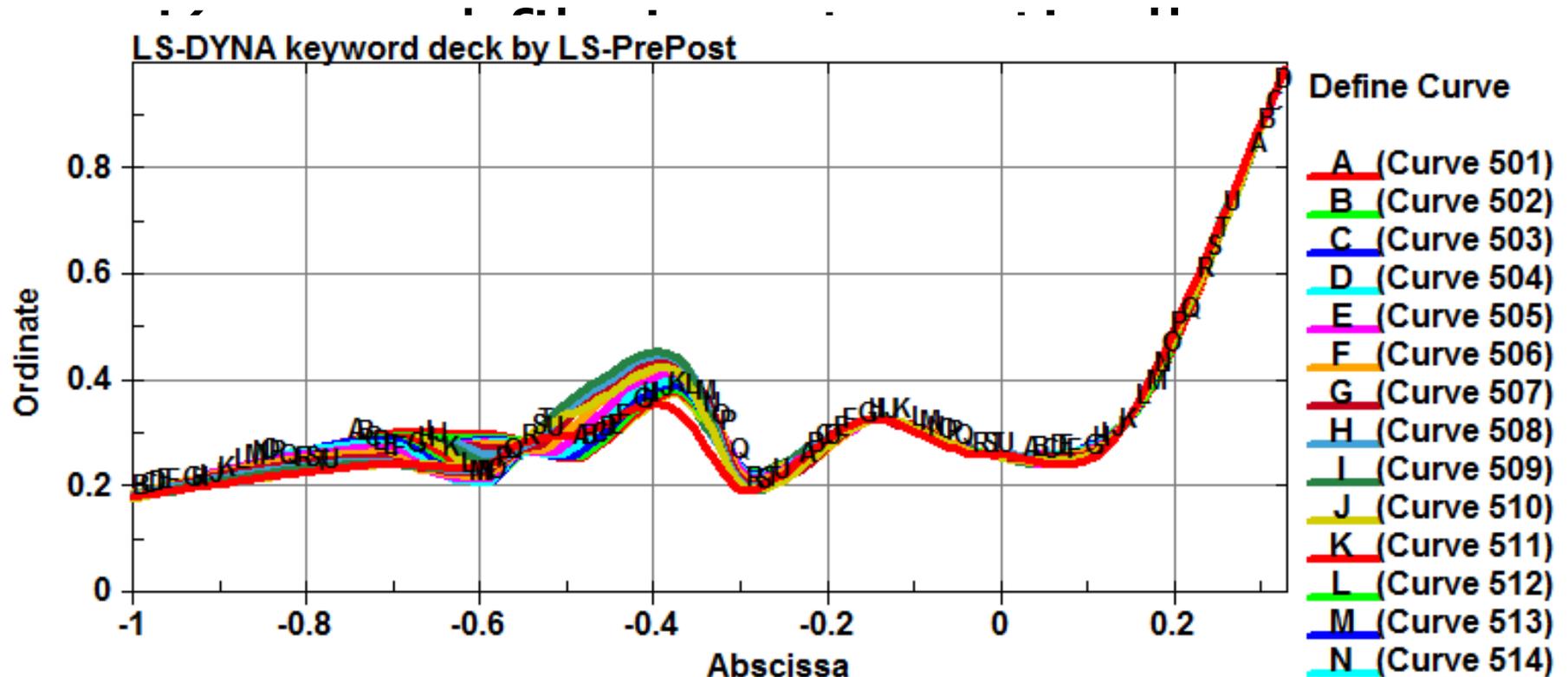
## Surface Fitting Result



## Evaluation Points Result

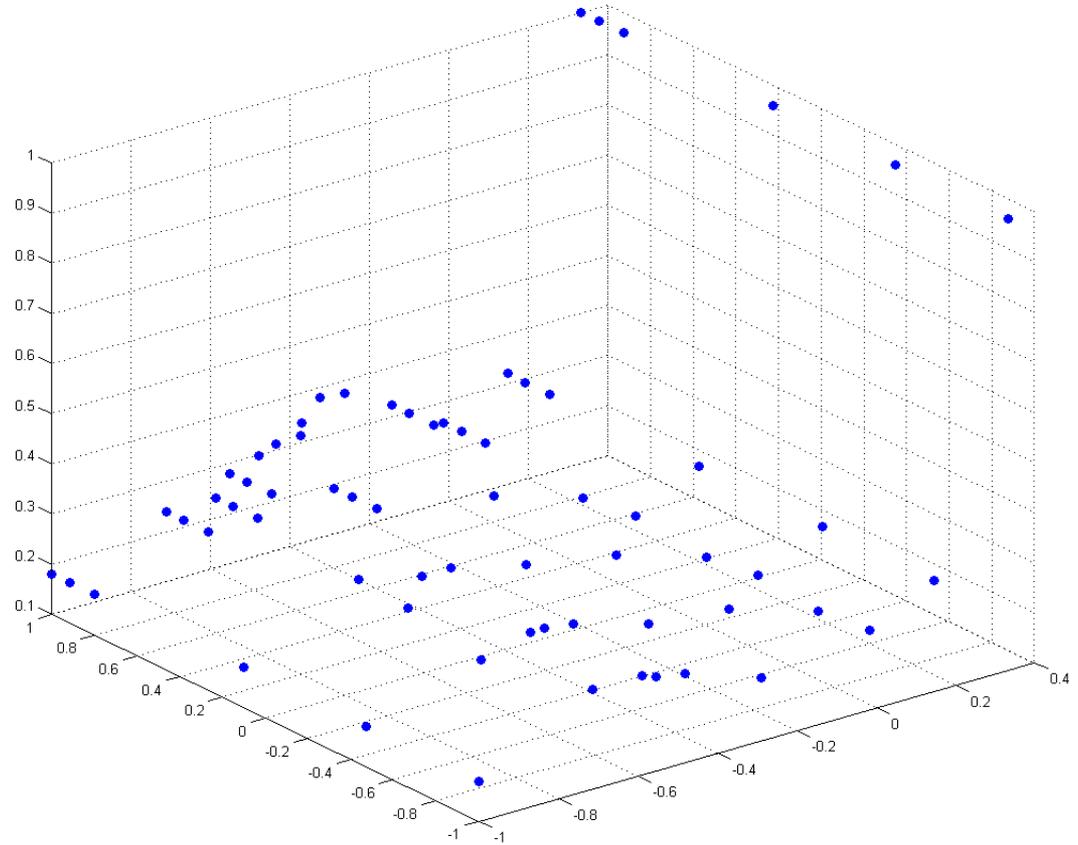


# Keyword File

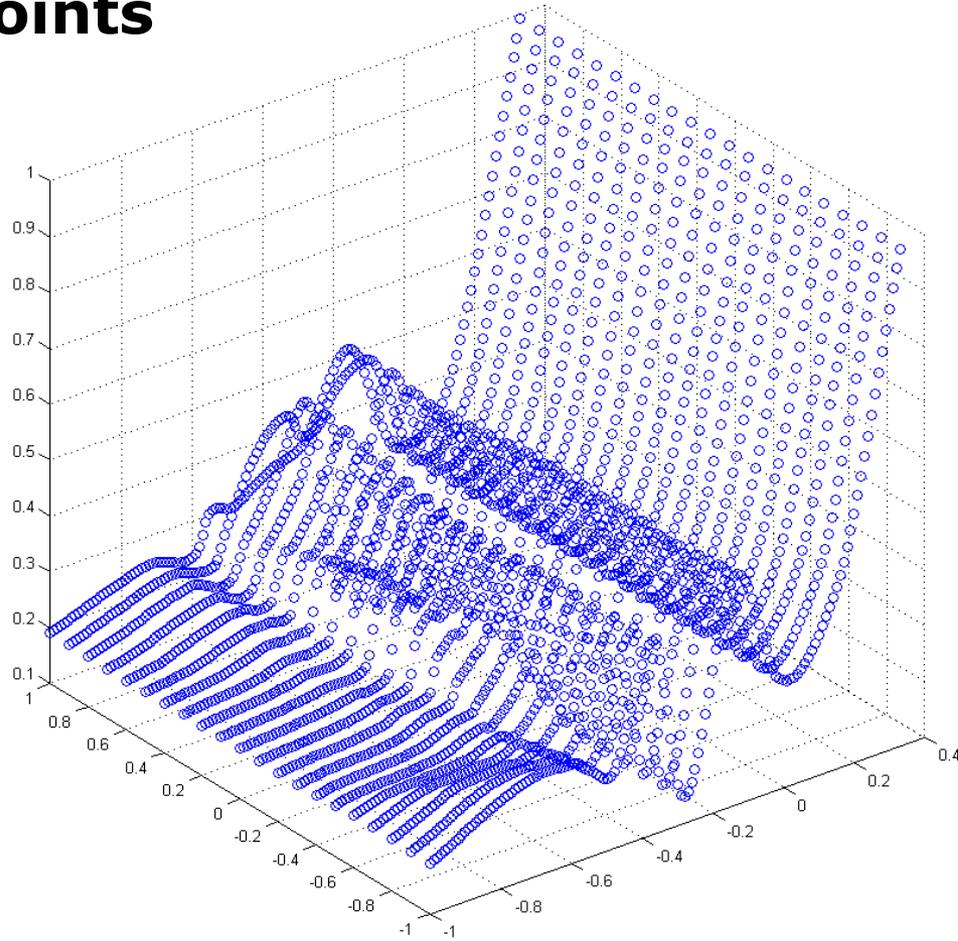


## Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

Test #	GMU		
	Triaxiality	Lode Parameter	Failure Strain
SG1	-0.390	0.975	0.460
SG2	-0.412	0.935	0.420
SG3	-0.475	0.803	0.380
SG4	-0.592	0.005	0.135
SG5	-0.370	1.000	0.340
SG6	-0.480	1.000	0.300
SG7	-0.553	1.000	0.280
SG8	-0.588	1.000	0.240
SG9	-0.712	1.000	0.240
SG10	-1.000	1.000	0.180
SG11	-0.573	0.146	0.260
SG12	-0.643	0.099	0.220
SG13	-0.691	0.054	0.290
LR1	-0.289	0.949	0.191
LR2	-0.145	0.917	0.325
LR3	-0.014	0.059	0.259
LR4	0.147	-0.450	0.321
Punch 1	-0.653	-0.969	0.297
Punch 2	-0.565	-0.475	0.276

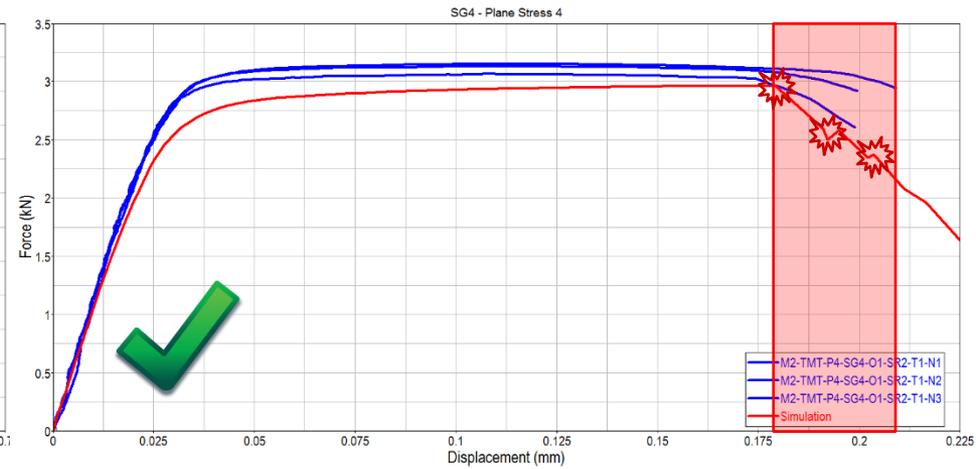
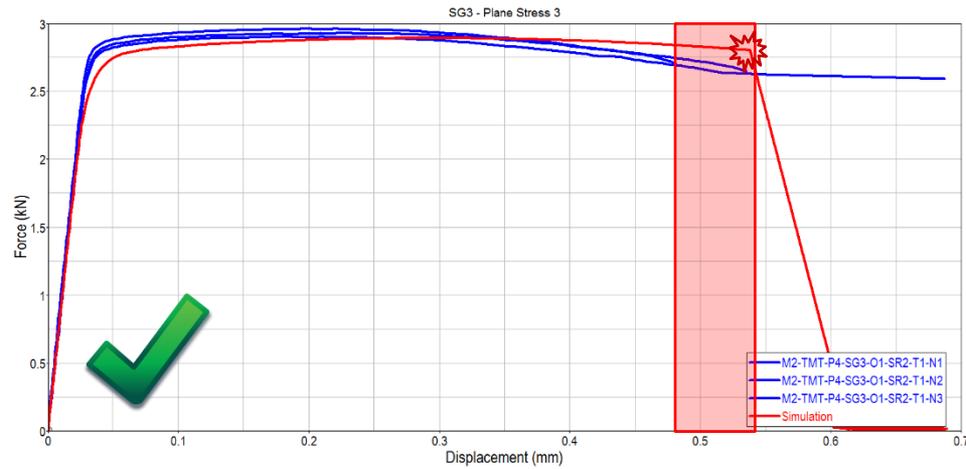
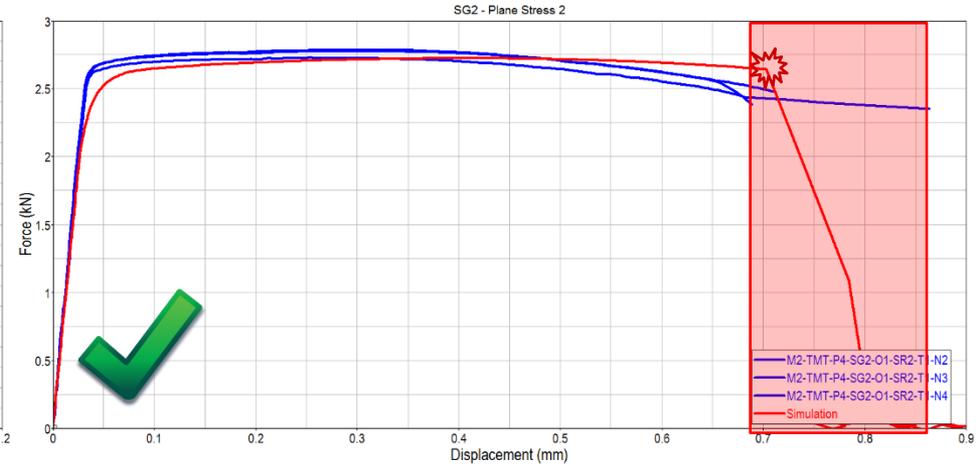
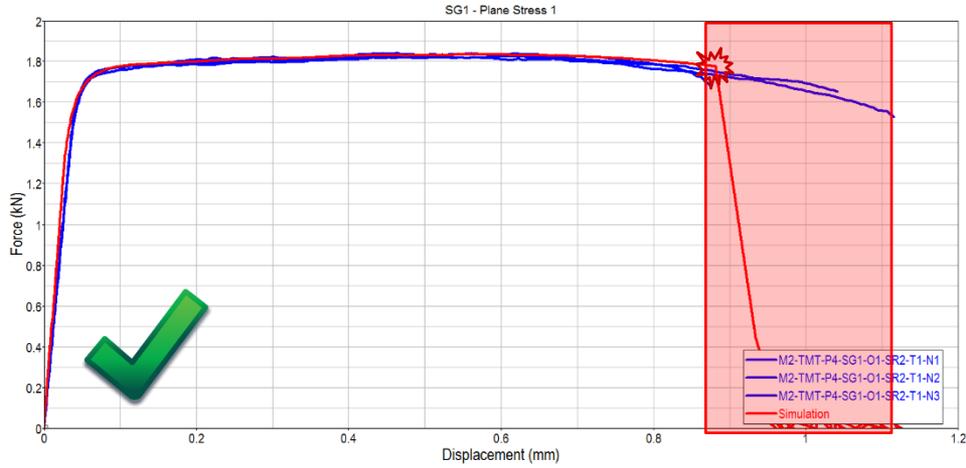


## Surface Points



# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

## Plane Stress



Correct Failure



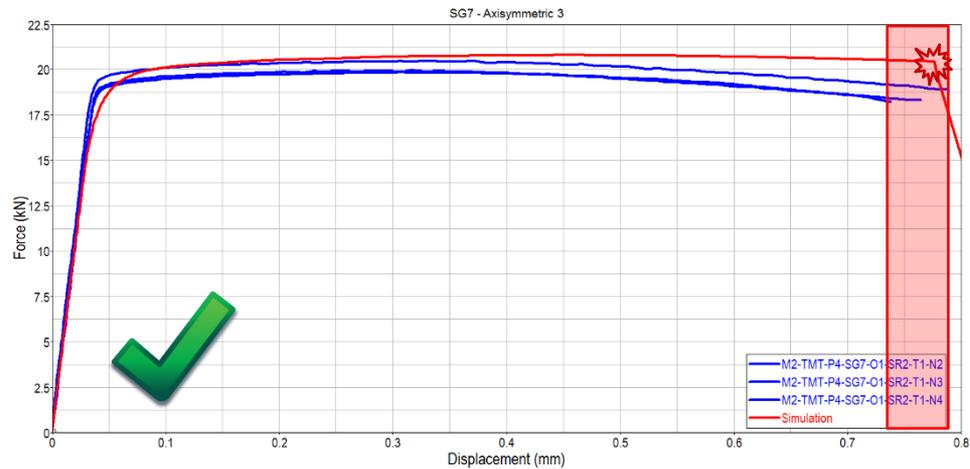
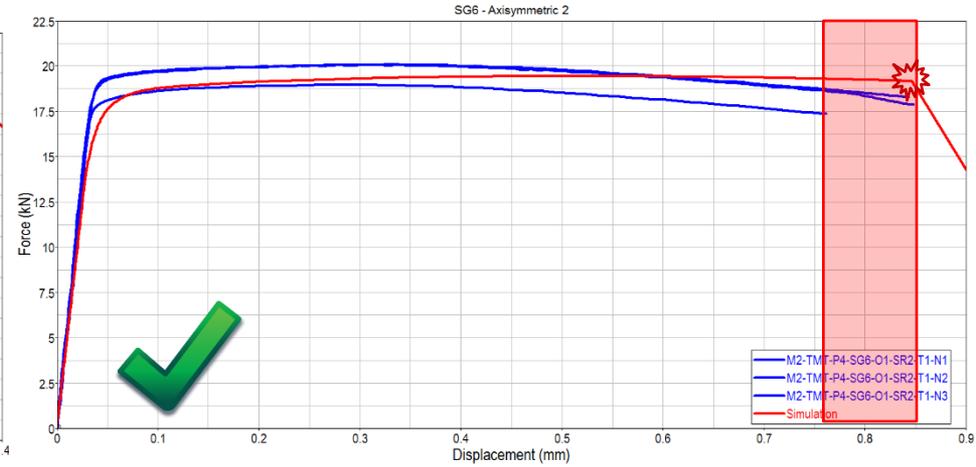
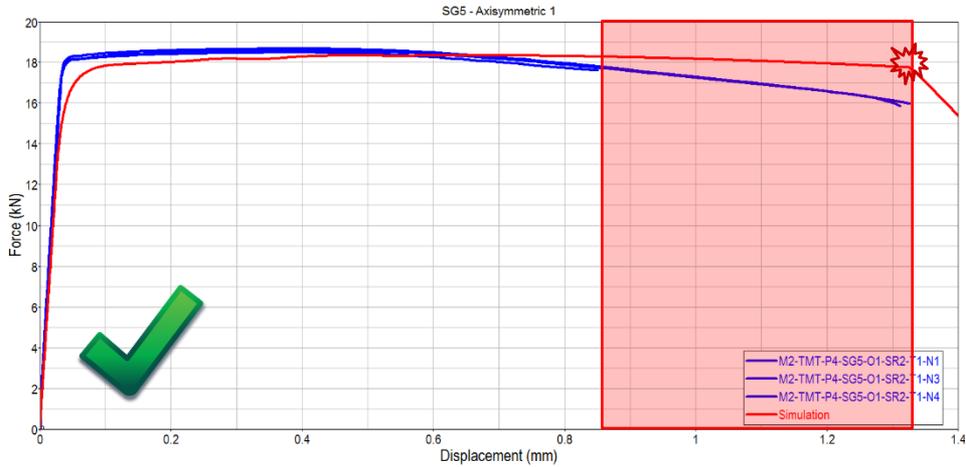
Incorrect Failure



No Failure

# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

## Axisymmetric



Correct Failure



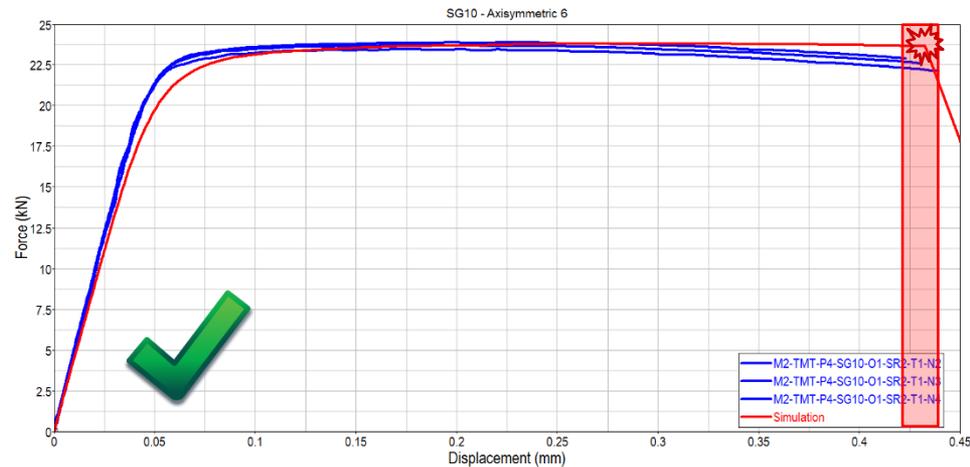
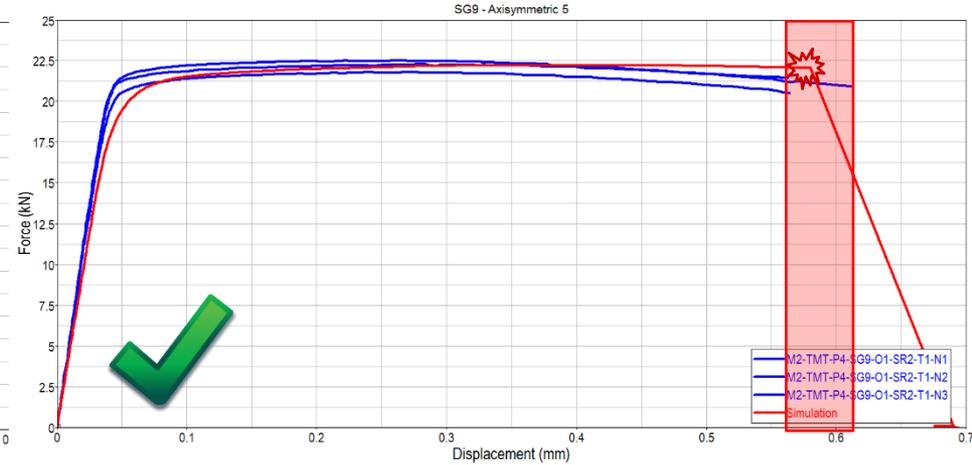
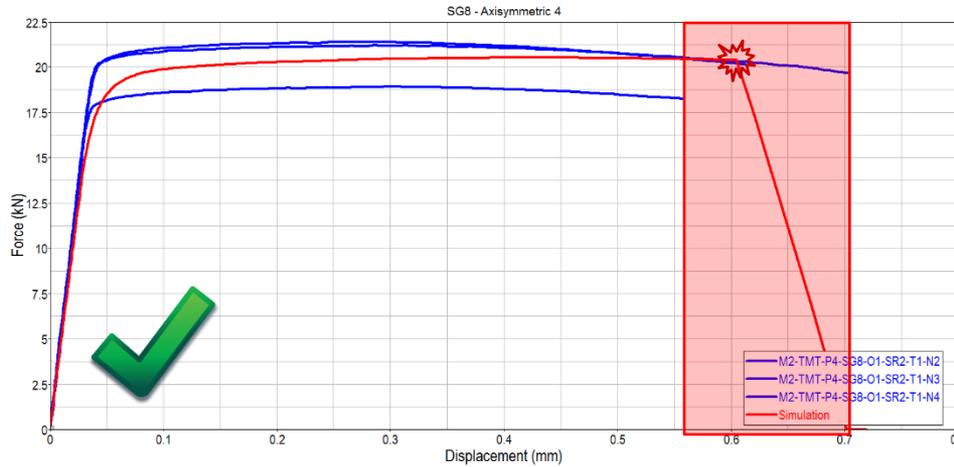
Incorrect Failure



No Failure

# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

## Axisymmetric



Correct Failure



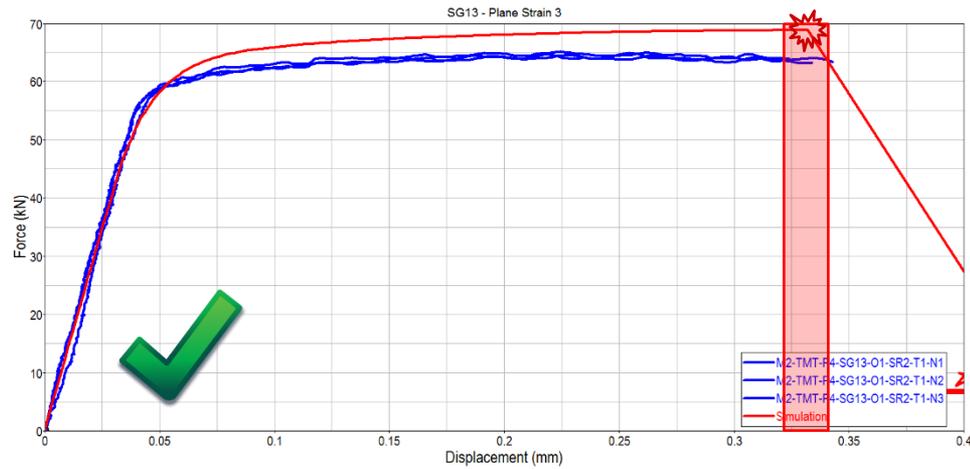
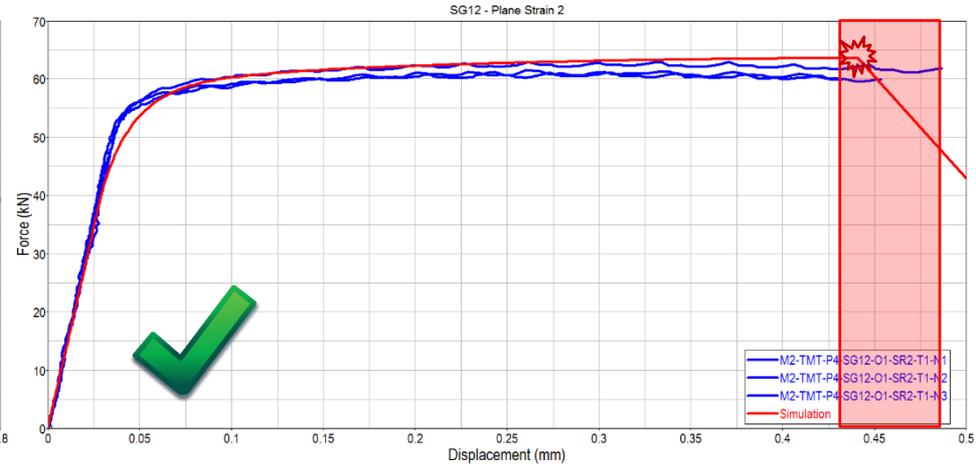
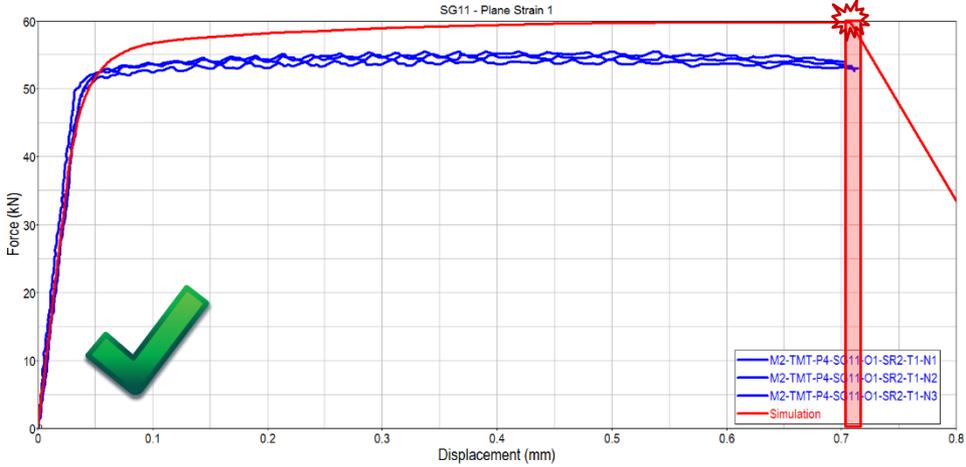
Incorrect Failure



No Failure

# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

## Plane Strain



Correct Failure

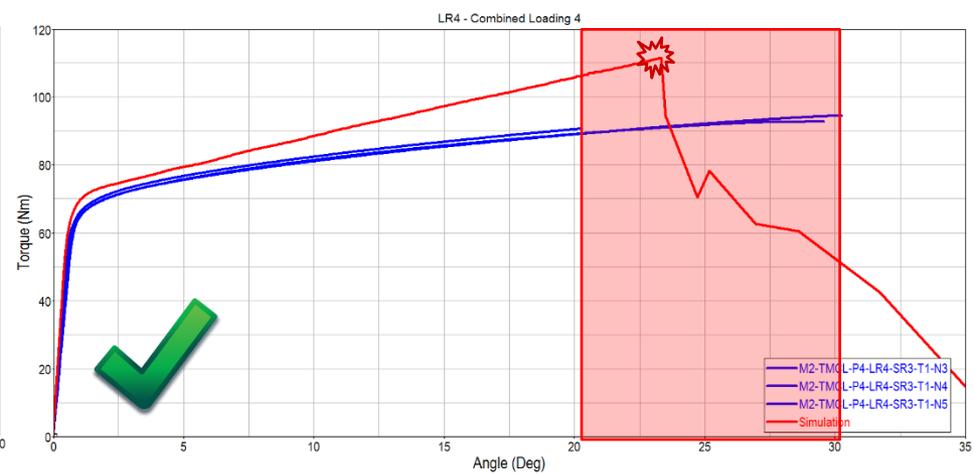
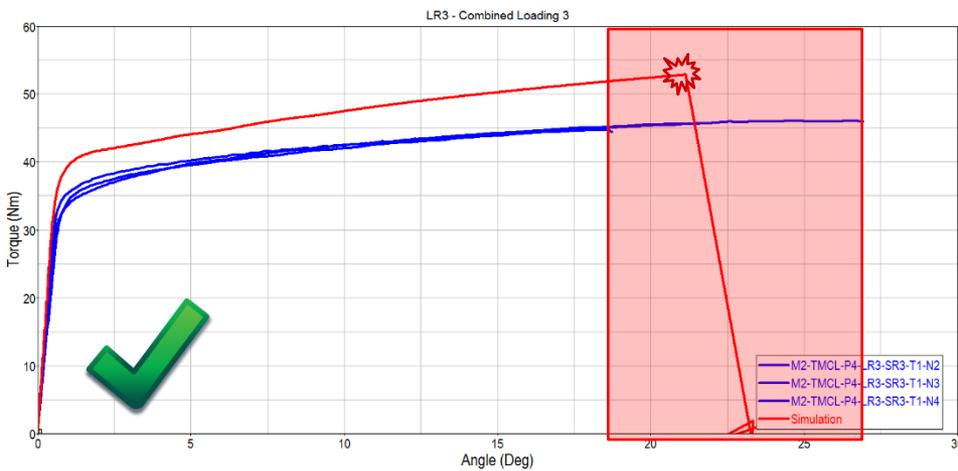
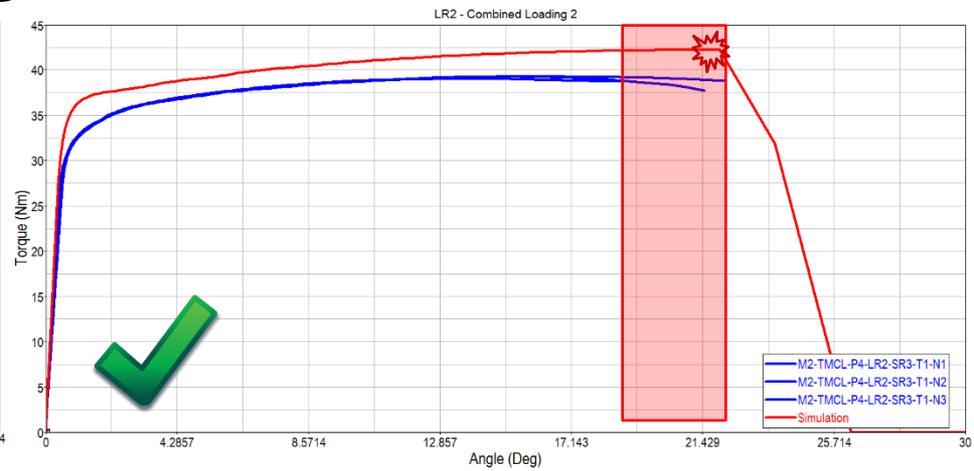
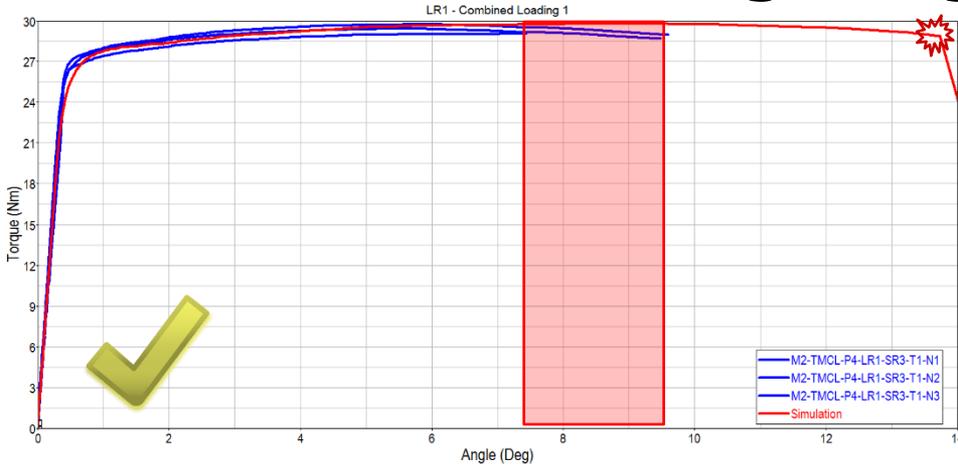


Incorrect Failure



No Failure

## Combined Loading - Angle



Correct Failure

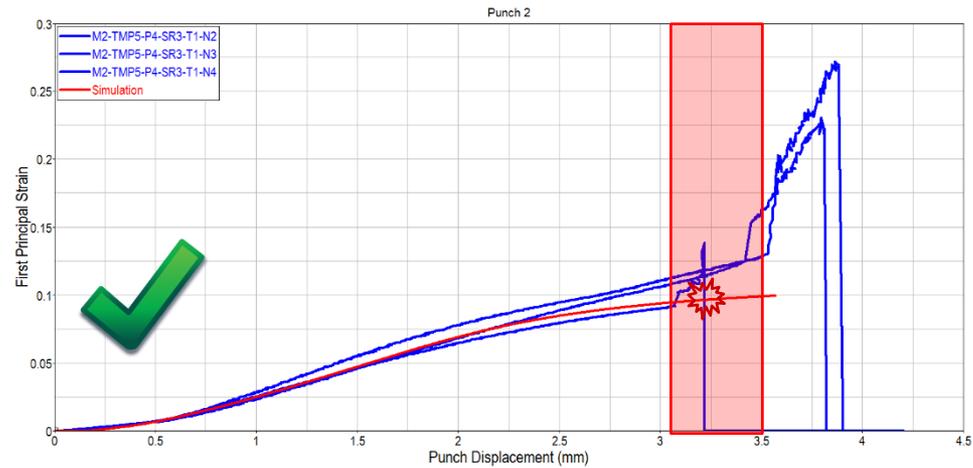
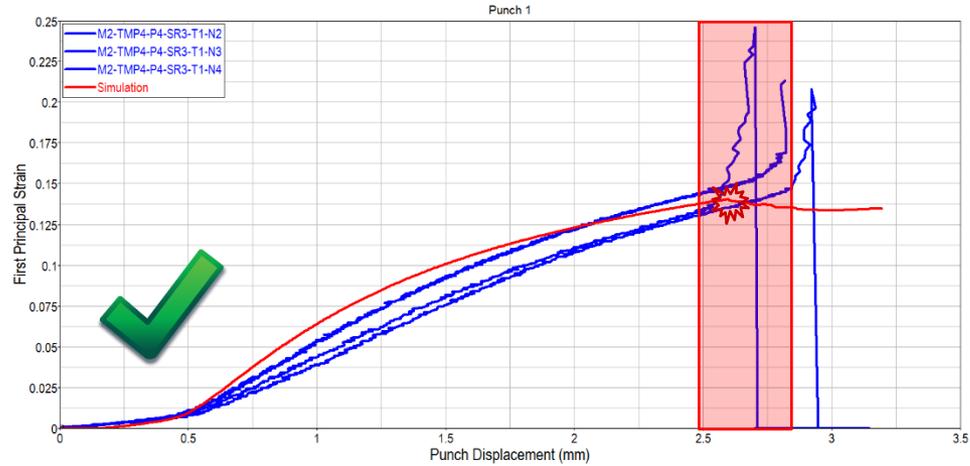


Incorrect Failure



No Failure

## Punch Tests



Correct Failure

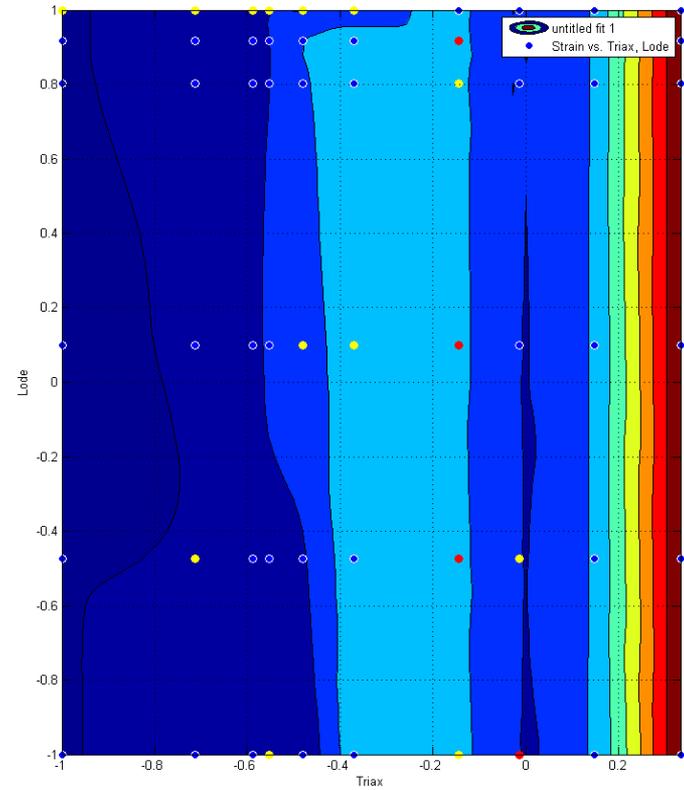
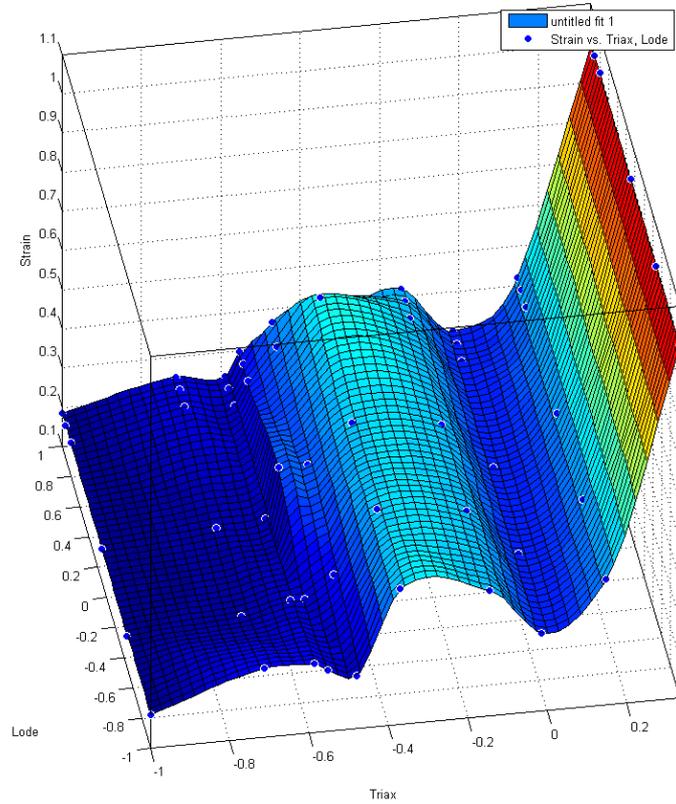


Incorrect Failure



No Failure

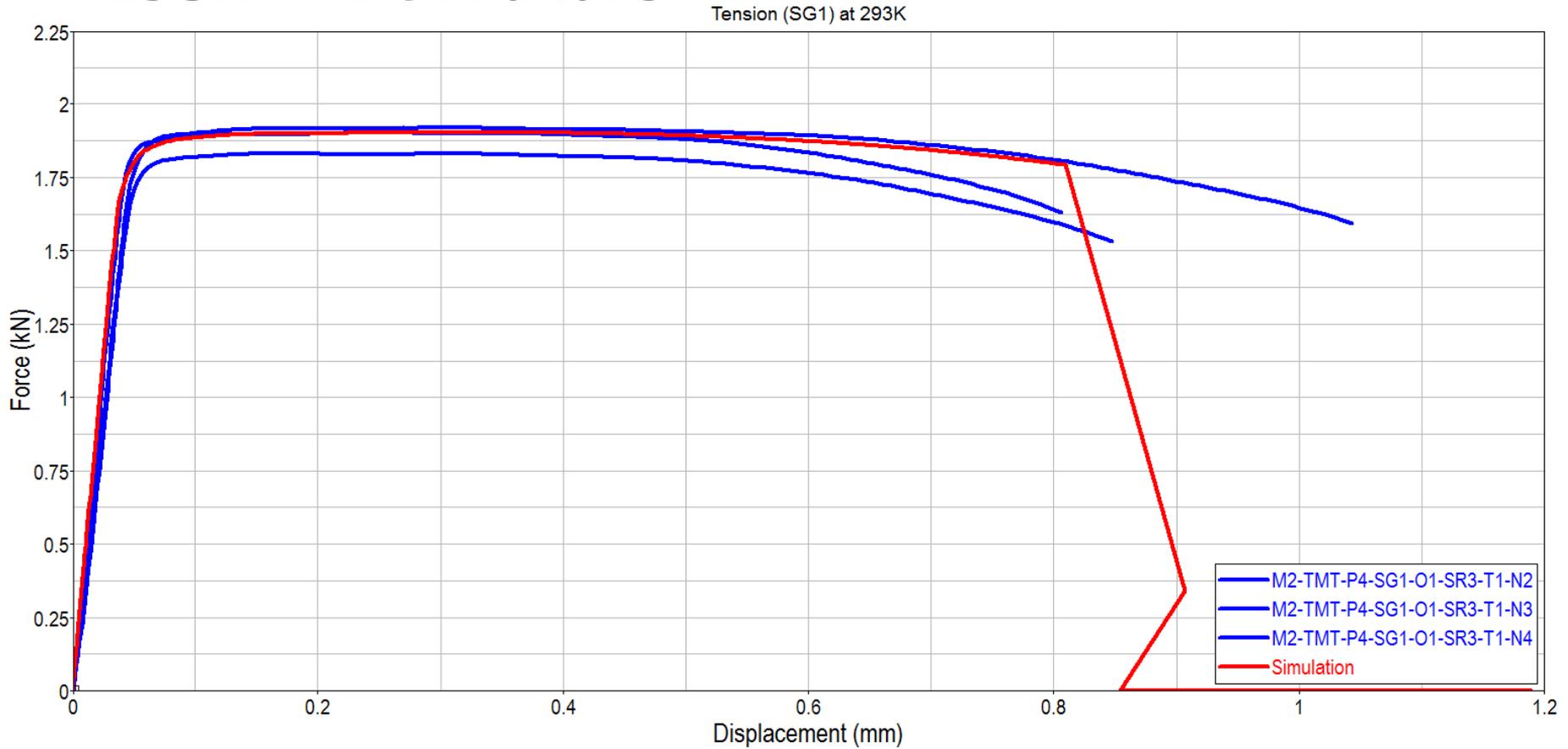
## Final Failure Surface



## Temperature dependency

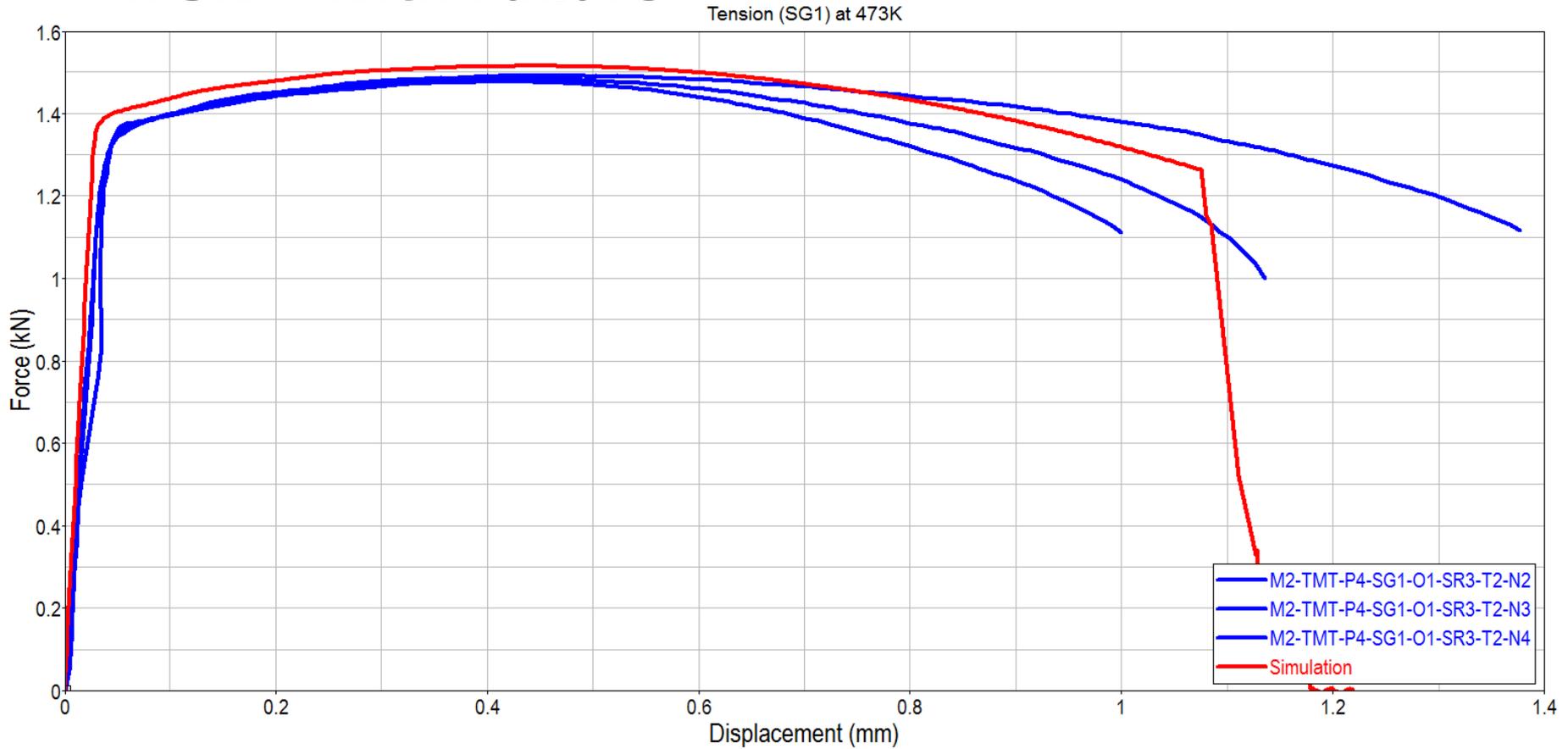
- **Generate scaling factor for failure surface based on temperature simulation results**
- **Temperature tests were quasi-static, so we assume no rate effects, they were also isothermal, so we assume no adiabatic (heat) effects**
- **Therefore every temperature test is simulated at an arbitrary speed (1000 times higher than reality) using a SINGLE hardening curve specified as the rate dependent 'table'**
- **This allows to determine a failure strain for every temperature in the usual way by matching the displacement at failure**
- **From the failure strain for every temperature we derive a scaling function for the failure surface, thus we assume that the temperature effects on the failure surface are independent of the state of stress**
- **This is 'optimistic' but compatible with the multiplicative split approach used in the Johnson-Cook type laws**

## 293K – With Failure



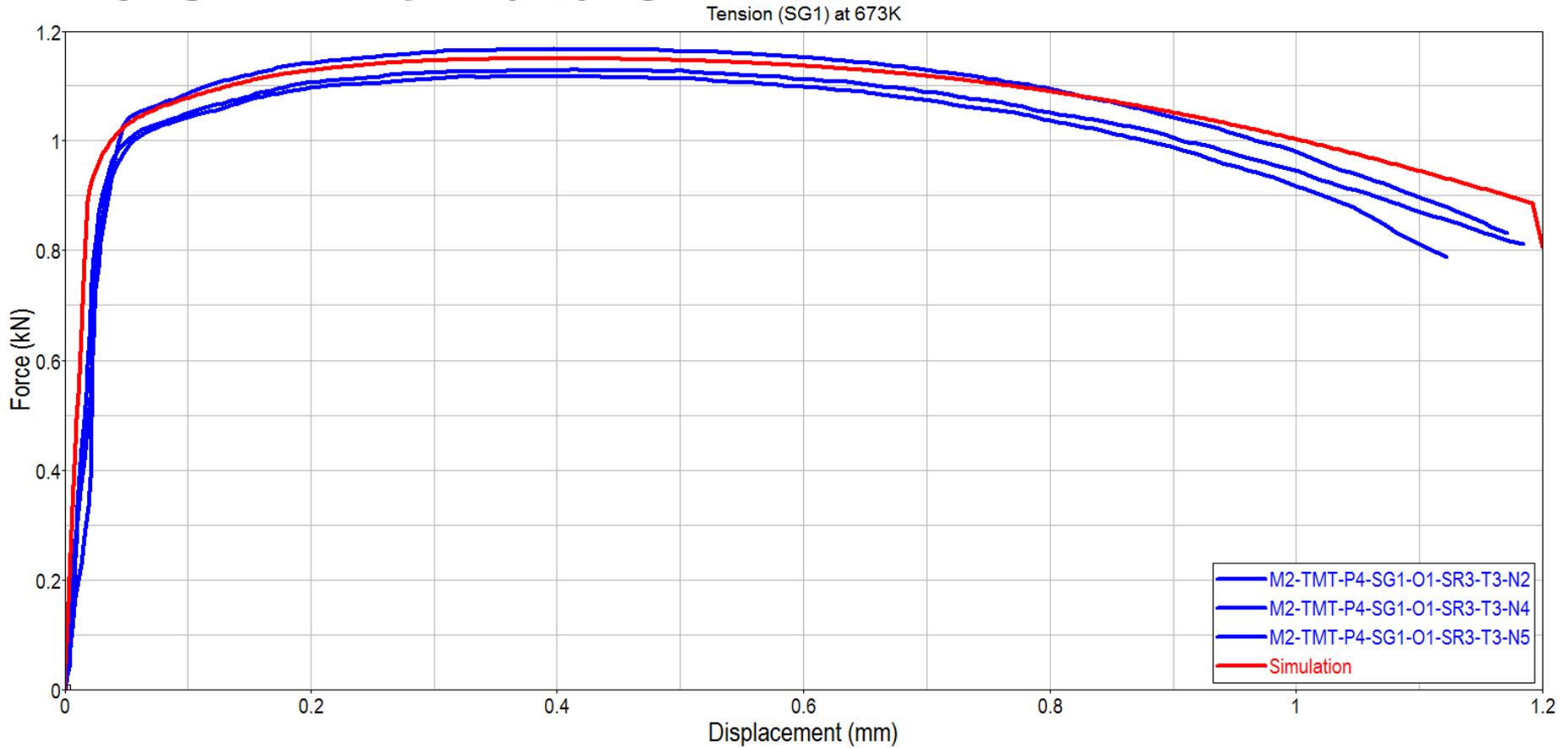
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```

## 473K – With Failure



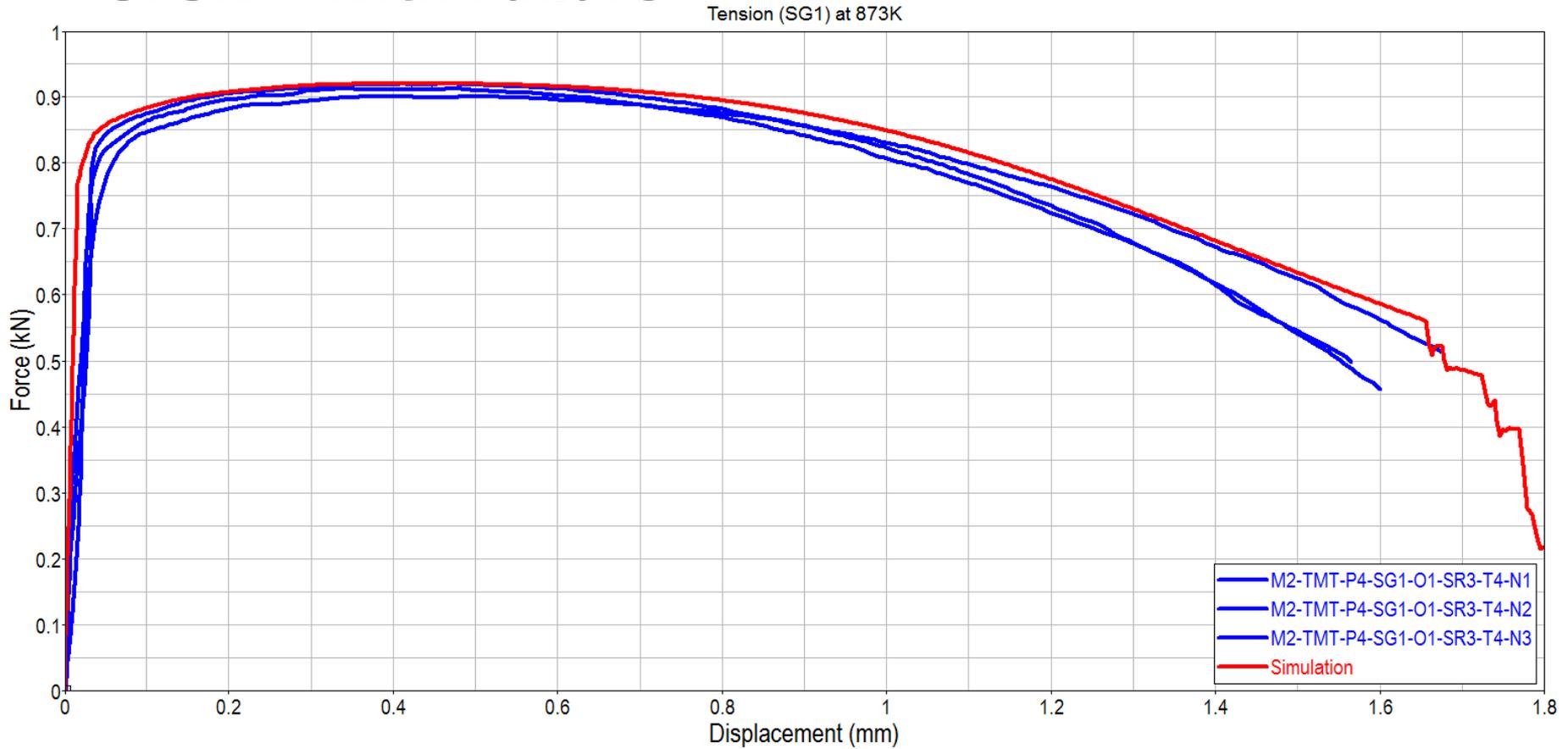
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$ mid ro e pr cp tr beta numint
2 4.4300E-6 110.00000 0.342000 526.3 293 0.800000 1.000000
$ lck1 lckt lcf lcg lch lci
8004 8004 3000 5000
```

## 673K – With Failure



```
*MAT_TABULATED_JOHNSON_COOK_TITLE
MAT_224_Ti64
$ mid ro e pr cp tr beta numint
  2 4.4300E-6 110.00000 0.342000 526.3 293 0.800000 1.000000
$ lck1 lckt lcf lcg lch lei
  8005 8005 3000 5000
```

## 873K – With Failure



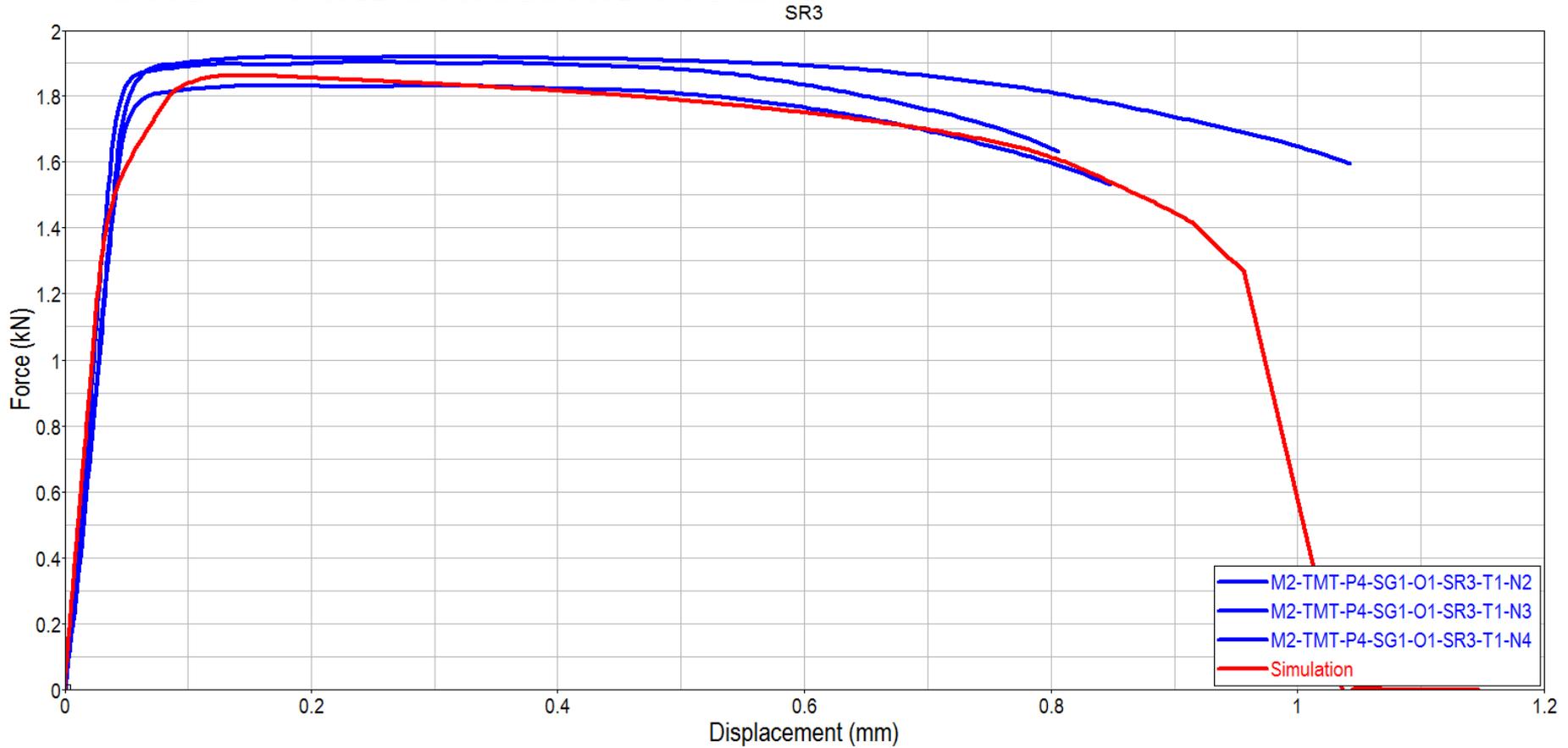
```
*MAT_TABULATED_JOHNSON_COOK_TITLE
MAT_224_Ti64
$ mid ro e pr cp tr beta numint
  2 4.4300E-6 110.00000 0.342000 526.3 293 0.800000 1.000000
$ lck1 lck2 lcf lcg lch lci
  8006 8006 3000 5000
```



## Strain Rate dependency

- We now simulate the rate dependent tests SR3, SR4 and SR5 at real life velocities using the full input: rate dependent table, temperature dependent table, failure surface, temperature scaling function and strain rate scaling function
- SR3 is 1/s, SR4 is 600/s and SR5 is 1500/s, in dynamic experiments we have temperature as well as strain rate effects
- Some tuning of the strain rate scaling function is necessary to obtain good failure response in all 3 cases

## SR3 - Full Material Model

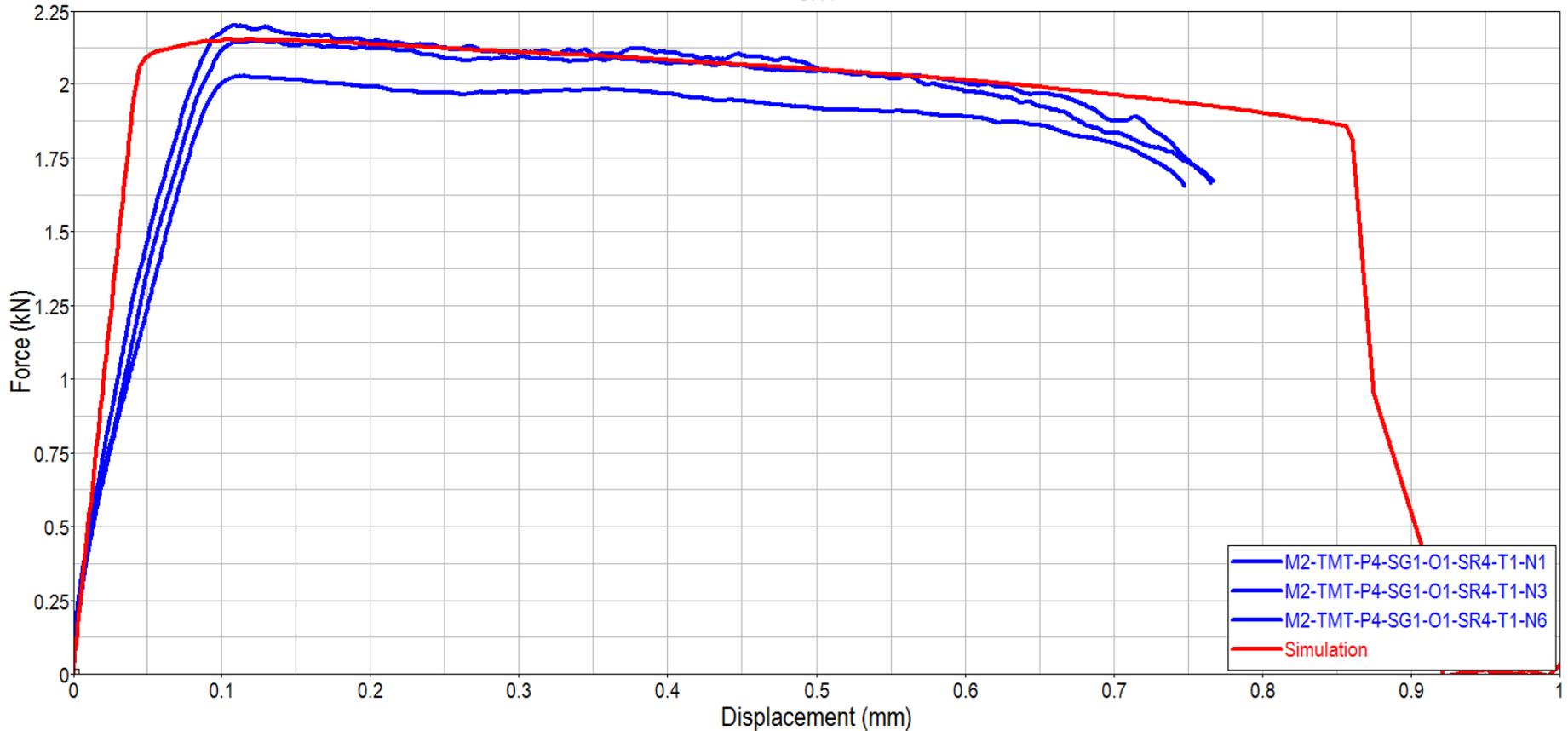


```
*MAT_TABULATED_JOHNSON_COOK_TITLE
MAT_224_Ti64
$ mid ro e pr cp tr beta numint
  2 4.4300E-6 110.00000 0.342000 526.3 293 0.800000 1.000000
$ lck1 lck2 lck3 lck4 lck5 lck6
  1000 2000 3000 4000 5000
```

## SR4 – Full Material Model

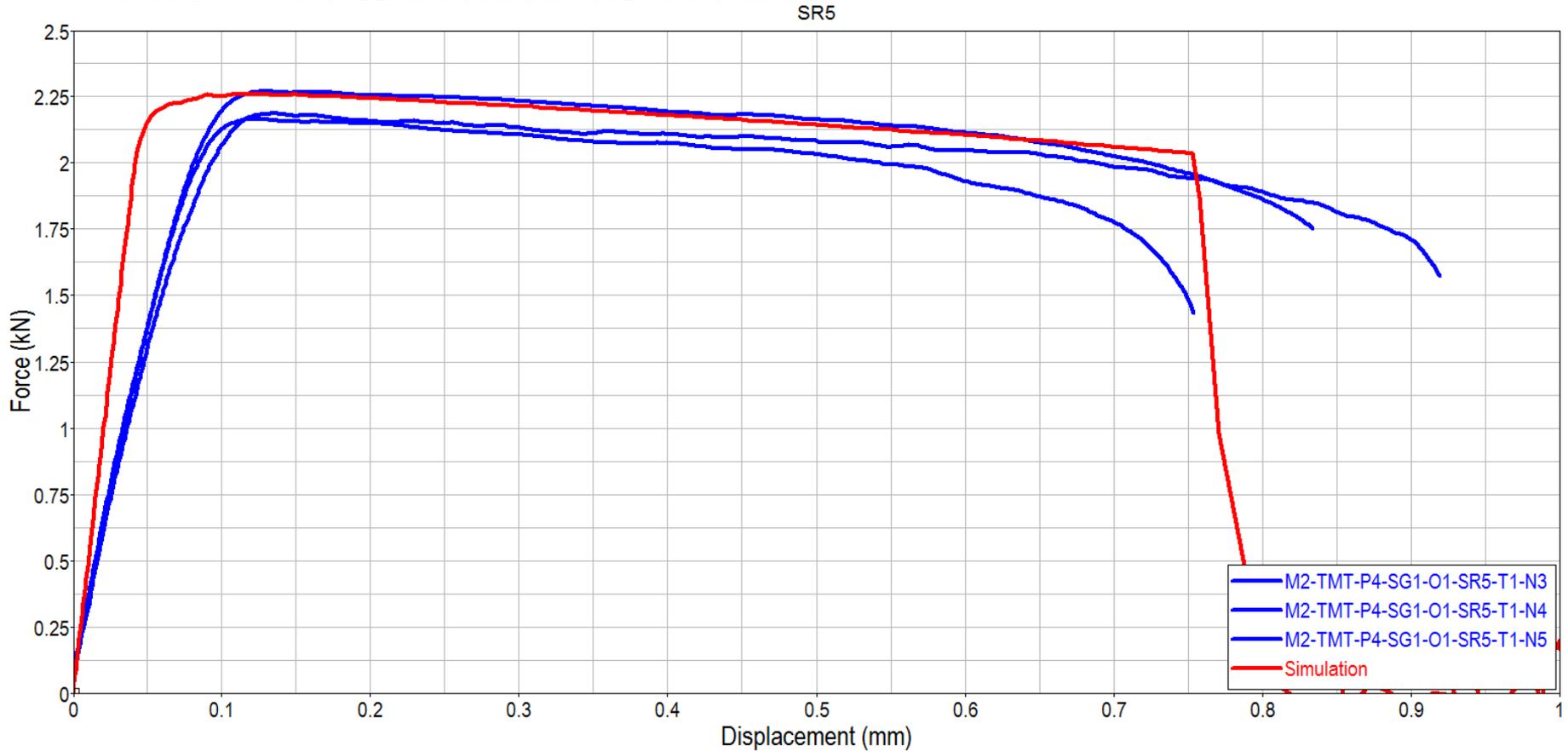
SR4

Failure displacement  
due to non-monotonic  
curve constraints



```
*MAT_TABULATED_JOHNSON_COOK_TITLE
MAT_224_Ti64
$ mid ro e pr cp tr beta numint
$ lck1 lck2 lck3 lck4 lck5 lck6 lck7 lck8 lck9 lck10
1000 2000 3000 4000 5000
```

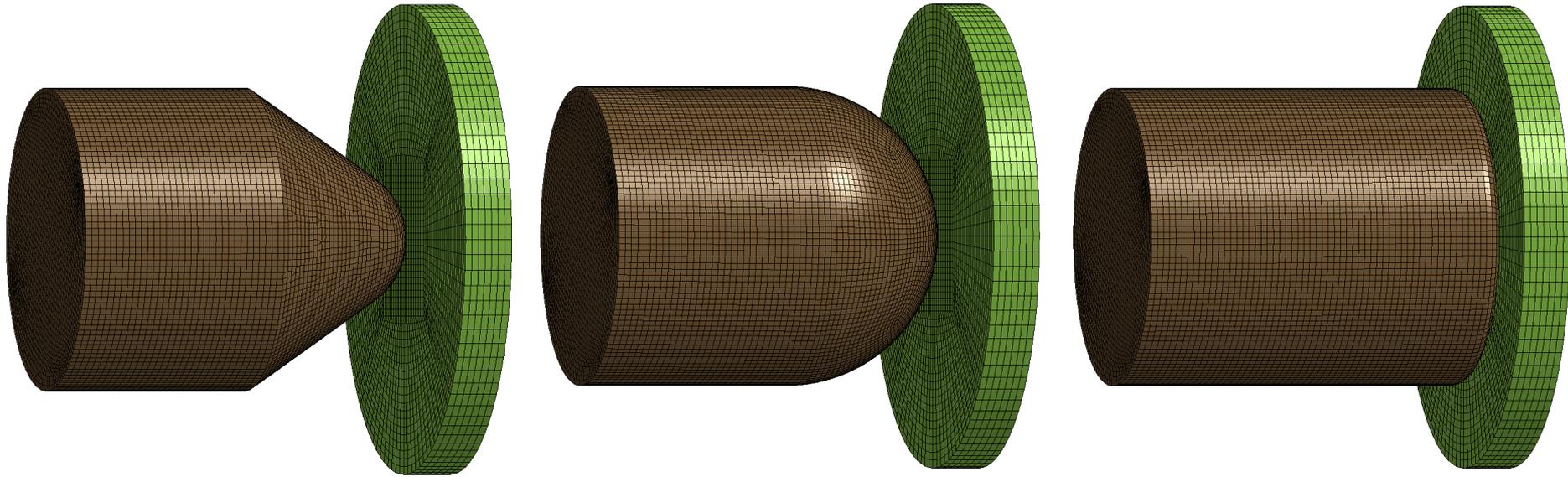
# SR5 - Full Material Model



```
*MAT_TABULATED_JOHNSON_COOK_TITLE
MAT_224_Ti64
$ mid ro e pr cp tr beta numint
  2 4.4300E-6 110.00000 0.342000 526.3 293 0.800000 1.000000
$ lck1 lck2 lck3 lck4 lck5 lck6
  1000 2000 3000 4000 5000
```



## Dynamic Punch Tests



### Full Material Model:

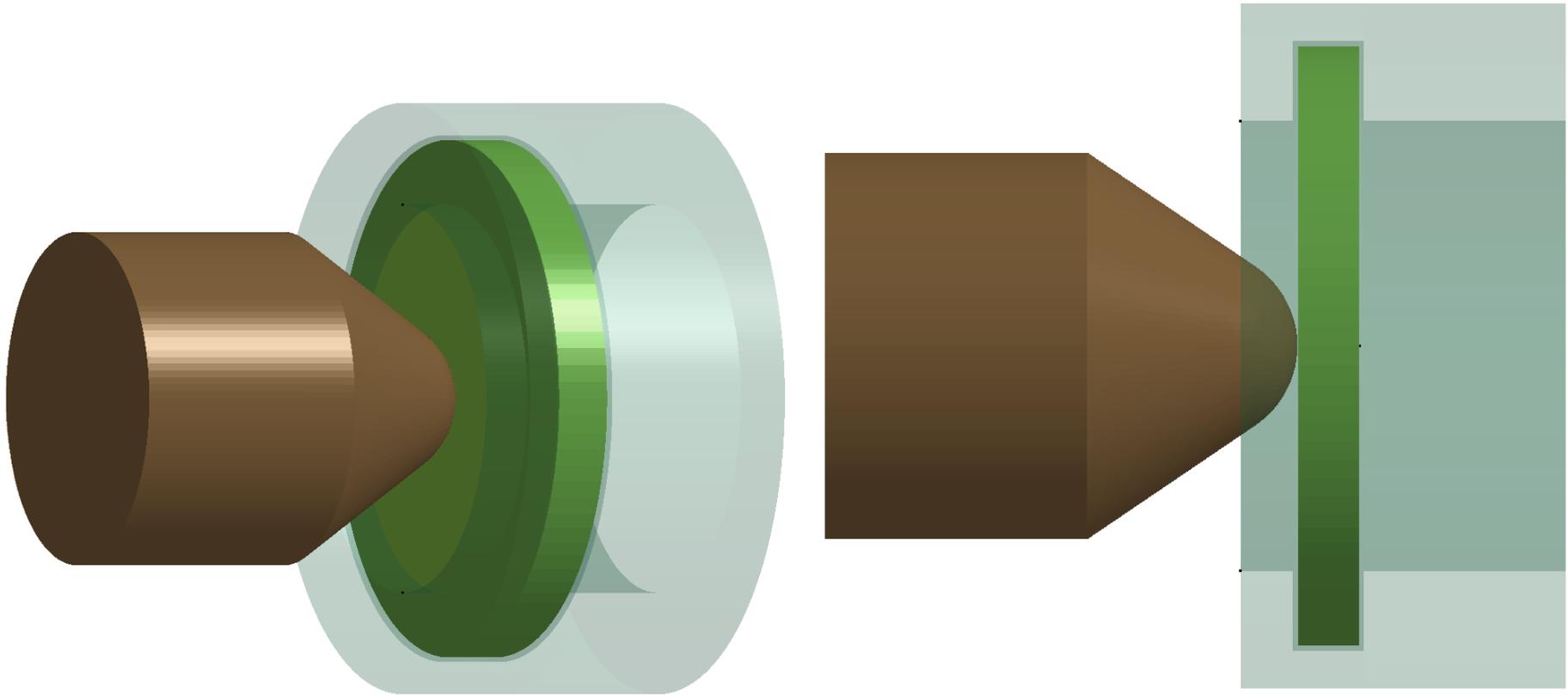
```
*MAT_TABULATED_JOHNSON_COOK_TITLE
```

```
MAT_224_Ti64
```

```
$      mid      ro      e      pr      cp      tr      beta      numint
      2 4.4300E-6 110.00000 0.342000 526.3 293 0.800000 1.000000
$      lck1      lckt      lcf      lcg      lch      lci
      1000      2000      3000      4000      5000
```

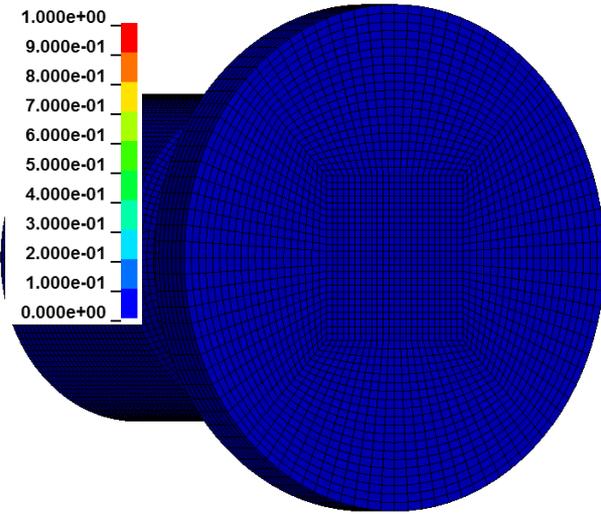
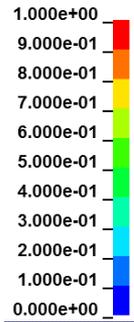
## Boundary Conditions

- The adapter and clamp assembly was modeled

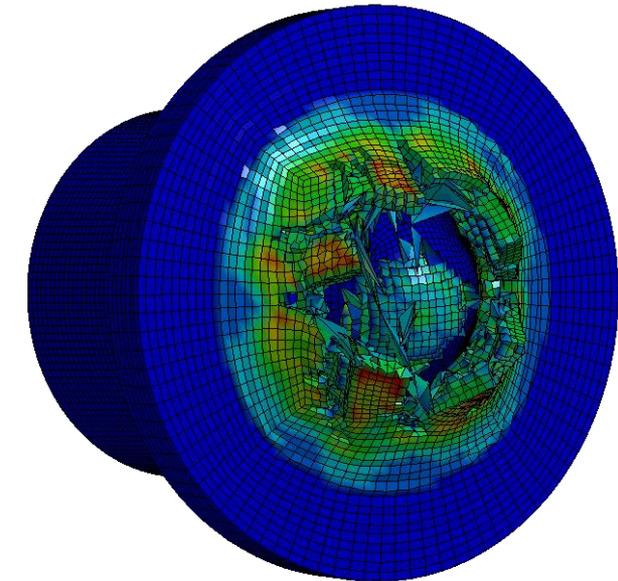
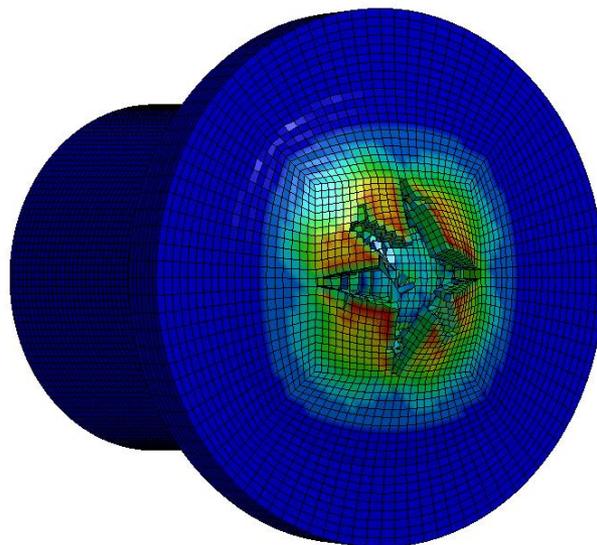
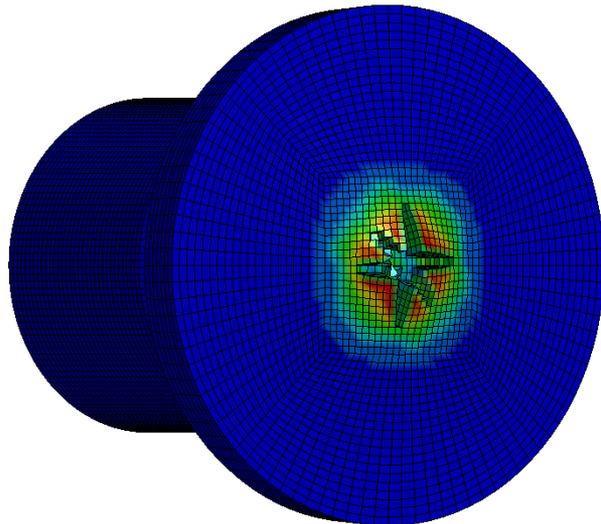
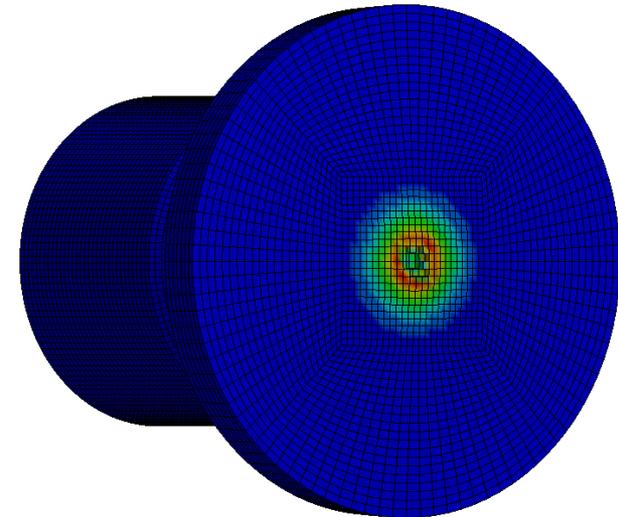
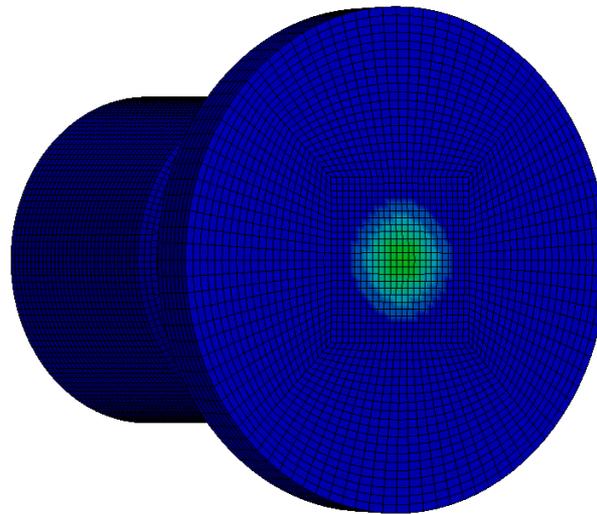


# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

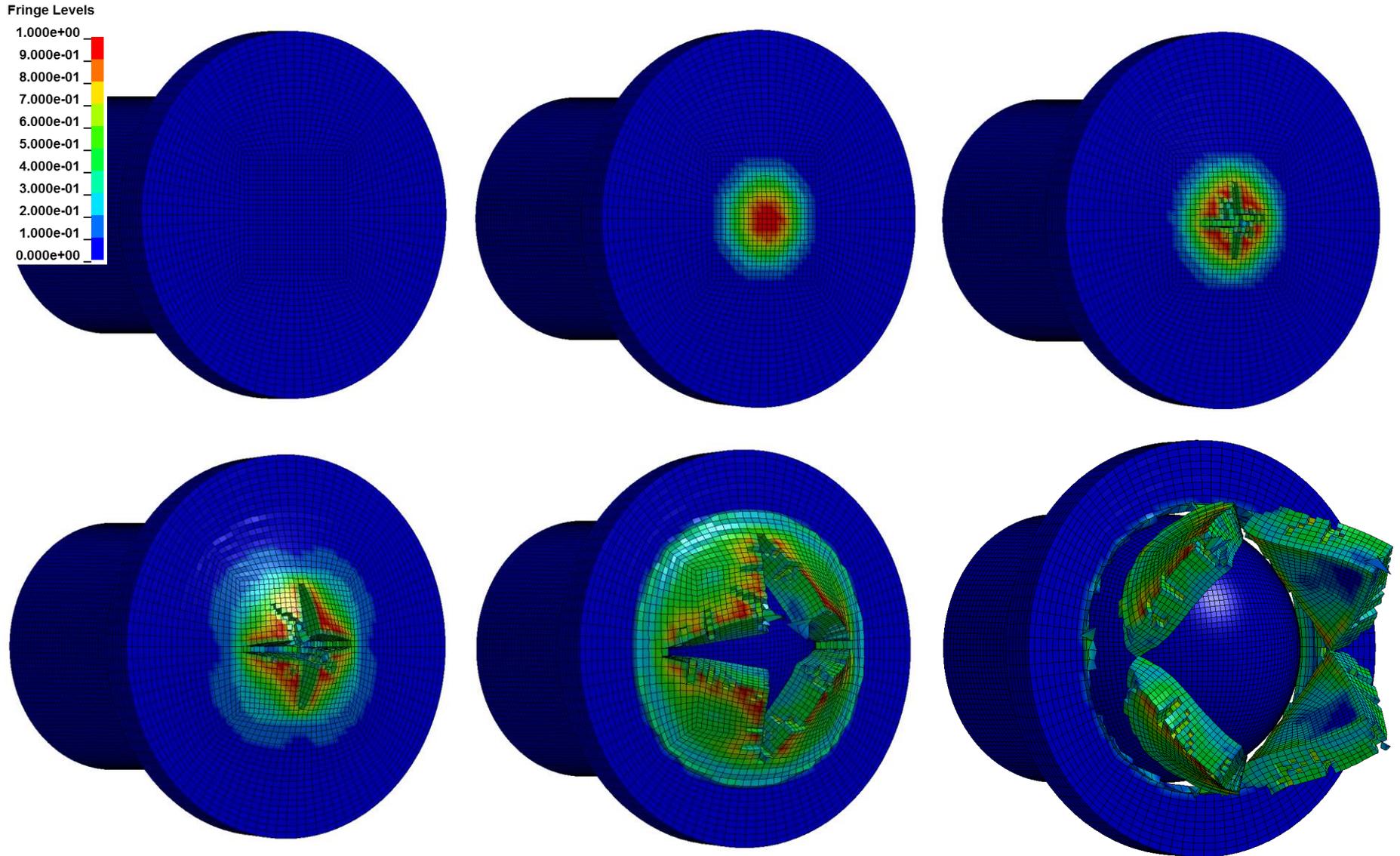
Fringe Levels



1  
2



# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

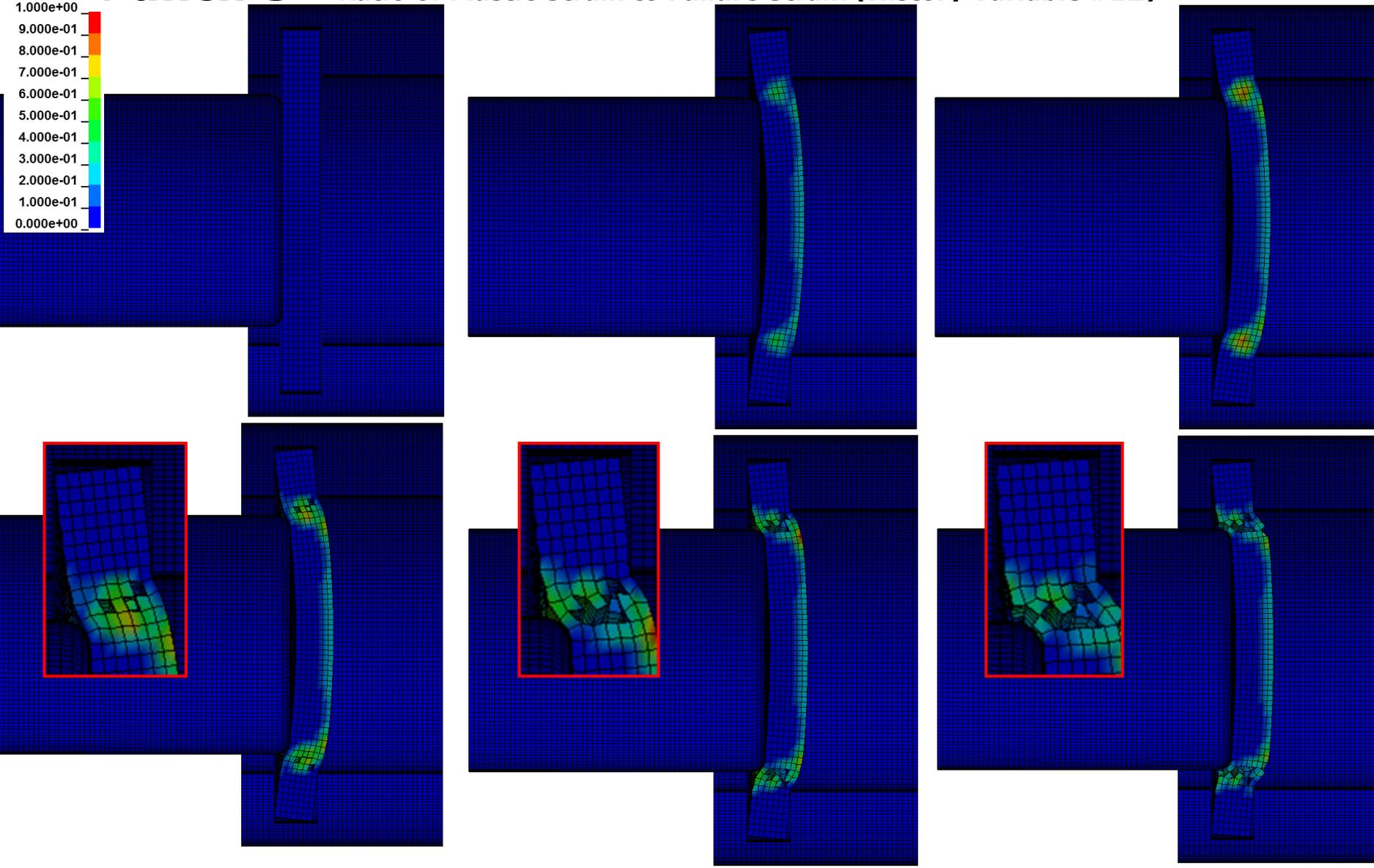


**Punch 2**

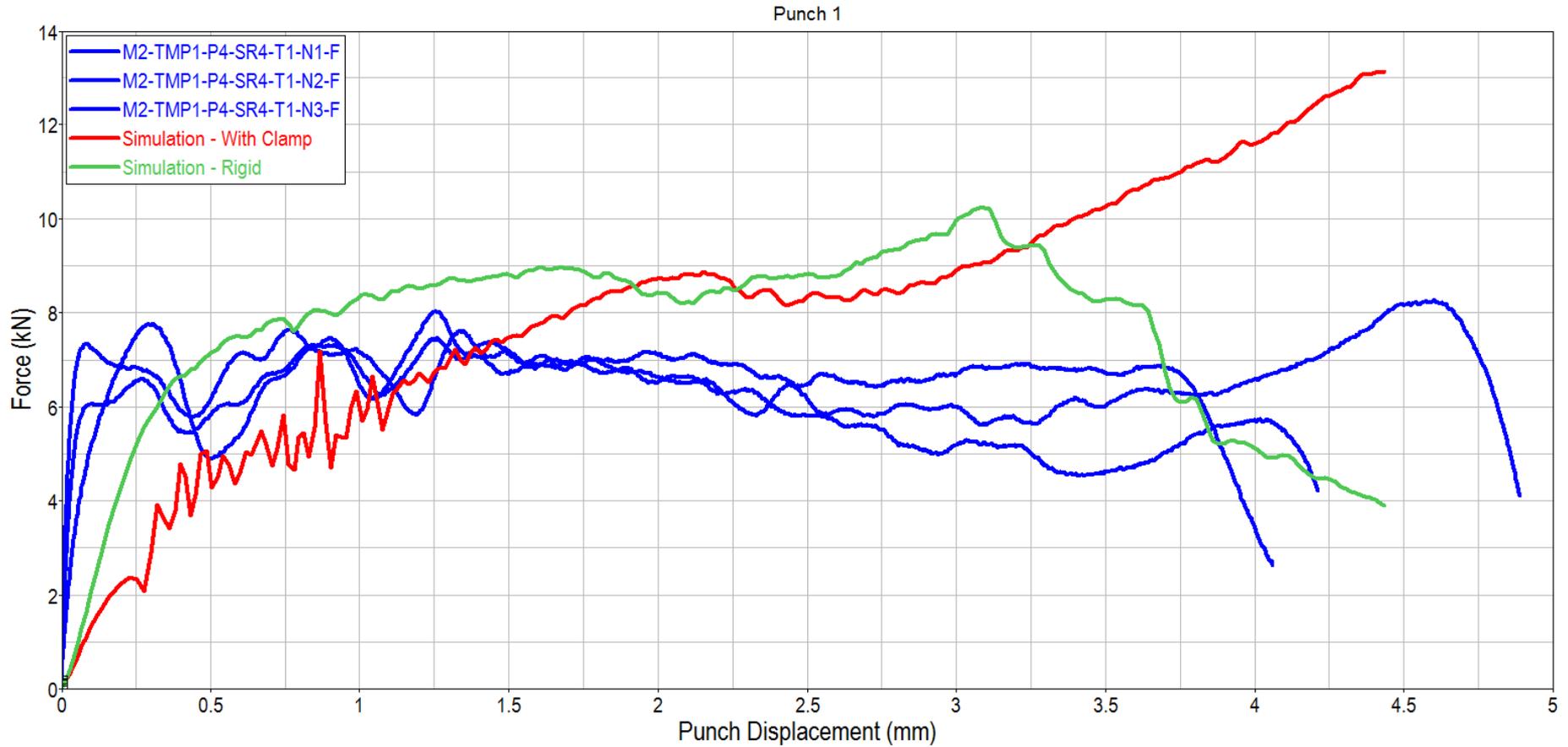
Ratio of Plastic Strain to Failure Strain (History Variable #12)

# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

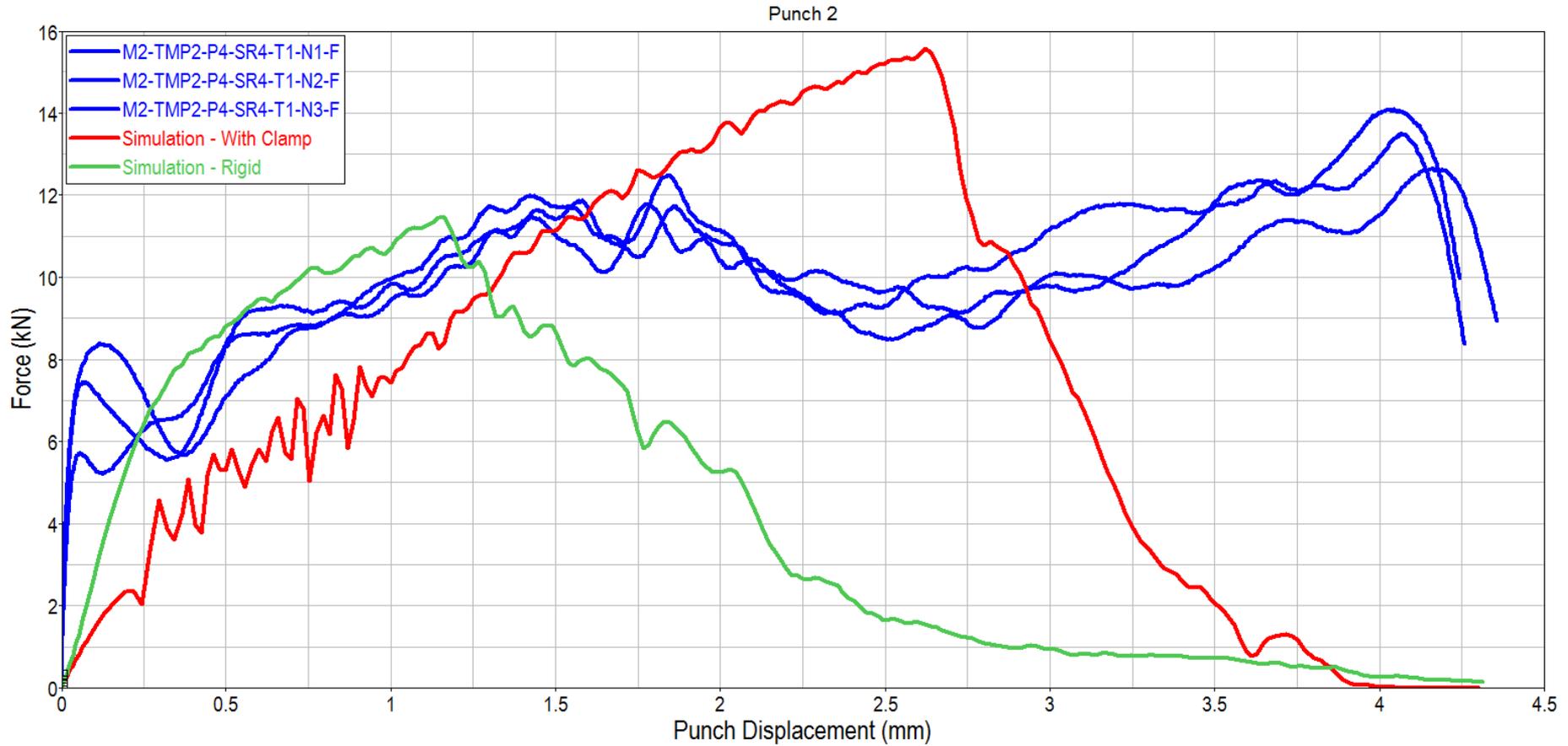
**Punch 3** Ratio of Plastic Strain to Failure Strain (History Variable #12)



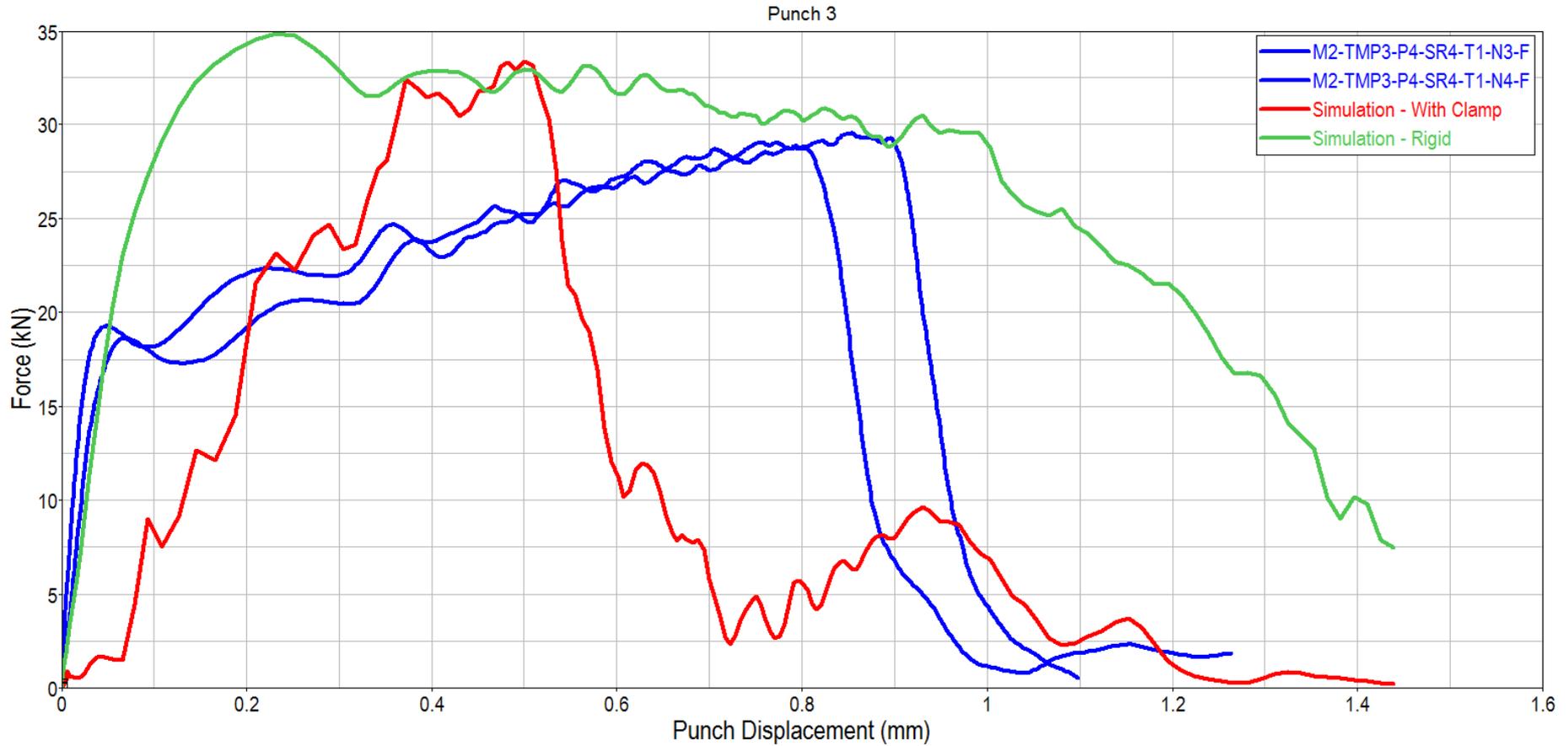
## Punch 1 Clamp vs. Rigid



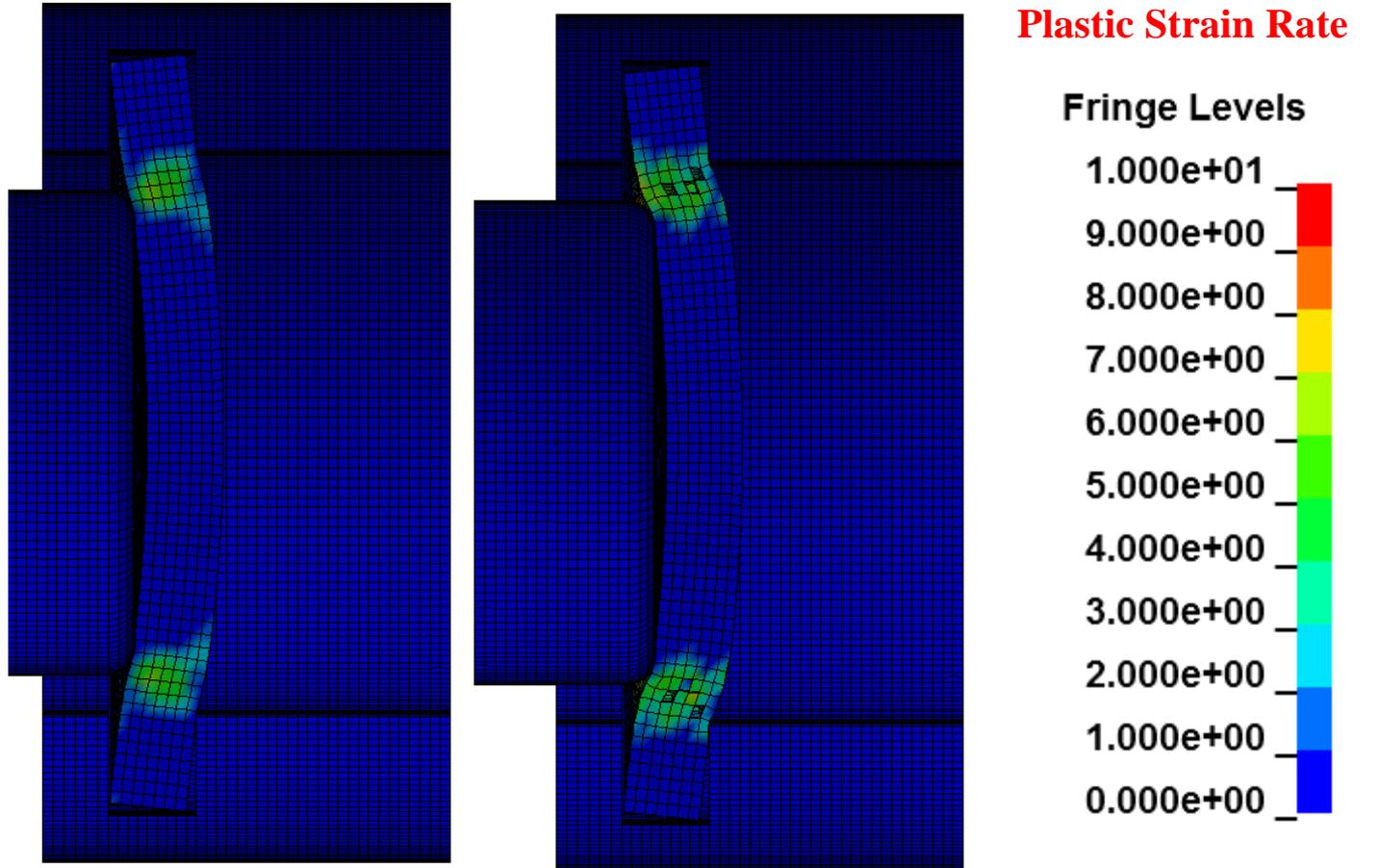
## Punch 2 Clamp vs. Rigid



# Punch 3 Clamp vs. Rigid



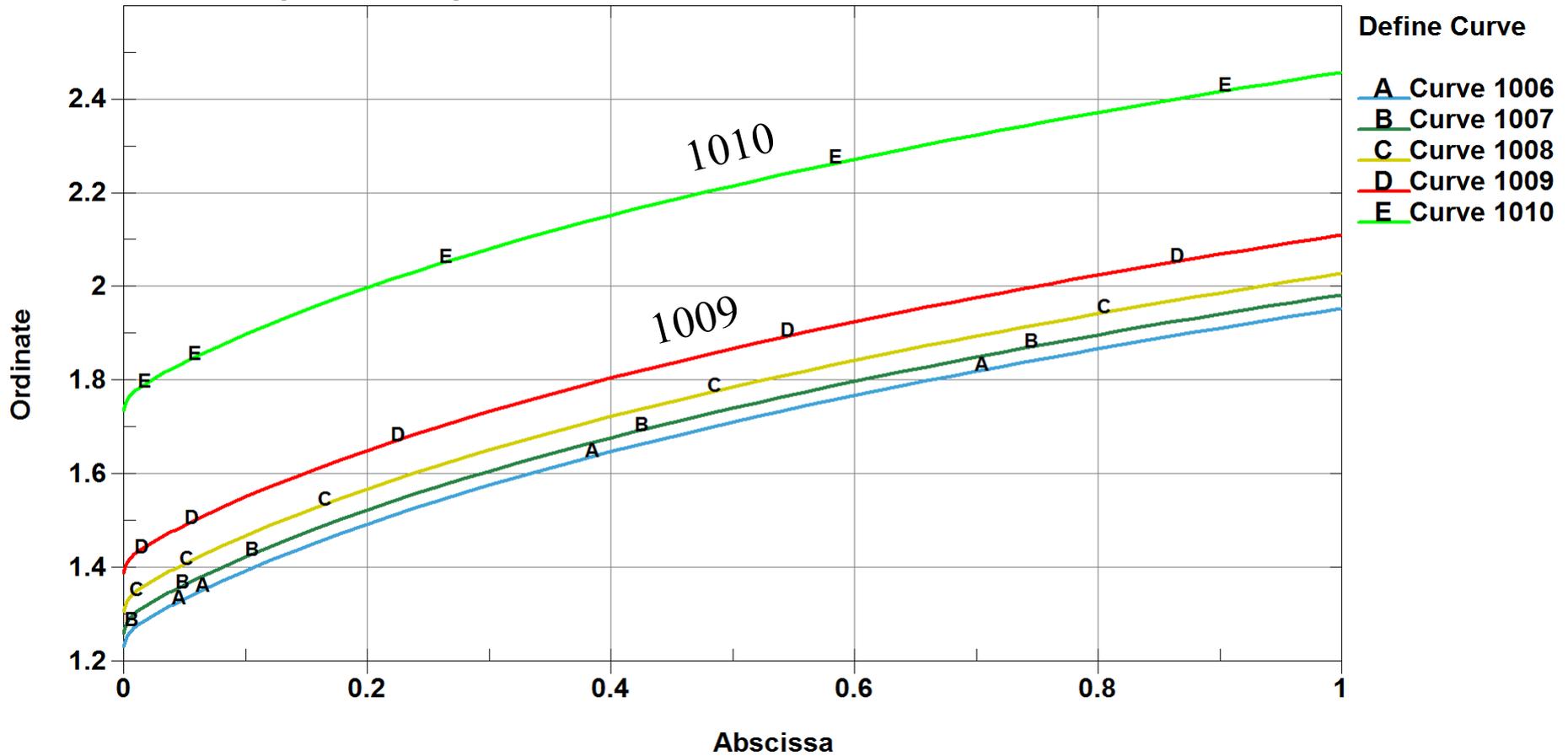
- Strain rates range from **5 1/ms** to **10 1/ms**
- This corresponds to curves **1009** and **1010**



## Strain Rates for Punch 3

# Yield Curves

LS-DYNA keyword deck by LS-PrePost



# NASA Ballistic Test Series

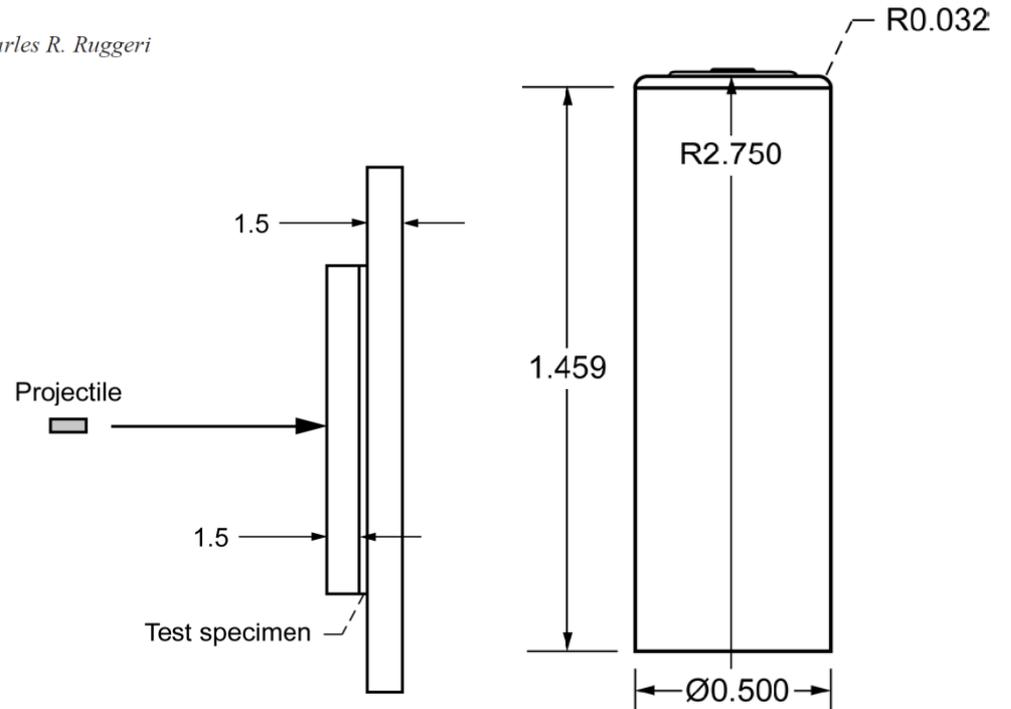
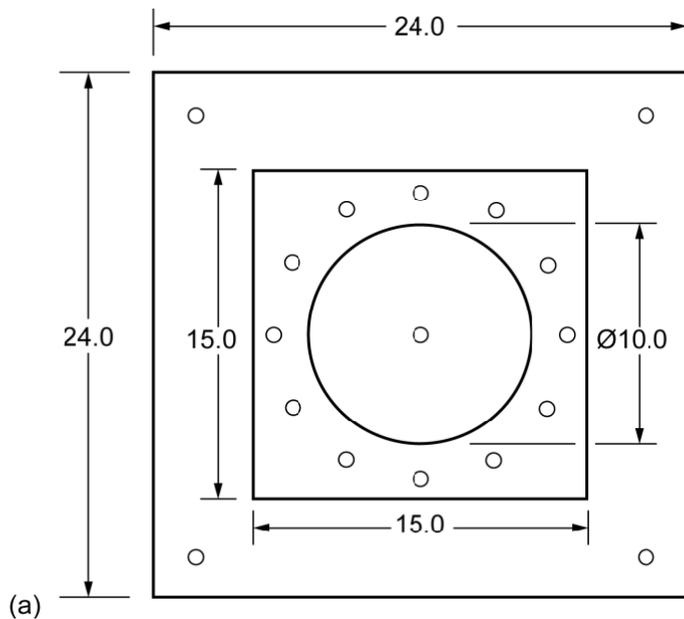
NASA/TM—2013-217869

DOT/FAA/TC-12/58



## Impact Testing of Aluminum 2024 and Titanium 6Al-4V for Material Model Development

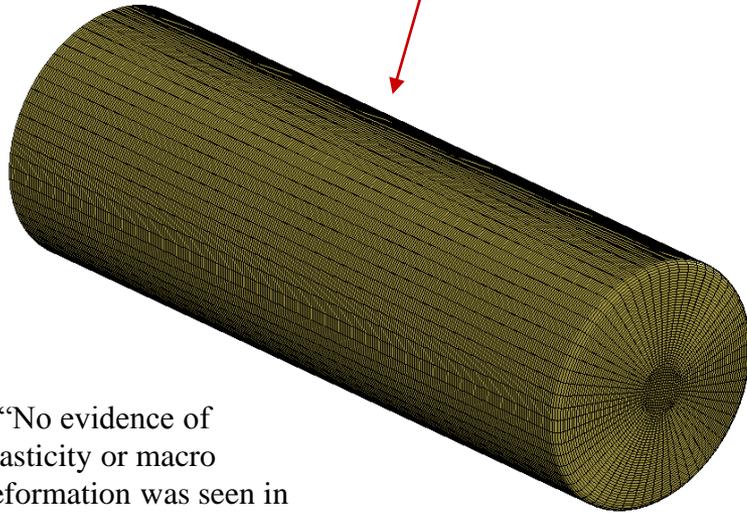
*J. Michael Pereira, Duane M. Revilock, Bradley A. Lerch and Charles R. Ruggeri  
Glenn Research Center, Cleveland, Ohio*



## Finite Element Model

MAT\_ELASTIC\*

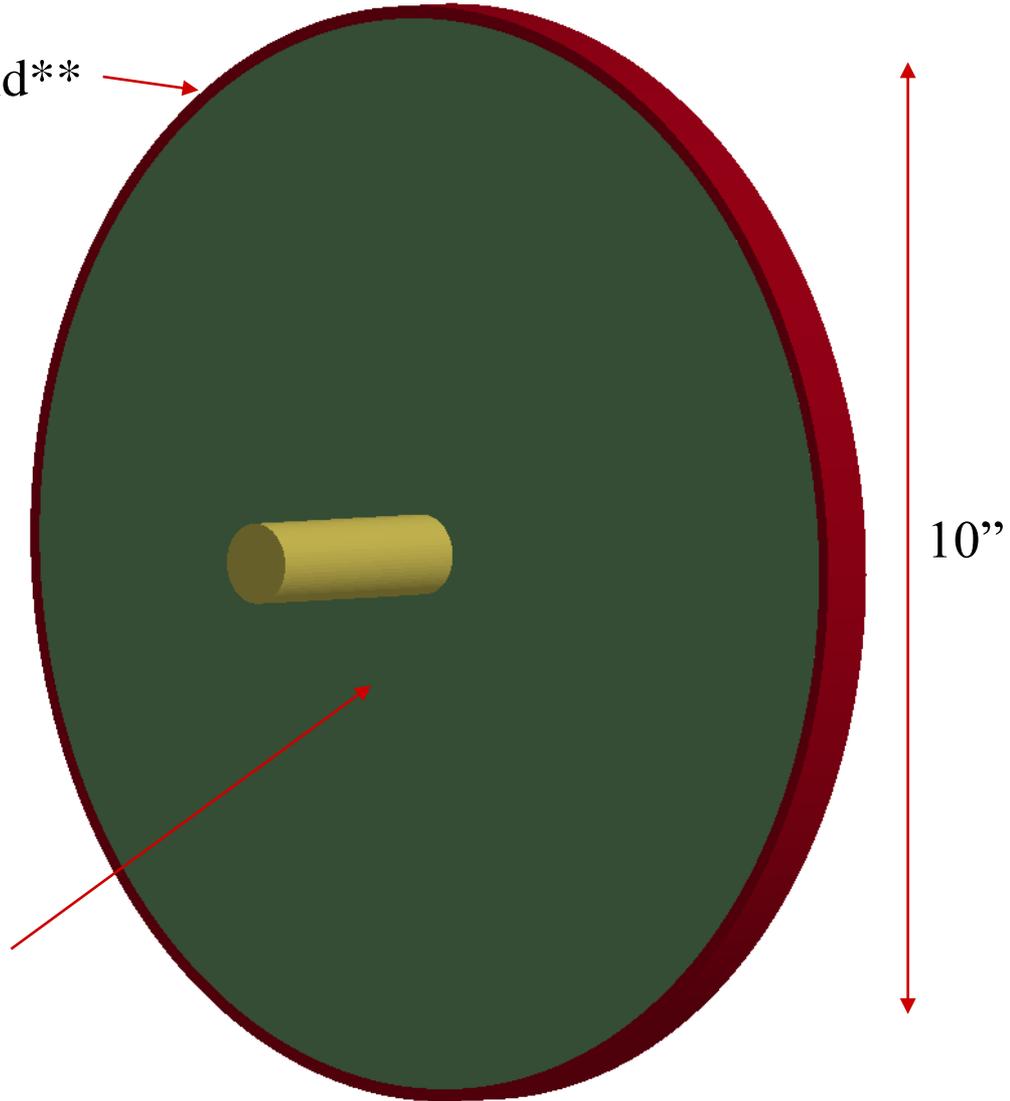
Rigid\*\*



\* “No evidence of plasticity or macro deformation was seen in the hardened A2 projectiles.”

\*\* “This leads to the conclusion that the boundary conditions do not play a role in panels of this thickness (0.14”)”

MAT\_224  
0.2 mm mesh  
1.8M elements



10"

## Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V

Test	Projectile mass, gram	Projectile impact velocity, ft/sec	Projectile exit velocity, ft/sec
DB177	126.3	896	475
DB178	126.4	865	424
DB179	126.2	713	241
DB180	126.2	646	152
DB182	126.4	527	0
DB184	126.3	581	0
DB185	126.2	597	0
DB186	126.4	578	0
DB192	126.3	630	0
DB193	126.3	629	104
DB195	126.3	616	0

\* Assume zero degree impact angle

## Triaxialities in ballistic impact tests

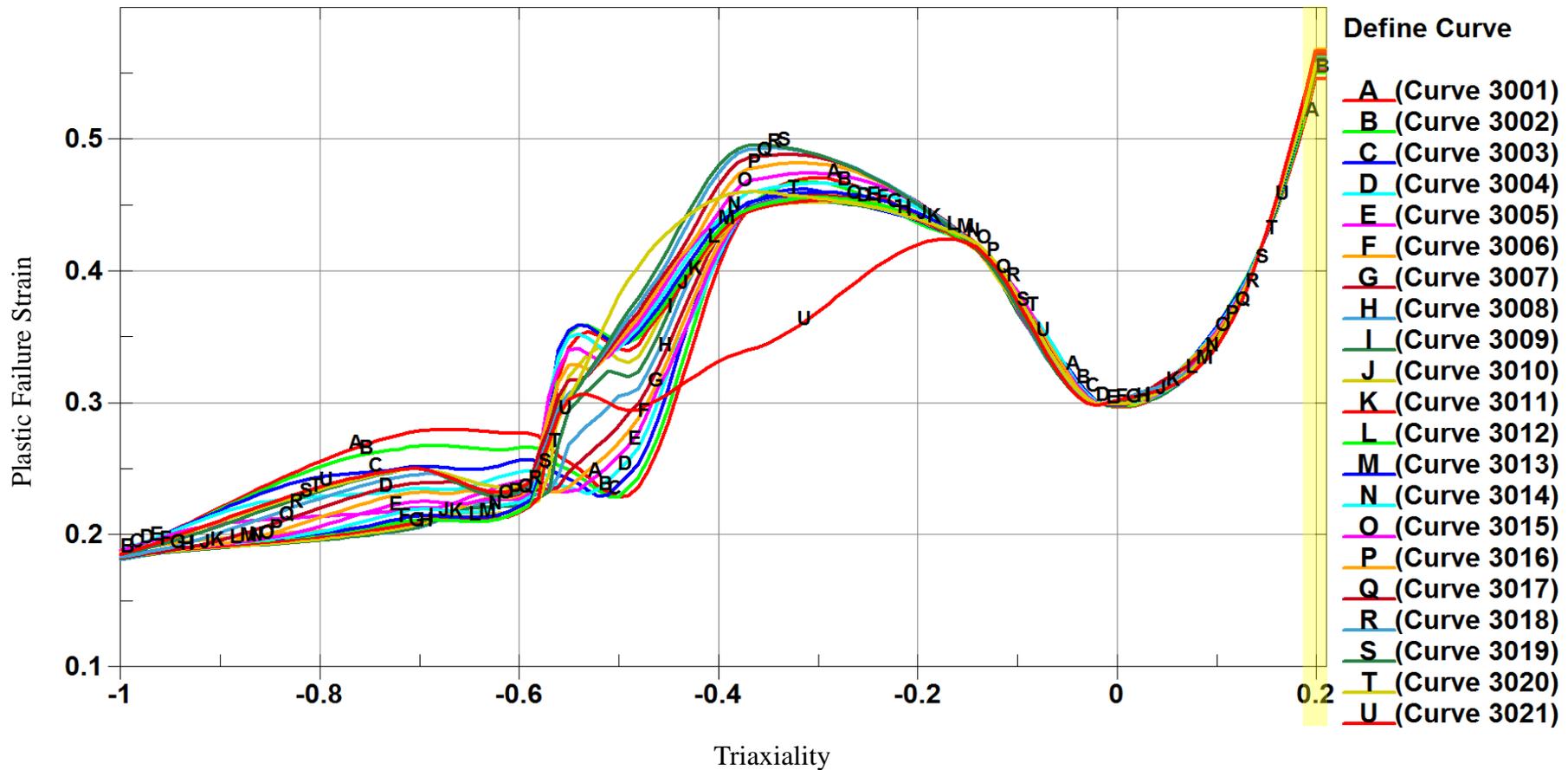
- Triaxialities in and around the ASB were between 0.2 and 5 showing a very high hydrostatic pressure contents

$$\boldsymbol{\sigma} = \begin{pmatrix} -p & 0 & 0 \\ 0 & -p & 0 \\ 0 & 0 & -p \end{pmatrix} + \begin{pmatrix} \tau & 0 & 0 \\ 0 & 0 & 0 \\ 0 & 0 & -\tau \end{pmatrix} \quad p > 0 \text{ and } \tau > 0 \quad t = \frac{p}{\tau\sqrt{3}} > 0$$

*if  $p > \tau$  all stress components  $< 0$*

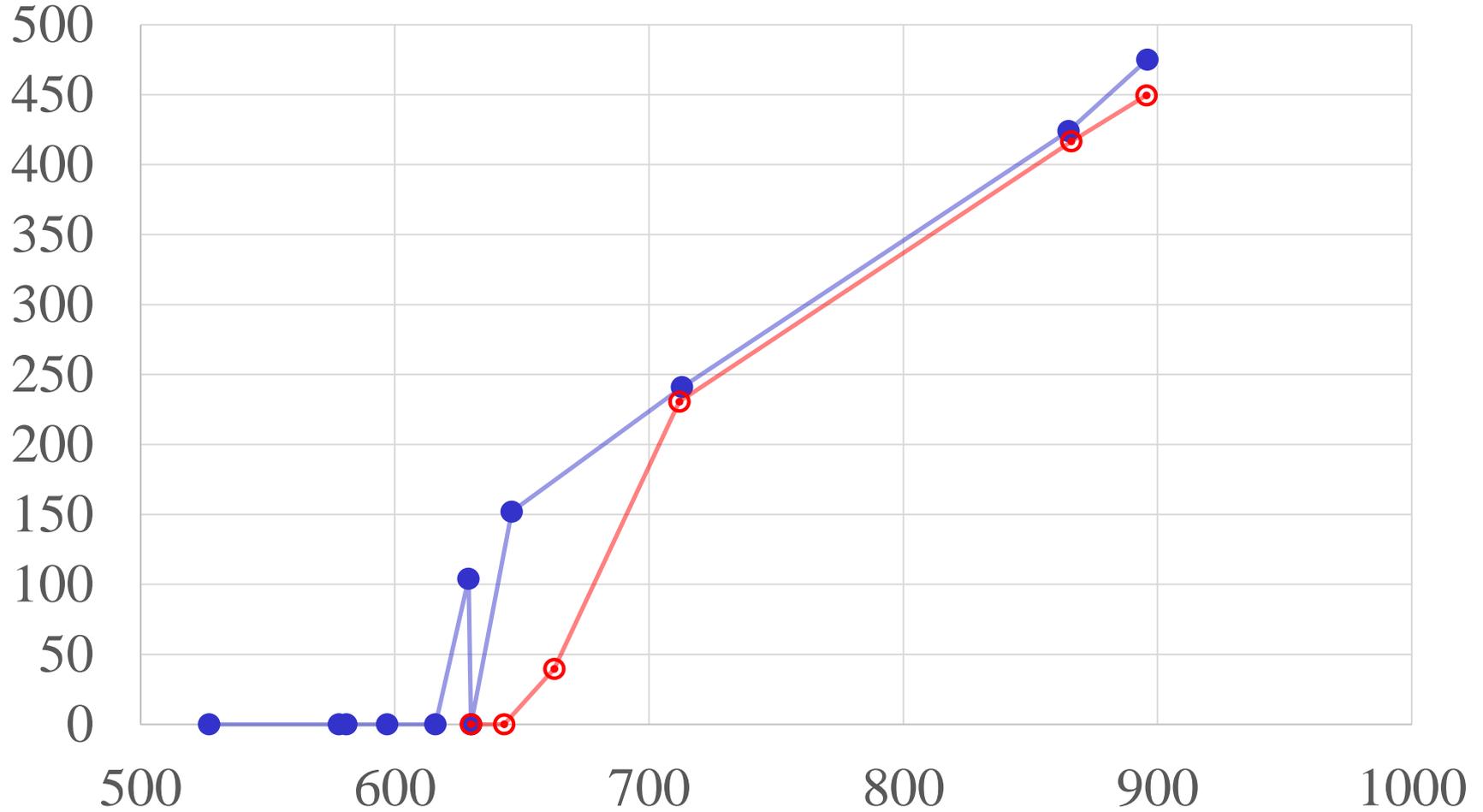
- Our failure surface had test data for triaxialities up to 0.147 (combined loading test LR4)
- The decisive failure in the ballistic impact tests (ASB) was completely in a dark region of the failure surface ( triaxialities  $> 0.2$  )
- An update of the failure surface was needed
- The assumption that no failure occurs when all 3 principal stress components are negative ( triaxiality  $> 0.3333$ ) is FALSE

# Adjustments to Failure Surface

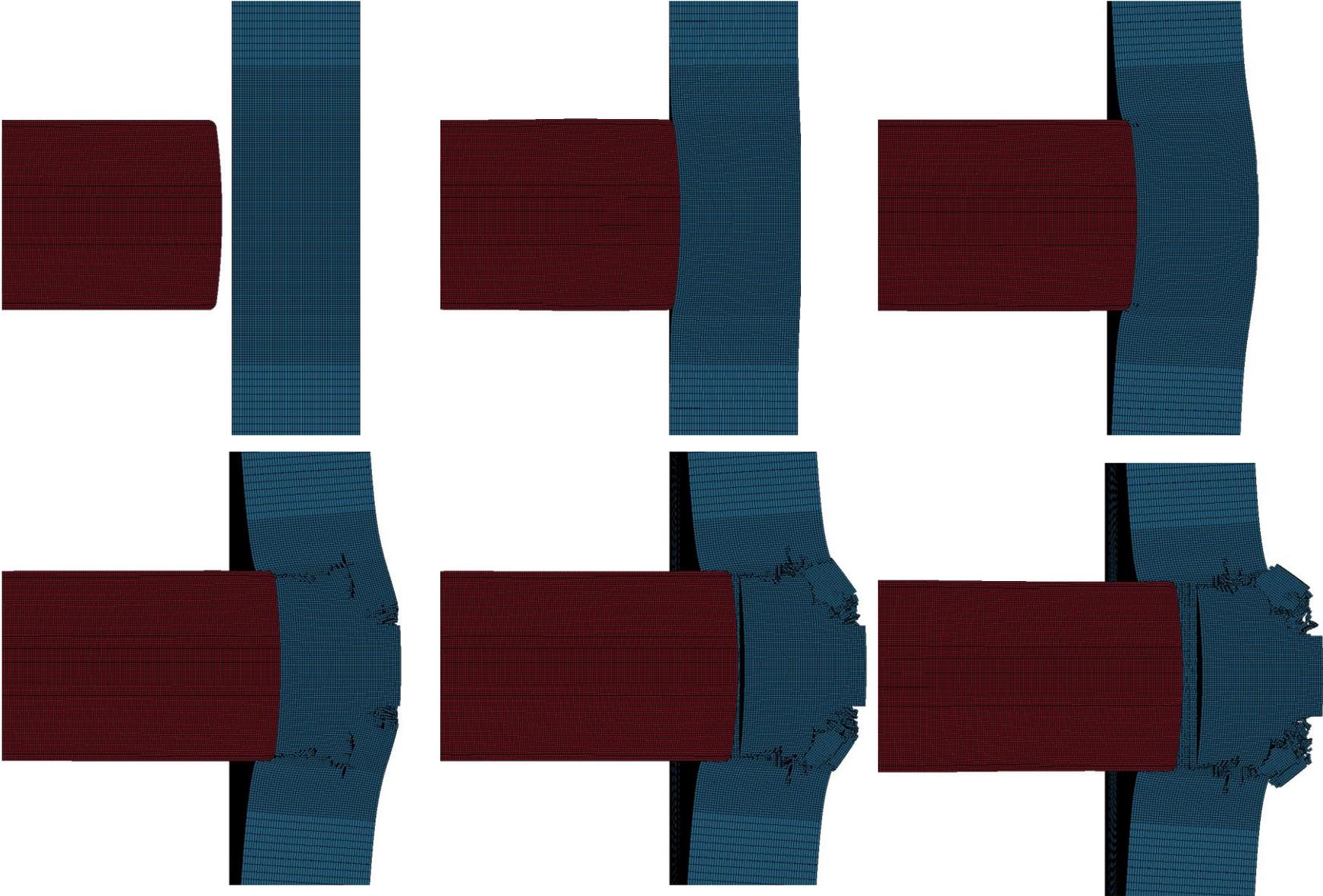


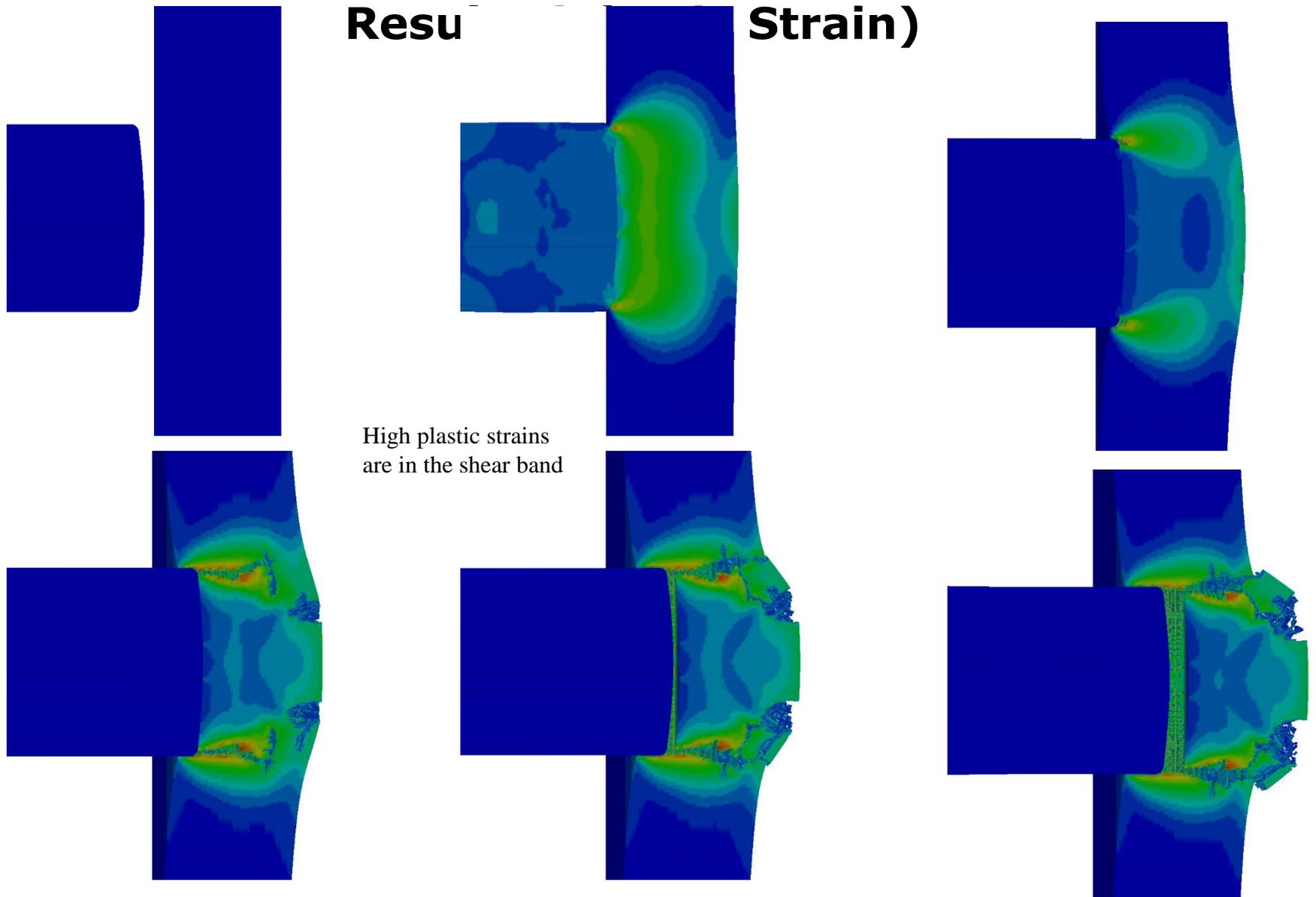
Introduction of a plateau in the failure surface at 60% for triaxialities > 0.2, NO previous tests are affected

## Simulation Results



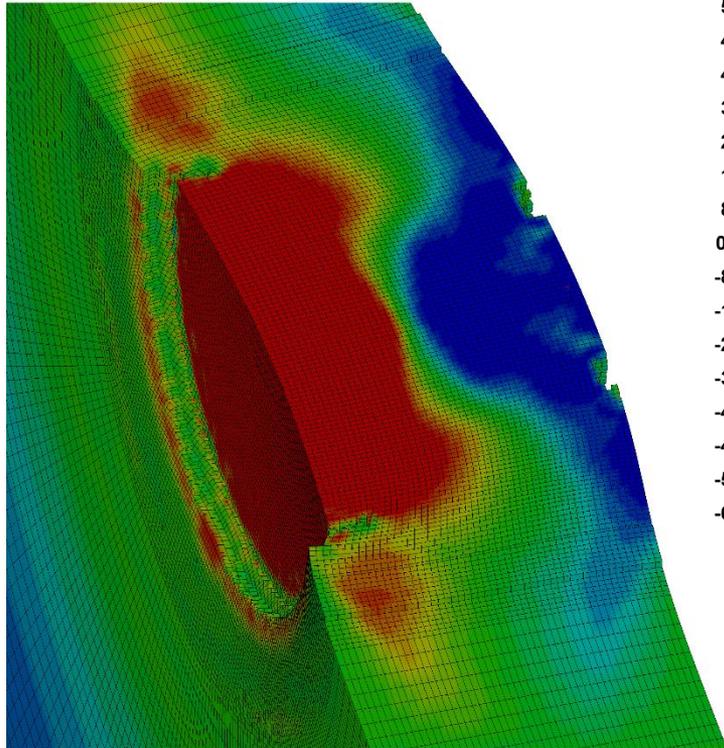
# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



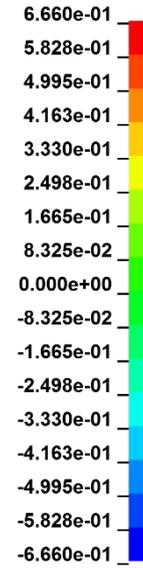


## Results (Triaxiality and Lode)

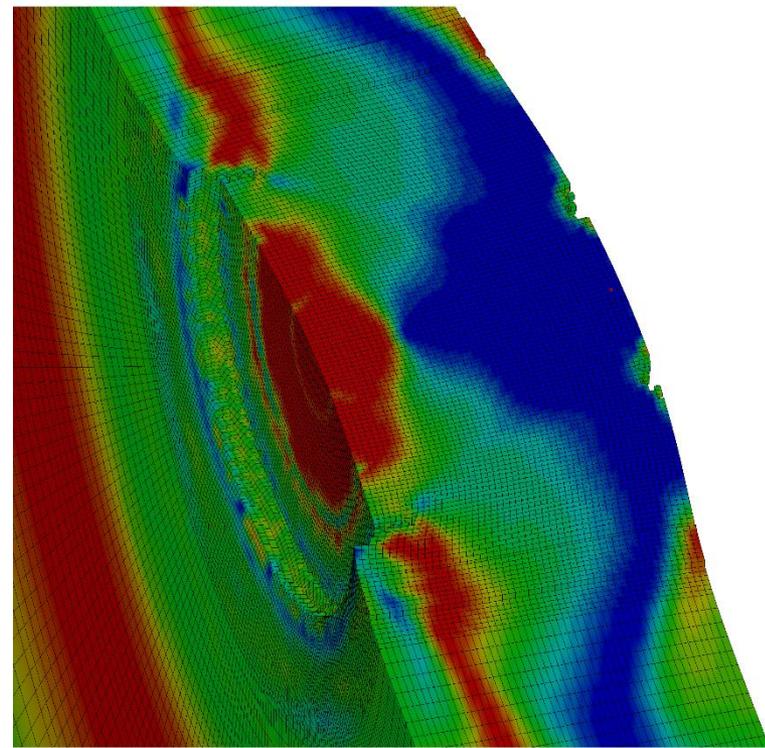
Triaxiality



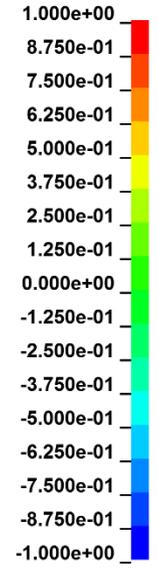
Fringe Levels



Lode Angle



Fringe Levels

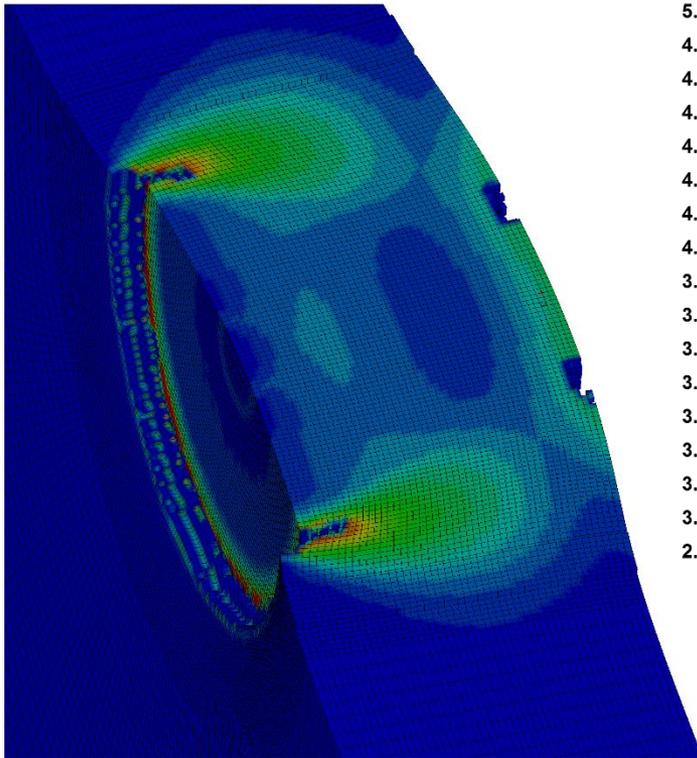


Triaxialities exceed 0.666

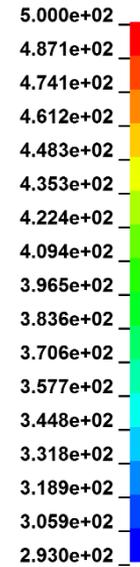
Lode parameters in the ASB between -0.5 and -1

# Results (Temperature and Strain Rate)

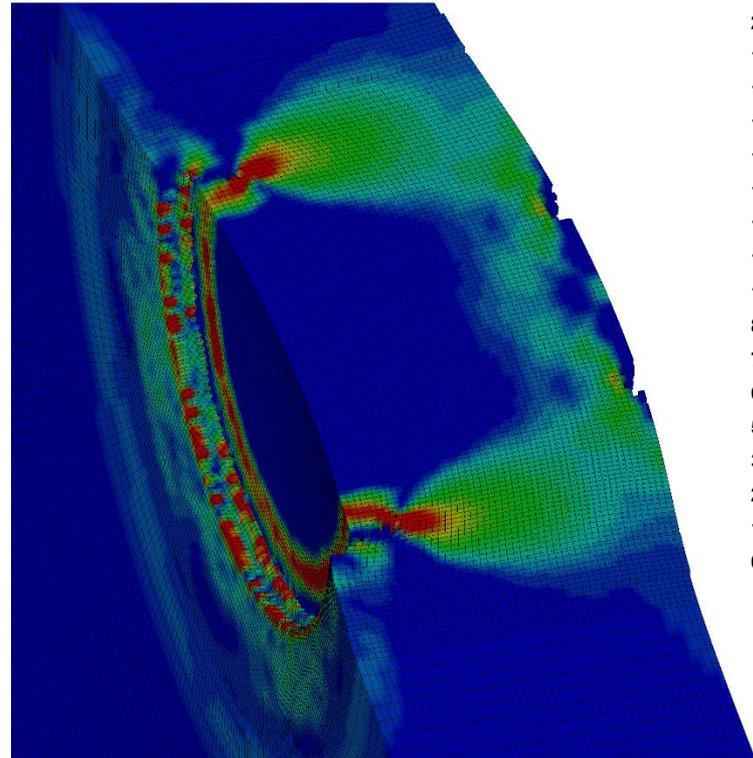
Temperature



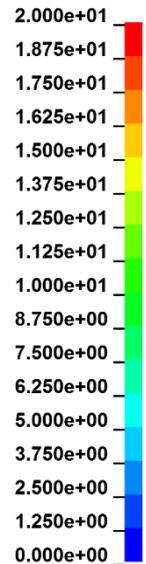
Fringe Levels



Strain Rate



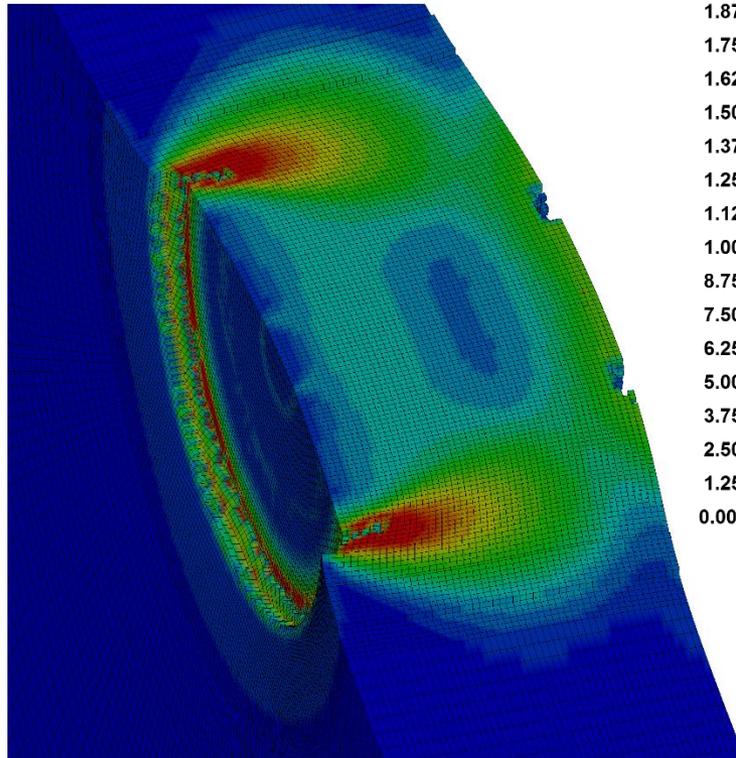
Fringe Levels



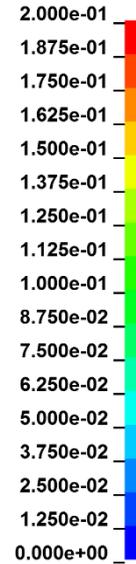
Temperature rise of around 200C in the shear band, negligible in the spall plane  
Strain rate values of 20000/s and higher

# Results (Plastic Strain and Damage)

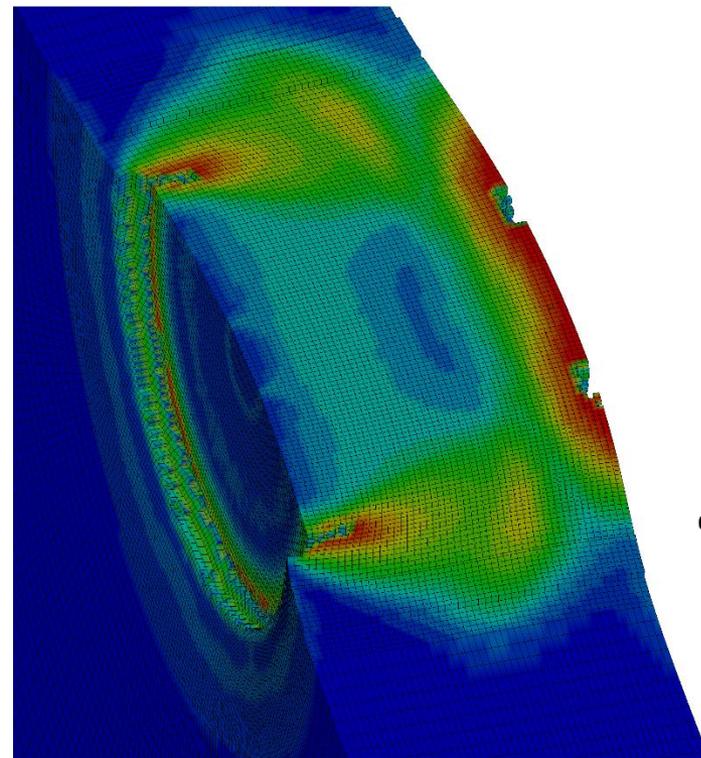
Plastic Strain



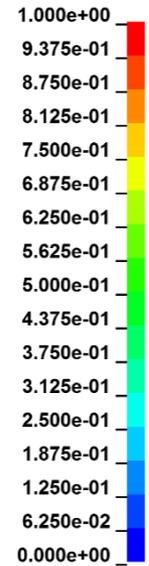
Fringe Levels



Damage



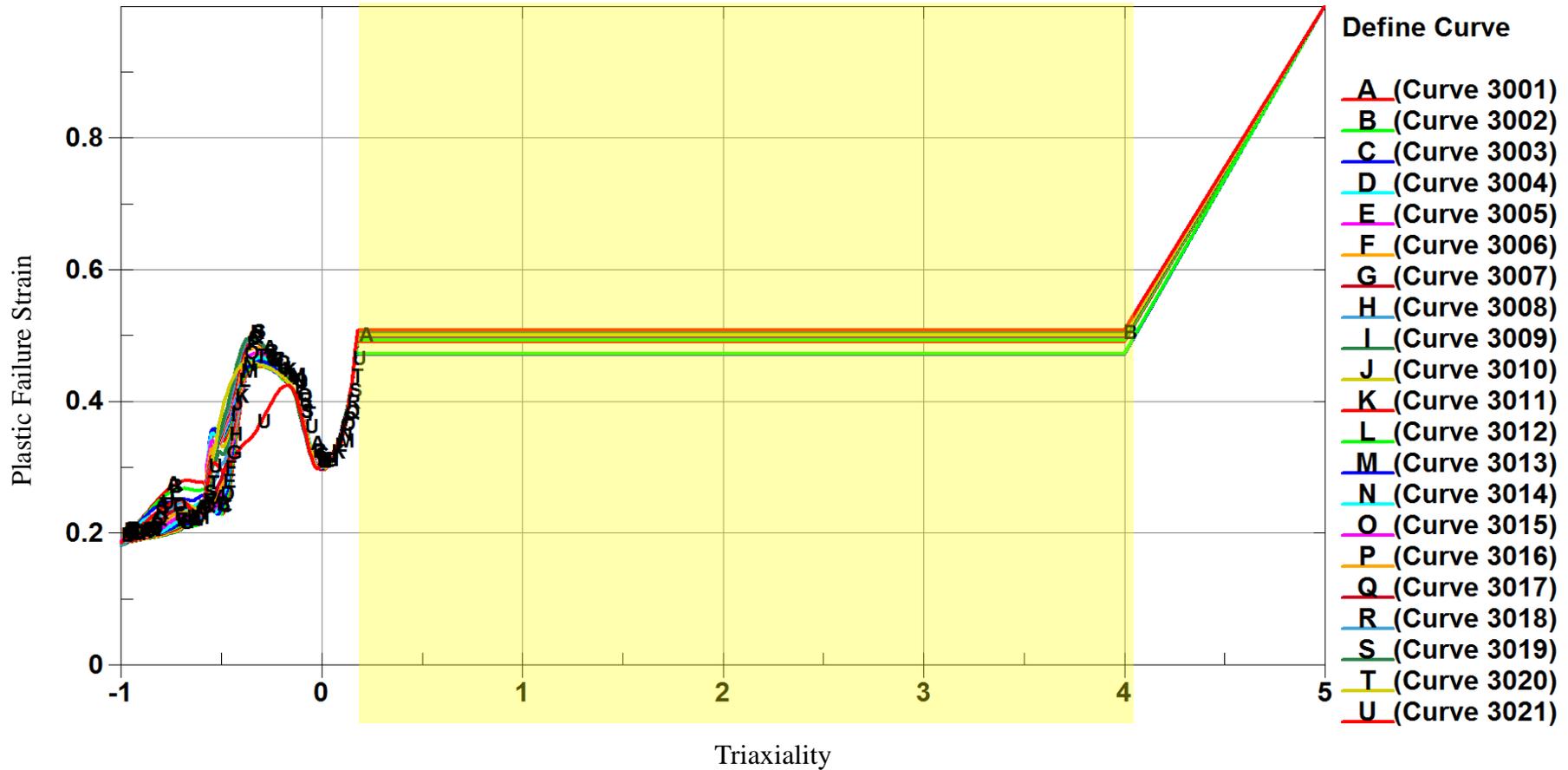
Fringe Levels



## **Discussion of results**

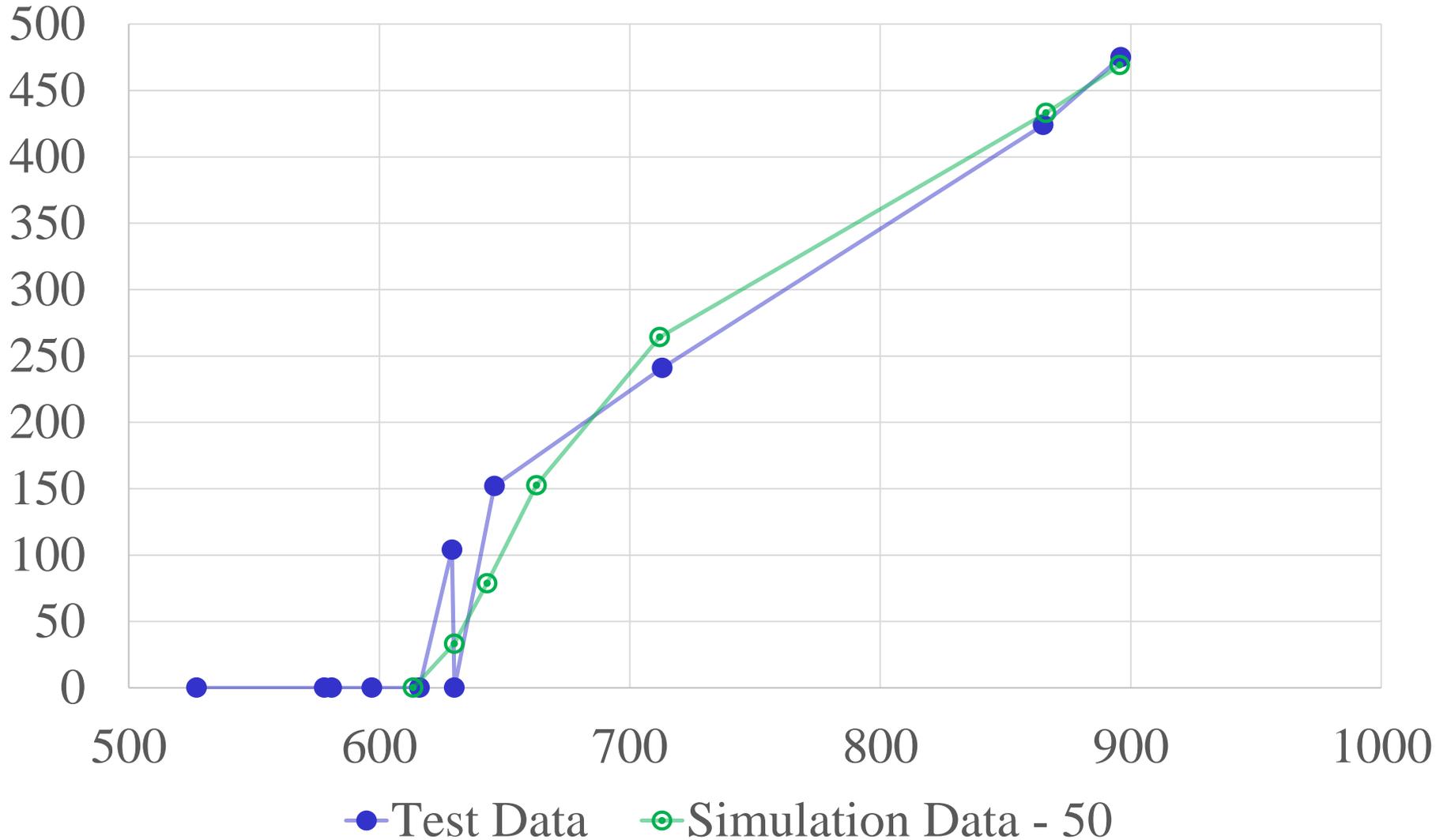
- **Some spalling occurs in the simulation, more than observed in the test, this could be due to the ASB forming somewhat too late**
- **According to Rittel : ' the influence of the hydrostatic pressure on the formation as ASB is still an open question'**
- **Our shearband can propagate too slow due to the 0.2mm mesh size, according to Meyers the width of shear bands in Ti-6-4 is between 2 and 20 micron, according to Borvik the mesh size should not exceed the half width of the shear band**

## Adjustments to the Failure Surface

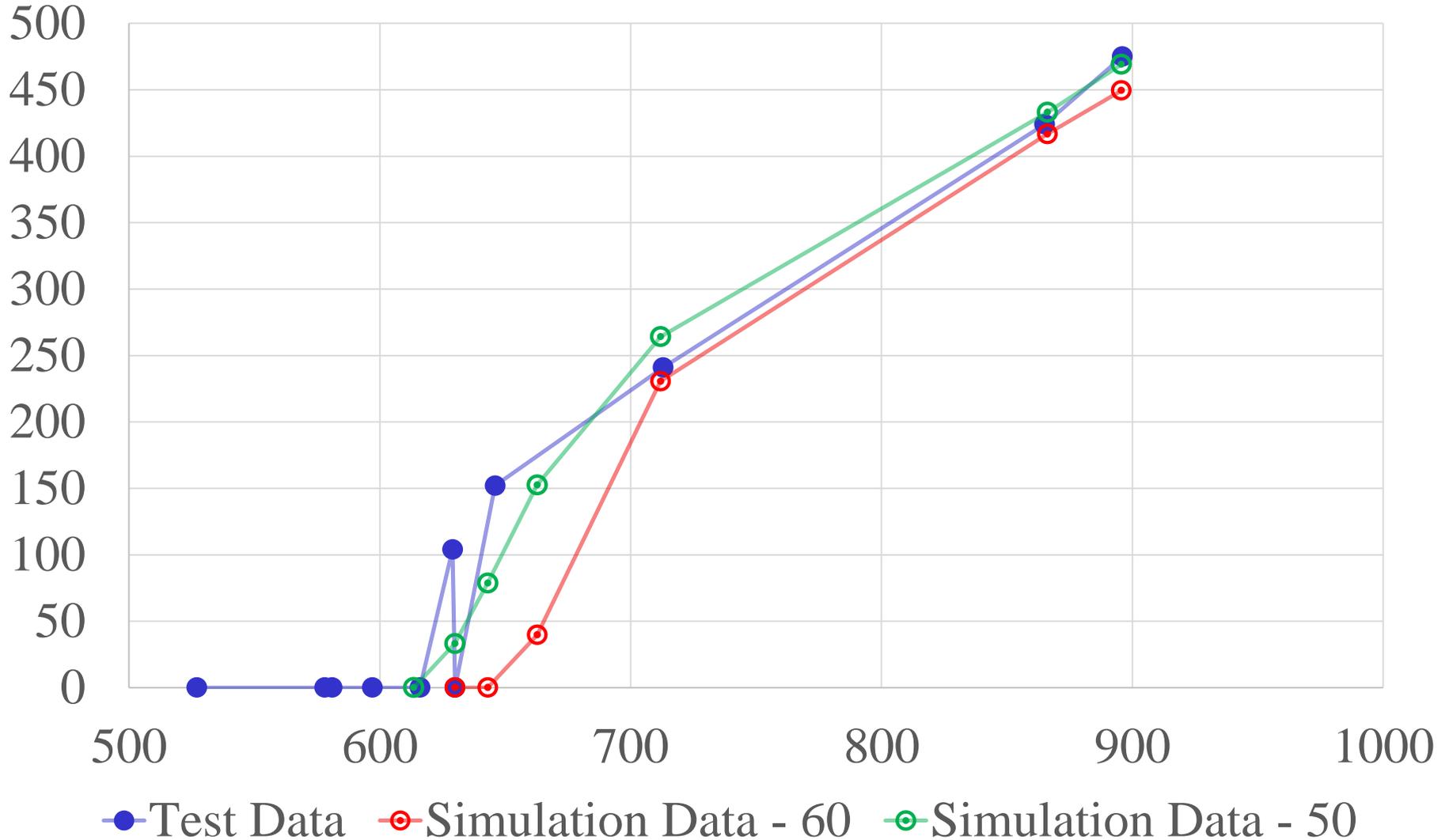


Shift of the plateau from 60% to 55%

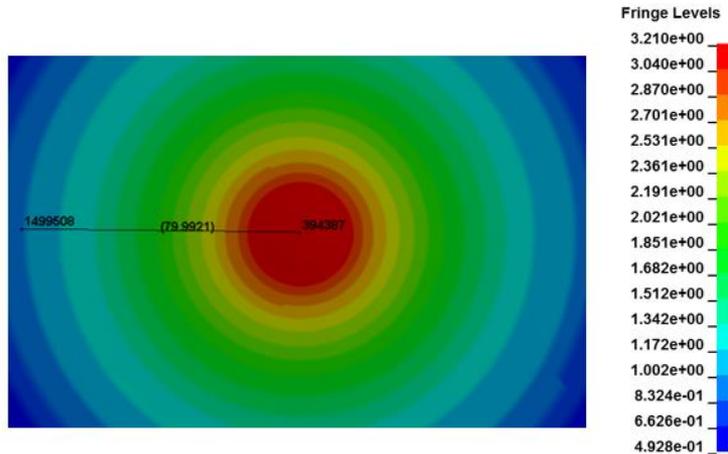
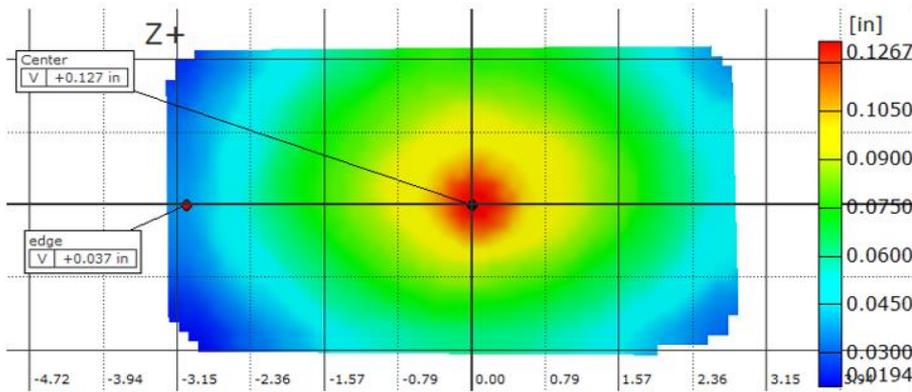
## Results



## Comparison



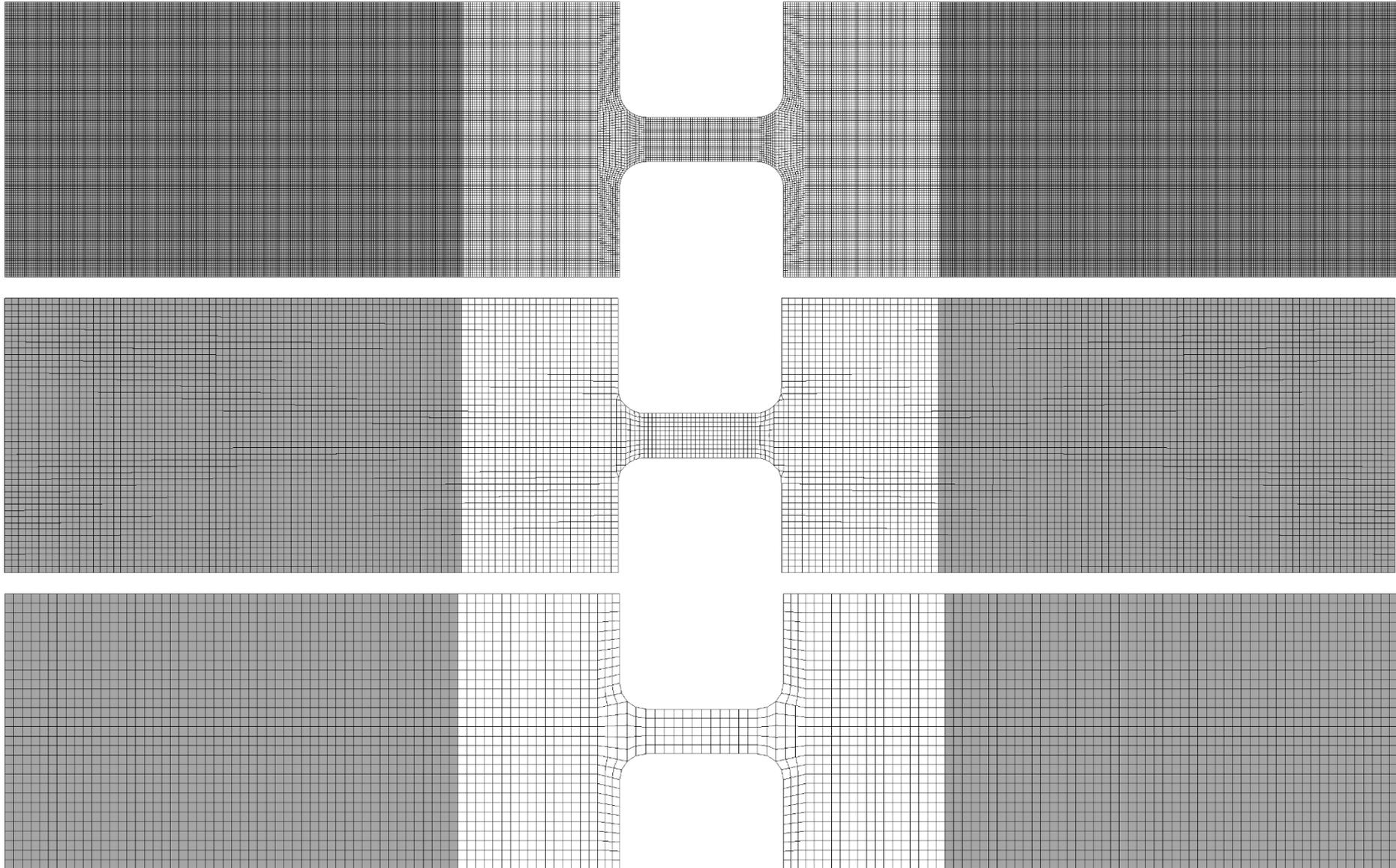
# Comparison of displacements for DB182 ( 527 ft/sec)



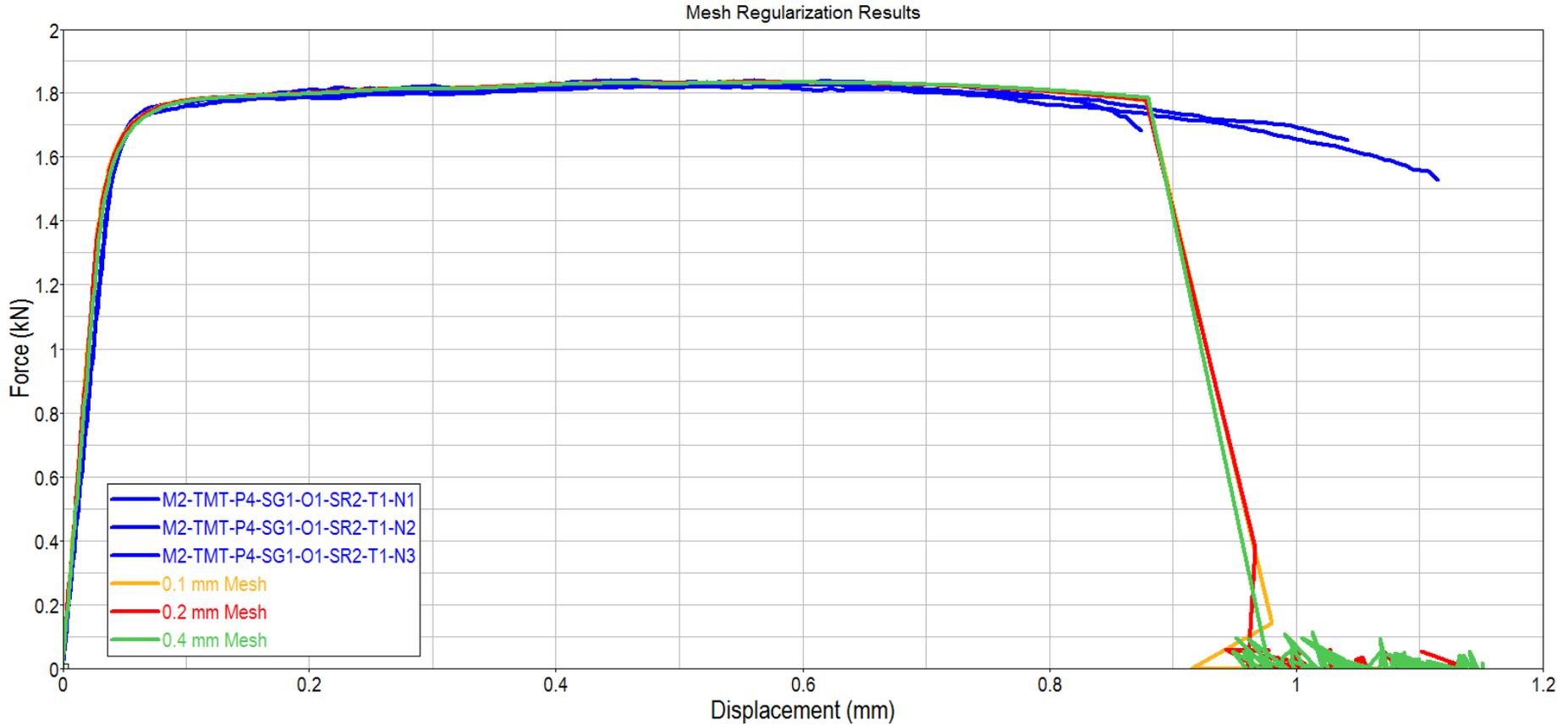
## **Regularization Curve**

- **A regularization curve was developed using three mesh sizes**
  - **0.1 mm**
  - **0.2 mm**
  - **0.4 mm**
- **Each specimen was simulated using the entire Ti-6-4 material model**
  - **Including failure regularization curve**
- **The regularization curve was adjusted with each iteration until the failure for each specimen occurred at the same displacement**

# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



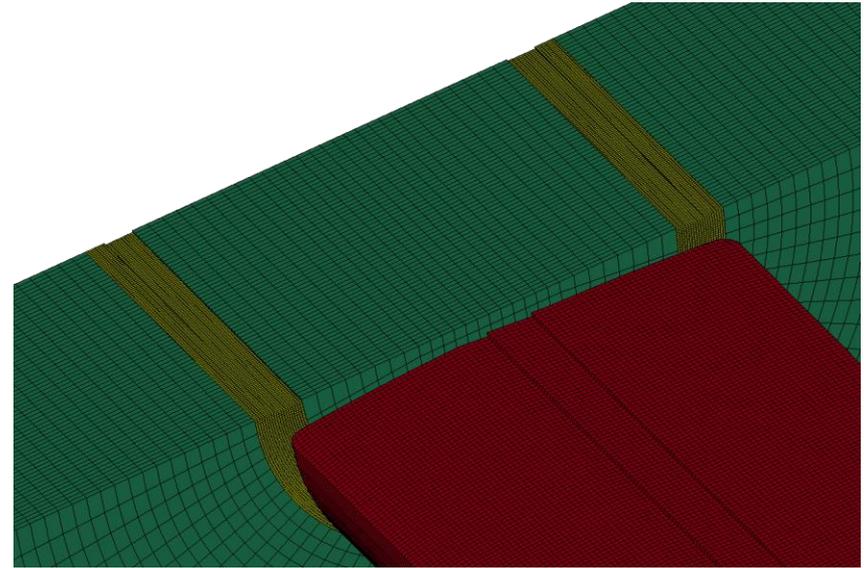
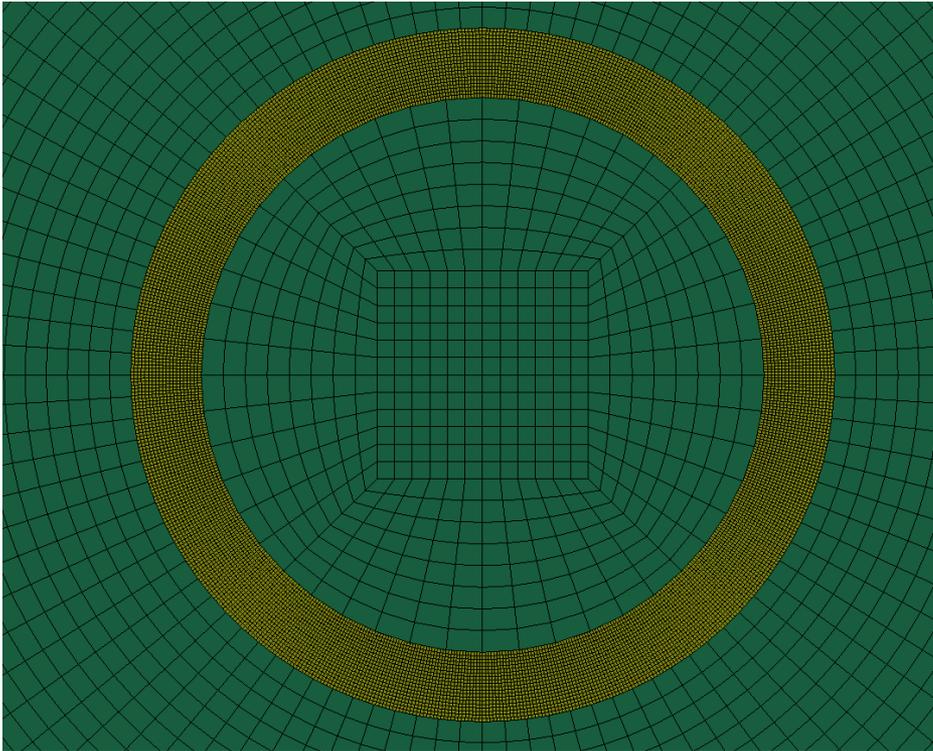
# Regularization Results



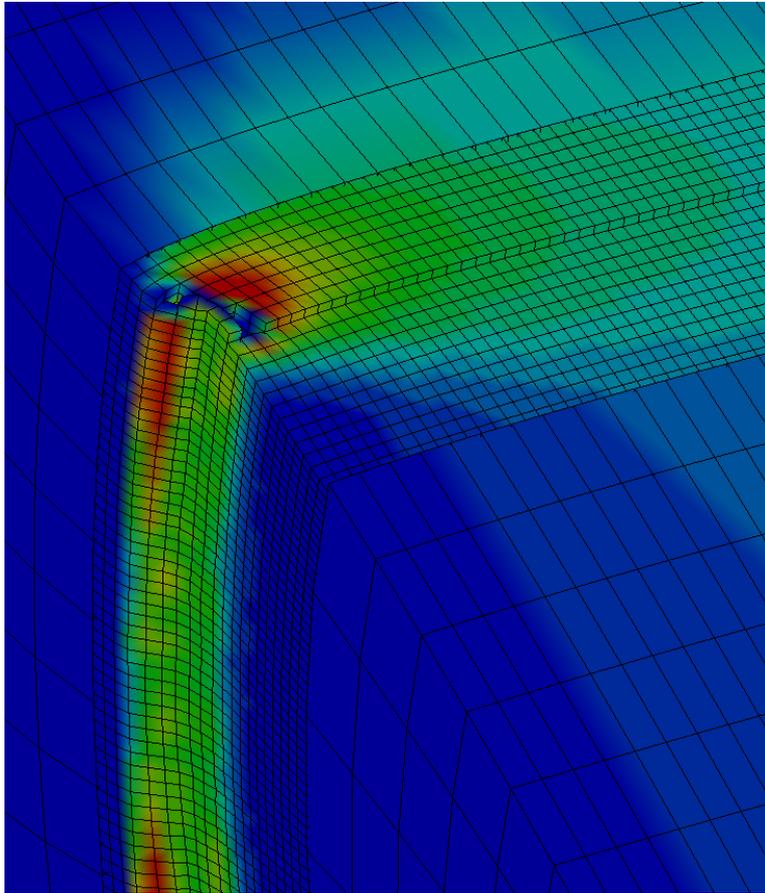


## NASA Ballistic Test (0.1 mm mesh)

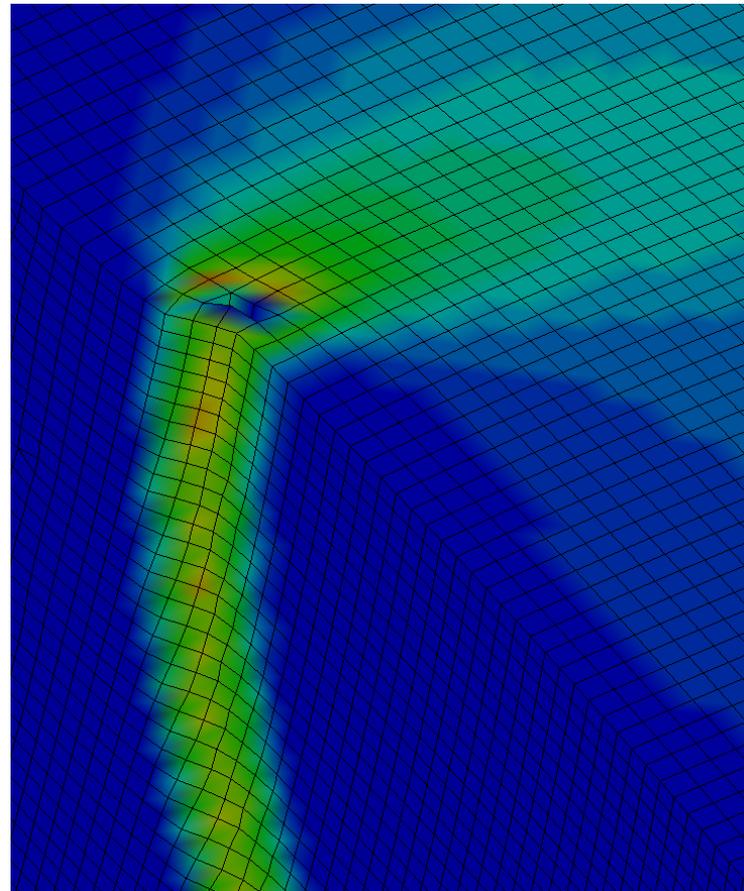
- NASA Ballistic test plate was re-meshed with 0.1 mm elements in the area of the adiabatic shear band
  - 0.1 mm elements
  - Connected with \*CONTACT\_TIED\_SURFACE\_TO\_SURFACE
  - 217 m/s impact velocity



## Temperature Comparison (0.02ms)

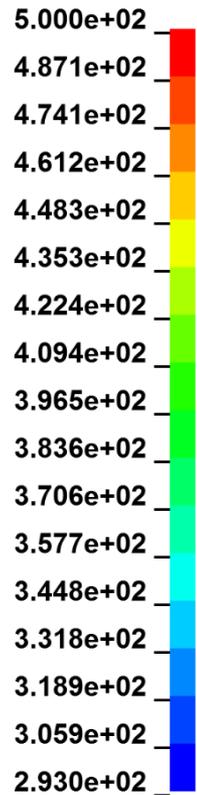


Max Temperature = 634 K

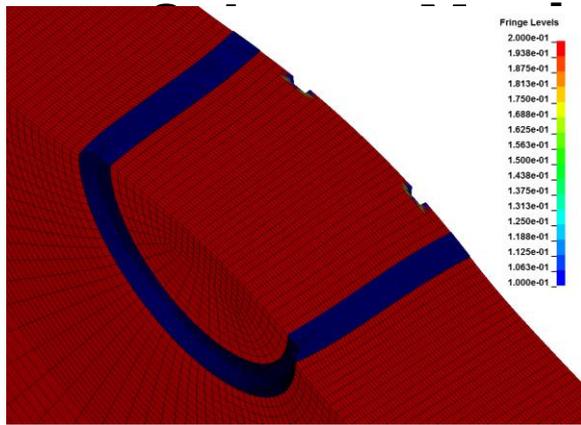


Max Temperature = 569 K

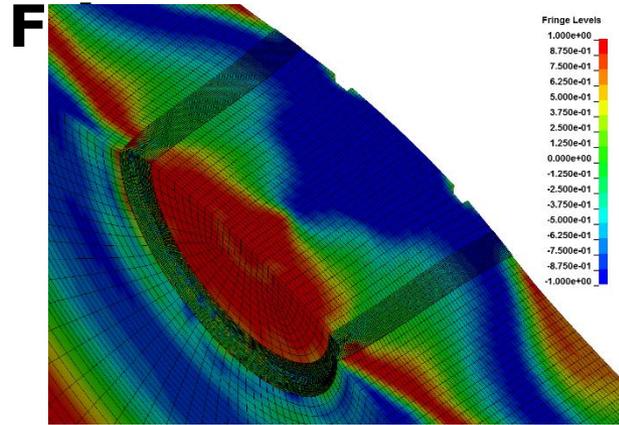
Fringe Levels



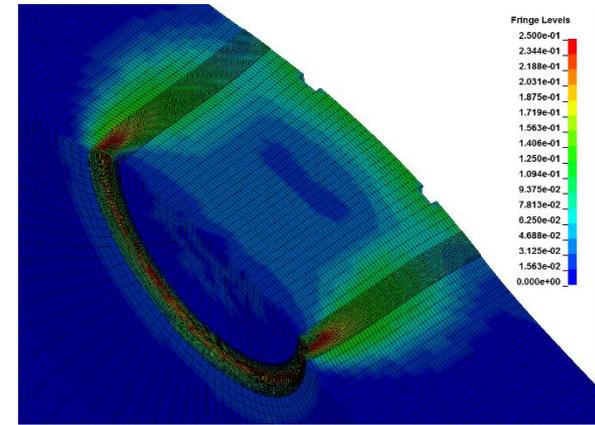
# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



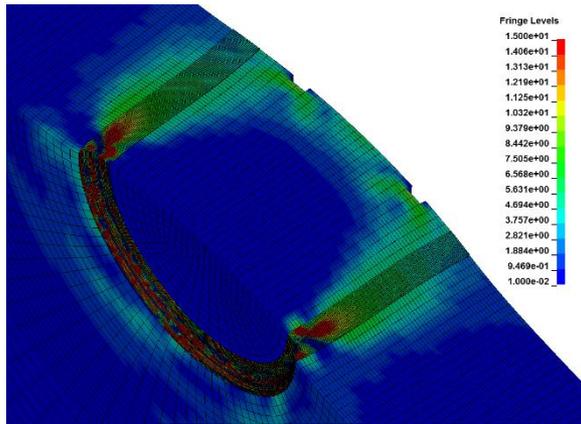
Element Size



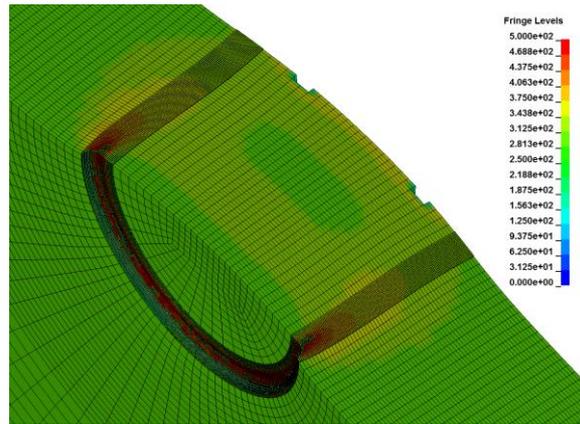
Lode Parameter



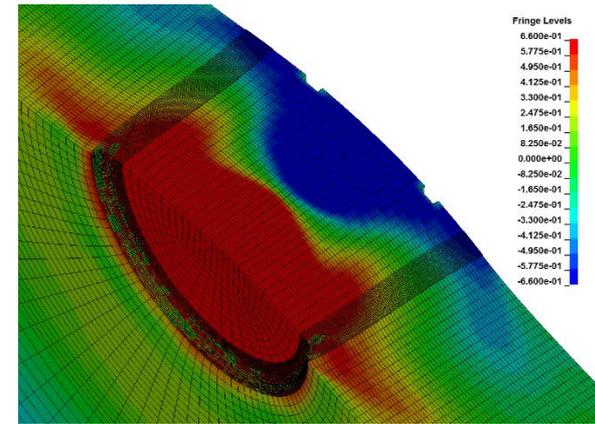
Plastic Strain



Strain Rate

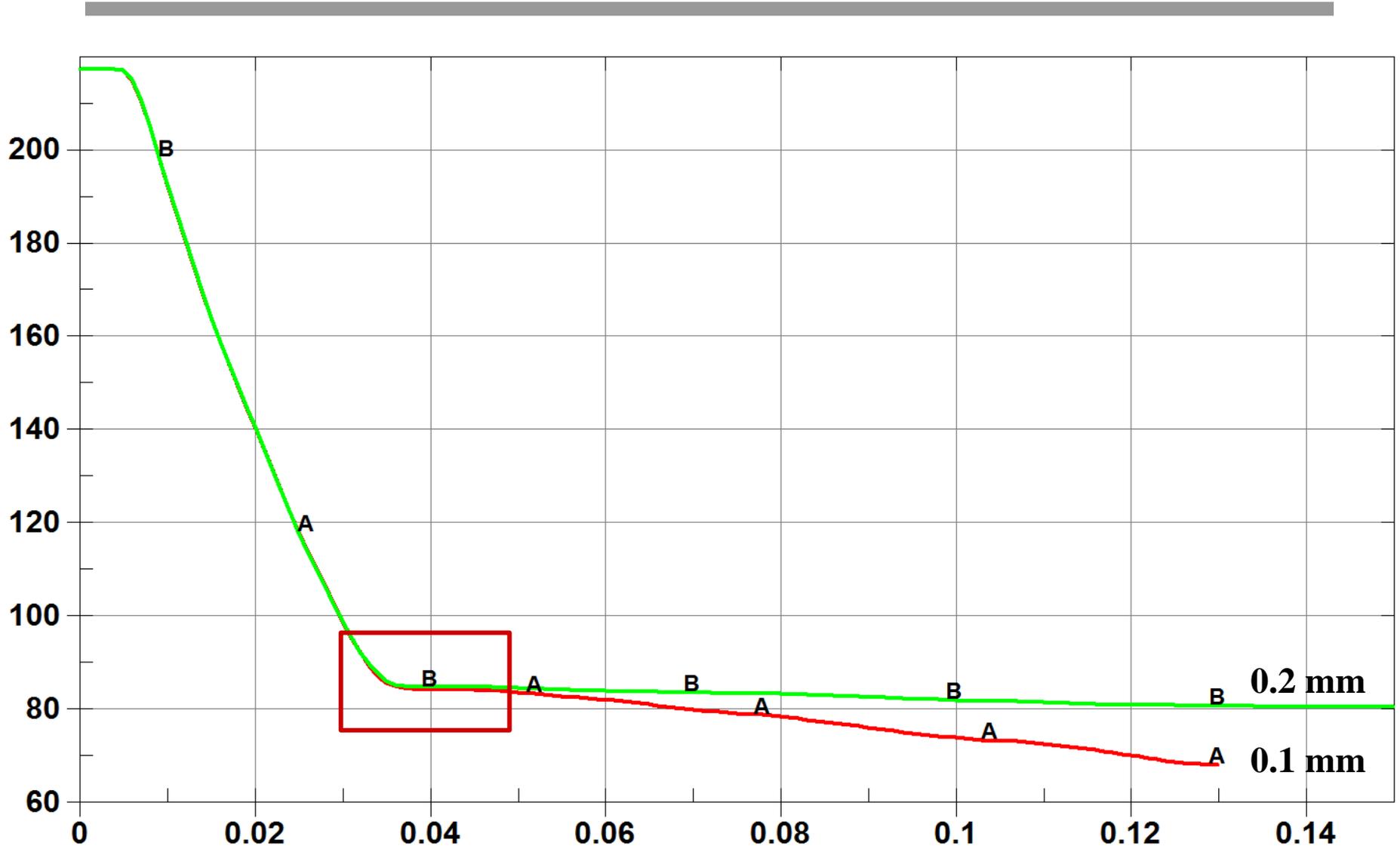


Temperature

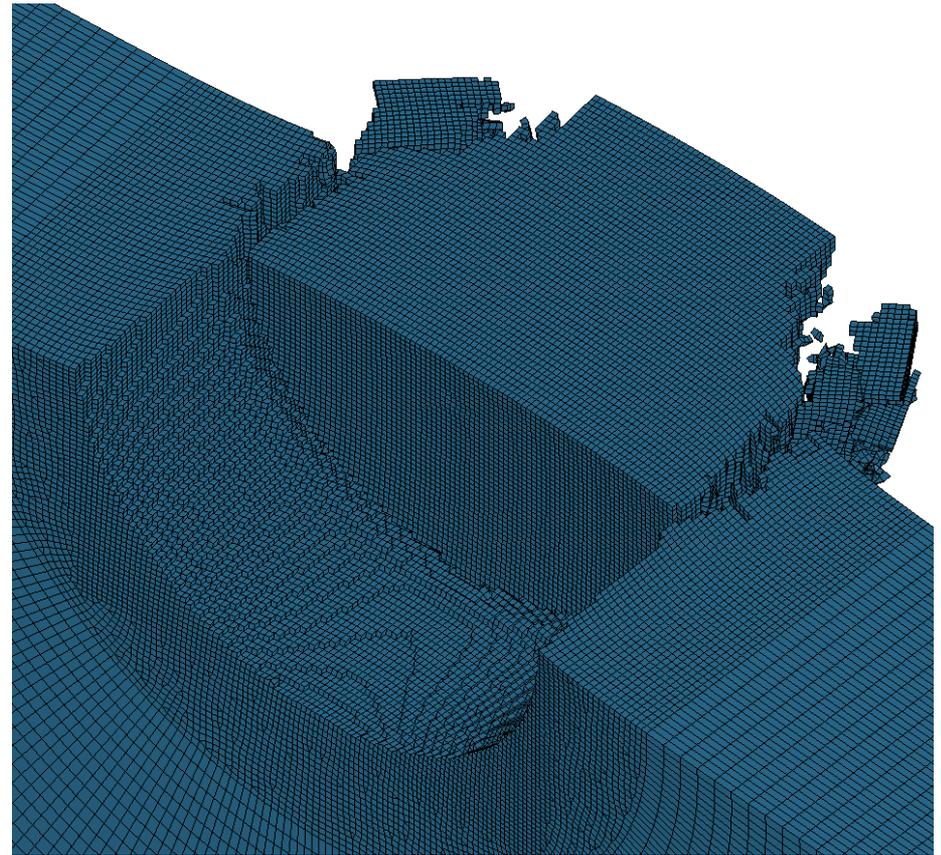
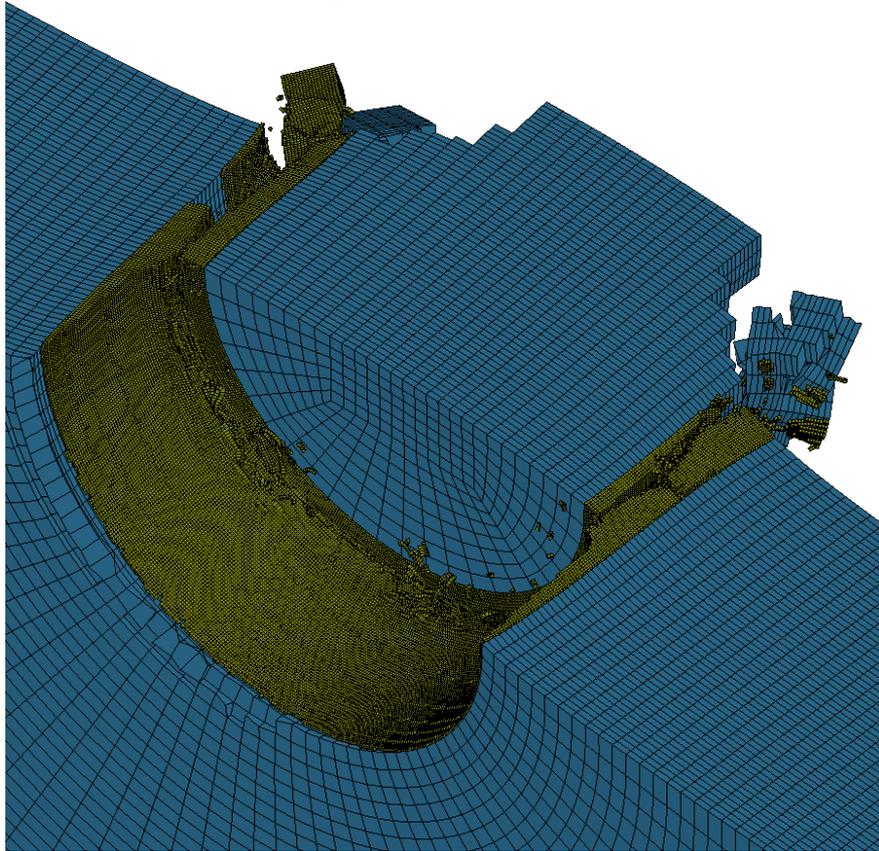


Triaxiality

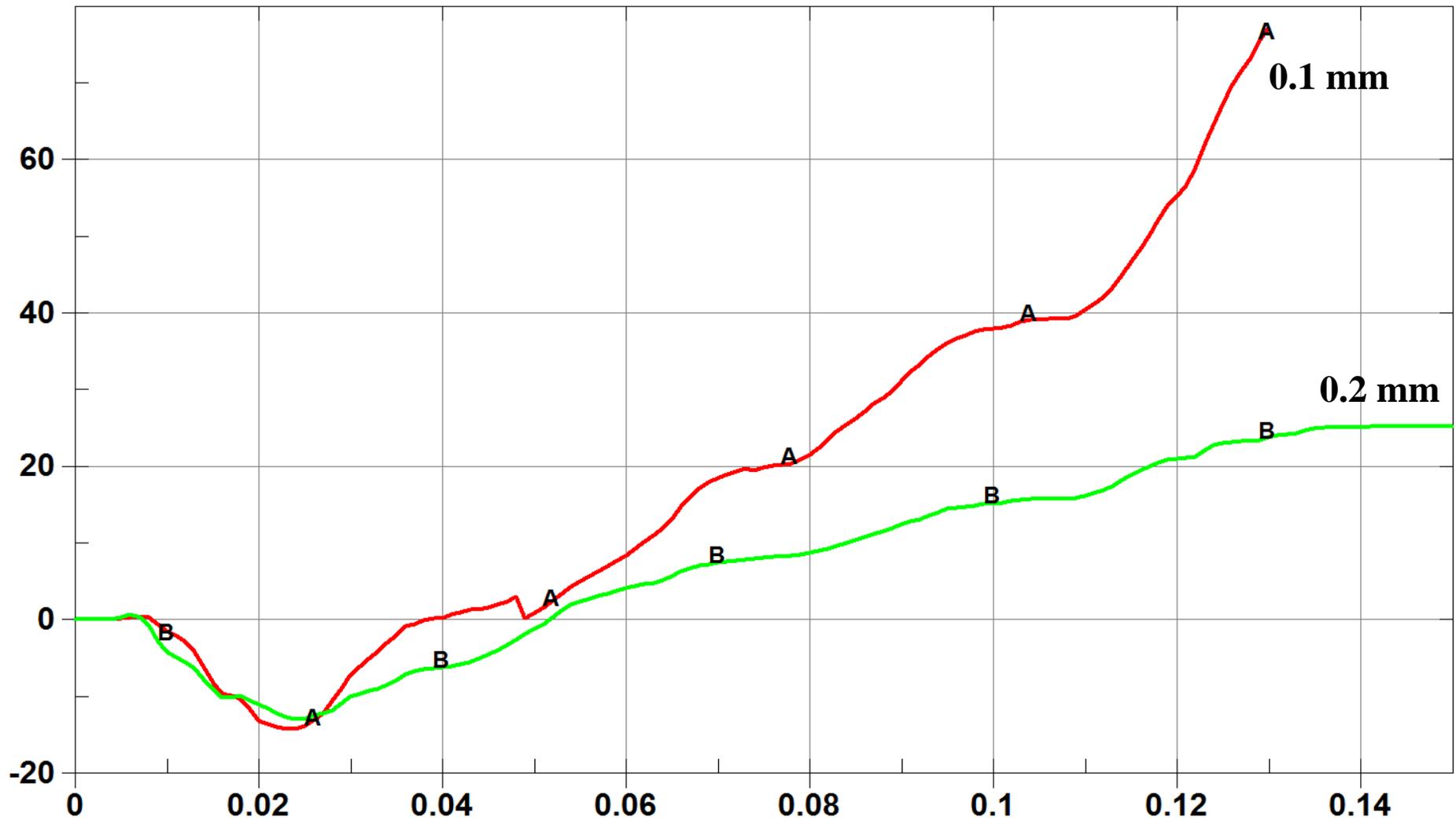
# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



## Sliding Surface



# Numerical Simulation of Dynamic Failure : ballistic tests on Titanium-6Al-4V



## **Conclusions**

- **The mechanisms of plastic instability and failure can be very different under static and dynamic loads**
- **Whereas regularization is often not needed to model shear failure under quasistatic, isothermal conditions, the situation may be different under high dynamic loads**
- **In the present study the mesh size was much larger than the width of the physical ASB, still a shear failure occurred in the simulation and realistic exit velocities could be obtained**
- **It is wrong to assume that no failure will occur in the region of stress space where triaxialities have large positive values, more attention will be given to this part of the failure surface in the future**